Short communication

Dielectric behavior of porous PMMA: From the micrometer to the nanometer scale

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Abstract
In recent years, there has been a significant interest of the scientific community on nanocellular polymeric foams as a possible next generation of materials with a low dielectric constant for microelectronics applications. In this work, the dielectric behavior of microcellular and nanocellular poly (methyl methacrylate) (PMMA) based foams has been characterized, both as a function of frequency and temperature, in order to analyze the effect of reducing the cell size to the nanoscale on the dielectric properties. Experimental results have shown clear differences in the dielectric behavior of the samples with cell sizes in the nanoscale as well as a sharp reduction of the dielectric constant when the porosity increases.

1. Introduction

Nanocellular foams have recently attracted significant attention in the microelectronics industry as a means of producing materials with a low dielectric constant (k) [1–6]. As the devices scale to smaller feature sizes, a new generation of low dielectric constant materials is needed to minimize cross talk and maximize signal propagation speed. The low dielectric constant of polymers and air (k = 1) make polymeric foams potential candidates as low k systems [7].

Poly (methyl methacrylate) (PMMA) is a thermoplastic material with a medium to high thermal stability and was one of the first polymers used in microelectronic systems [8]. In fact, this material is frequently used as a dielectric thin film [9–11]. In addition, PMMA has been recently used to produce nanocellular foams with different densities and cell sizes. Due to these reasons PMMA was selected for this study.

In microelectronics, the decreasing dimensions of the devices induces severe size restrictions to the cell size, which should ideally be an order of magnitude smaller than the thickness of the dielectric film [12], i.e. if a polymer foam is used as a dielectric film, as the thickness of the film could be of a few micrometers the cells should have dimensions in the nanoscale. Therefore, according to the literature, these required cell sizes are merely the result of the dimensions of the devices and it is not expected a significant difference between the well-known dielectric behavior of conventional, or microcellular foams, and that of the recently developed nanocellular foams [13]. However, a preliminary work by the authors has showed the emergence of a Maxwell Wagner Sillars (MWS) phenomenon in PMMA-based nanocellular foams at room temperature [14].

Further research to analyze this unexpected behavior as a function of temperature and frequency could provide important information for the application of nanocellular foams in microelectronic devices [14], where thermal stability is critical and one of the main drawbacks of polymer-based materials [15]. Therefore, the aim of the present study was to investigate in detail the dielectric behavior of both micro and nanocellular foams as a function of frequency and temperature showing the main differences between the behaviors of the two type of materials when cell size is reduced to the nano-scale.

2. Experimental section

PMMA was supplied by Arkema Company (France) in the form of pellets. This polymer presents a density (ρ) of 1180 kg/m³ and a glass transition temperature (Tg) around 112 °C.

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PMMA pellets were injected into pieces of 50 × 15 mm² with 3 mm in thickness (model DSM Xplore), and used later for foaming. Foaming experiments were performed at room temperature following the solid state foaming process (details can be found in the Supporting Information), modifying the saturation pressure between 10 and 30 MPa. The saturation pressures used are included in Table 1.

Dense skin of foamed samples was removed using a polishing machine (model LaboPol2-LaboForce3, Struers). Then solid and polished samples were machined using a precision cutting machine (model LaboPol2-LaboForce3, Struers). Dense skin of foamed samples was removed using a polishing machine (model LaboPol2-LaboForce3, Struers). Then solid and polished samples were machined using a precision cutting machine (model LaboPol2-LaboForce3, Struers). Dense skin of foamed samples was removed using a polishing machine (model LaboPol2-LaboForce3, Struers). Then solid and polished samples were machined using a precision cutting machine (model LaboPol2-LaboForce3, Struers). Dense skin of foamed samples was removed using a polishing machine (model LaboPol2-LaboForce3, Struers). Then solid and polished samples were machined using a precision cutting machine (model LaboPol2-LaboForce3, Struers).

DC resistivity data were determined according to ASTM D257-99 [16]. Solid and foamed samples were measured four times at ±500 V, −500 V, ±500 V, −500 V. Time of electrostatic was 60 s, and the time of discharge before making a measurement with reversed voltage was 4 min. Resistivity (R) was calculated as follows:

$$ R = \frac{A \cdot V}{t \cdot I} $$  (1)

where A and t are the area and thickness of the sample respectively, V is the applied voltage, and I is the intensity measured.

In the case of broadband dielectric measurements, foamed and solid samples were held in a dielectric cell between two parallel gold-plated electrodes. The thickness of the samples was taken as the distance between the electrodes and was determined using a micrometer gauge. The complex dielectric permittivity ($\varepsilon' - i\varepsilon''$) of the solid and foamed materials was measured over a frequency window of $10^{-2} < F/Hz < 10^6$ (F is the frequency of the applied electric field) in the temperature range from −20 to 110 °C. The amplitude of the alternating current (ac) electric signal applied to the samples was 1 V. Furthermore, the real part of the complex dielectric permittivity, $\varepsilon'$, (or dielectric constant, k) was modeled using the series model [17], the parallel model [17], and the Maxwell Garnett model [18].

3. Results and discussion

DC resistivity ($R$, Ω/cm) at room temperature of PMMA foams with pore sizes between 90 and 3290 nm (Table 1, representative SEM micrographs of the foams can be found in Fig. S1, see Supporting Information) did not follow a simple trend as a function of the relative density (Fig. 1, left). The resistivity of the foams increases by two orders of magnitude when the pore size falls from 1460 nm to 710 nm, despite a simultaneous and slight decrease of the relative density (that should reduce the resistivity).

This behavior was previously observed by J. Pinto and coworkers [14], who demonstrated the appearance of an electrical conductivity component and an interfacial polarization phenomena (or MWS) in nanocellular PMMA based foams at low frequencies. MWS occurs in heterogeneous materials, such as blends or composites, at the interfaces, leading to a separation of charges (Fig. 1, right). Furthermore, they found an increase in the electrical resistivity of PMMA based foams when cell size shifts from the micro to the nanoscale due to an increase tortuosity of the solid phase (tortuosity of the solid phase is the ratio between the minimum distance of any real path between two areas of the cellular material and the shortest distance between these two areas (Fig. 1, right)).

The appearance of both phenomena can clearly be observed in Fig. 1 left. A clear increase of the resistivity is observed when cell size changes from the micro (1460 nm) to the nanometer range (710 nm); this is due to a higher tortuosity of the solid phase in the nanocellular foam (Fig. 1, right). In addition, for materials with cells sizes below 700 nm a reduction of the resistivity when the cell size is reduced is detected. The reason for this behavior is a conductive mechanism at low frequencies related to the MWS, which is due to the accumulation of charges in opposing cell walls.

As mentioned before, nanocellular polymeric foams are required in the microelectronics due to the reduced thickness of the interconnects, a meter range. Once these changes are fully understood they should be introduced on more advanced theoretical models able to provide accurate predictions of the dielectric constant for both microcellular and nanocellular foams.

It should be noticed that the values of the dielectric constant of the materials included in this paper are far from the ones currently required by microelectronics industry ($k < 2$ [22]). However, the

<table>
<thead>
<tr>
<th>Saturation Pressure (MPa)</th>
<th>Sample cell size (Φ) (nm)</th>
<th>Relative density ($\rho_{rel}$)</th>
<th>K at 1 KHz</th>
<th>K Series Model [17]</th>
<th>K Parallel Model [17]</th>
<th>K Maxwell Garnett Model [18]</th>
</tr>
</thead>
<tbody>
<tr>
<td>0 (Solid)</td>
<td>− (Solid)</td>
<td>1</td>
<td>7.73</td>
<td>−</td>
<td>−</td>
<td>−</td>
</tr>
<tr>
<td>30</td>
<td>90</td>
<td>0.46</td>
<td>3.56</td>
<td>1.66</td>
<td>4.01</td>
<td>3.53</td>
</tr>
<tr>
<td>25</td>
<td>200</td>
<td>0.49</td>
<td>4.25</td>
<td>1.74</td>
<td>4.28</td>
<td>3.73</td>
</tr>
<tr>
<td>20</td>
<td>710</td>
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<td>4.50</td>
<td>1.85</td>
<td>4.63</td>
<td>3.99</td>
</tr>
<tr>
<td>15</td>
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<td>0.56</td>
<td>5.07</td>
<td>1.95</td>
<td>4.89</td>
<td>4.19</td>
</tr>
<tr>
<td>10</td>
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<td>0.66</td>
<td>5.61</td>
<td>2.35</td>
<td>5.76</td>
<td>4.90</td>
</tr>
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</table>
results presented here show a very promising way of achieving low dielectric constant materials by the production of low density nanocellular foams.

Second, it was analyzed if the previous evidences of a different behavior of microcellular and nanocellular foams could be confirmed in the frequency domain. With this aim the complex plot of the impedance (Z) or Nyquist plot was obtained for the solid, microcellular and nanocellular PMMA (Fig. 2). Nyquist curves include the entire range of frequencies measured and are used to obtain information about the equivalent electric circuit. In this study a clear change in the samples behaviour is observed. The solid sample presents a dominating capacitor behaviour over the entire frequency range. Nevertheless, since an ideal capacitor appears as a vertical straight line (infinite resistance), the slope that the solid samples presents indicates a slight resistive contribution most likely arising from the capacitor losses present in any dielectric material. This slope then starts to decrease with the presence of the cellular architecture reaching its minimum value in the nanocellular material. Therefore, the nanocellular PMMA develops a resistive behaviour and can be modeled as a capacitor in parallel with a resistor. Reduction of the capacitive behaviour found on nanoporous PMMA foams could be related to the progressive immobilization of the polymer chains leading to a progressive decrease of free dipoles able to rotate with the reduction of the pore size.

In short, the previous result confirms that the pore size present a clear influence on the dielectric behavior of PMMA foams. Moreover, dielectric measurements at room temperature confirmed previously published experimental results obtained by measuring the dynamic mechanic behavioral (tan δ) of microcellular and nanocellular PMMA [20]. These results showed that the tan δ at low and medium frequencies was increased when the cell size is reduced to the nanometer range (Fig. S2, see Supporting Information).

Finally, the thermal stability of these effects was analysed as a function of the cell size. In order to obtain this information, measurements in the temperature domain were also conducted. In the case of the shape parameters alpha and beta, no differences between microcellular and nanocellular foams were detected as a function of temperature. This behaviour must be further studied due to the previous evidences found in the tan δ. On the other hand, a decrease of the normalized dielectric strength, \( \Delta \varepsilon / \Delta \varepsilon_{\text{solid}} \) that was stable in the temperature range measured (from –20 to 110 °C) was observed as a function of cell size (Fig. 3).

This reduction of \( \Delta \varepsilon / \Delta \varepsilon_{\text{solid}} \) was previously observed only at room temperature [14] and is was related to a progressive decrease of the number of free dipoles able to rotate, i.e., it is related to the confinement effect of the solid phase in the nanocellular system. Therefore, this results proves that the modifications of the dielectric behavior of PMMA foams induced by the pore size are stable in the temperature domain; a relevant result from an applied point of view.

4. Conclusions

The dielectric properties of microcellular and nanocellular PMMA foams have been studied, finding a clear evolution from a
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Appendix A. Supplementary data

Supplementary data related to this article can be found at http://dx.doi.org/10.1016/j.polymer.2016.11.030.

References