# Lattice Dynamics of 4H-SiC by Inelastic X-ray Scattering 

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#### Abstract

We have measured the phonon dispersion relations in 4H-SiC by inelastic $x$ ray scattering (IXS) using monochromatized synchrotron radiation. The $\mathbf{q}$-space directions $\Gamma-\mathrm{K}-\mathrm{M}, \Gamma-\mathrm{M}$, and $\Gamma-\mathrm{A}$ were mapped out. Lattice dynamical calculations that allowed the prediction of phonon eigenvectors, as well as their symmetries, also helped in choosing the best scattering geometries. The IXS phonon data are compared with those previously obtained from low temperature photoluminescence measurements and from laser Raman spectroscopy.


## 1 Introduction

Silicon carbide is a wide band gap material which crystallizes in a large number of polytypes [1], exhibiting properties, such as high thermal conductivity, that make it a good candidate for optoelectronic applications. Despite the large number of articles dedicated to the vibrational properties of $\mathrm{SiC}[2,3,4]$, little experimental information is available covering the whole Brillouin zone (BZ). Such information has been recently obtained for $6 \mathrm{H}-\mathrm{SiC}[5]$ by inelastic neutron scattering (INS), and for $3 \mathrm{C}-\mathrm{SiC}$ by inelastic x-ray scattering (IXS) [6], a relatively new technique which employs highly monochromatized synchrotron radiation [7].

IXS allows the investigation of the phonon dispersion relations of crystals of submilimeter size, as demonstrated for the wide band gap materials diamond [8], AlN [9], and GaN [10], where large size samples are not available.

In [6] we reported IXS data for $3 \mathrm{C}-\mathrm{SiC}$ along several directions of the BZ and along the $\Gamma-\mathrm{A}$ direction for $4 \mathrm{H}-\mathrm{SiC}$. We present here more detailed results for the phonon dispersion of $4 \mathrm{H}-$ SiC along the $\Gamma-\mathrm{K}-\mathrm{M}, \Gamma-\mathrm{M}$ and $\Gamma-\mathrm{A}$ directions of the BZ obtained by IXS. Data extracted from photoluminescence for the M-point and from Raman spectroscopy for the $\Gamma$-point are analyzed and compared with our IXS results and with Born-von-Kármán lattice dynamical calculations [11]. These calculations were used to elucidate the best scattering geometries for our experiments and to characterize the symmetry of the different modes at the $\Gamma$ and M points.

## 2 Experimental details and calculation procedure

IXS measurements were performed for several samples with orientation (1 $\overline{2} 10$ ) and (0001), and typical sizes of $10 \times 5 \times 0.2 \mathrm{~mm}$, cut from the same single crystal. In order to decrease dead time due to sample mounting and alignment we kept the sample at room temperature.

The experiments were carried out at the European Synchrotron Radiation Facilities (ESRF) in Grenoble (ID28). Details of the resolution and calibration procedure can be found in [6].

In order to calculate the dispersion relations of $4 \mathrm{H}-\mathrm{SiC}$ by the Born-von-Kármán method, and the corresponding eigenvectors, we used force constants obtained by interpolation and extrapolation of those calculated ab initio for 3 C (zinc blende) and 2 H (wurtzite) SiC using linear response and the local density approximation of density functional perturbation theory. The details of the ab initio calculations for 3 C - and $2 \mathrm{H}-\mathrm{SiC}$ are given in $[6,12]$ whereas those involved in the determination of the force constants of $4 \mathrm{H}-\mathrm{SiC}$ will be given in [11]. We obtain a good agreement with a phonon dispersion calculated in [13] using the bond-charge model.

## 3 Results and Discussion

Figure 1 shows the theoretical phonon dispersion curves of $4 \mathrm{H}-\mathrm{SiC}$, together with experimental points obtained by IXS $(\bullet)$, along the high symmetry directions $\Gamma-\mathrm{K}-\mathrm{M}\left(q_{x}\right), \Gamma-\mathrm{M}\left(q_{y}\right)$, measured on the (12 10 ) surface, and $\Gamma-\mathrm{A}\left(q_{z}\right)$, measured on the (0001) surface; for a sketch of the BZ of the hexagonal SiC polytypes see $[5,14]$. The latter corresponds to a double backfolding of the $\Gamma$-L direction of the $3 \mathrm{C}-\mathrm{SiC}$. The concept of folding along $\Gamma$-A has been used in the


Fig. 1. Phonon dispersion relations of $4 \mathrm{H}-\mathrm{SiC}$. IXS data $(\bullet)$ are displayed together with calculated phonon dispersion (solid lines) and Raman data $(\diamond)$ from Ref. [13]. The $\mathbf{q}$-axes are given in reciprocal lattice units (r.l.u.). For a comparison of the M-point frequencies with photoluminescence data see Table 1.
past to predict phonon frequencies $[2,3]$ and eigenvectors [4] of 3C-SiC along $\Gamma$-L which agree
with IXS data [6]. The Raman data collected in [13] are also displayed in Fig. 1 by open diamonds. Notice that along the $\Gamma$-A direction the calculated frequencies are slightly lower than the measured ones. This underestimate is also reflected in Table 1, where we list the theoretical and IXS data together with those obtained from the indirect gap photoluminescence (LTPL) at $2 \mathrm{~K}[15,14]$. The LTPL spectra exhibit phonon peaks corresponding to indirect transitions between the maximum of the valence band at $\Gamma$, and the minima of the conduction band, at the M-point, aided by phonon emission.

We have assigned in Table 1 the symmetry of the different modes at the M-point on the basis of the calculated eigenvectors. The $\mathrm{M}_{1}$ and $\mathrm{M}_{3}$ symmetry phonons are emitted in LTPL processes polarized along the $c$-axis whereas $\mathrm{M}_{2}$ and $\mathrm{M}_{4}$ phonons are emitted for the perpendicular polarization [14]. The agreement between symmetry assignments and the observed polarizations is excellent.

| Symmetry | ${ }^{\text {a }}$ IXS | ${ }^{\text {b }}$ [15] |  | theory | Symmetry | ${ }^{\text {a }}$ IXS | ${ }^{\mathrm{b}}[15]^{\mathrm{b}}[14]$ | theory |
| :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: | :---: |
| $\mathrm{M}_{2}$ | 33.1(2) | 33.0 | $33.2 \perp$ | 32.5 | $\mathrm{M}_{2}$ | 94.0(3) | $94.5 \perp$ | 93.0 |
| $\mathrm{M}_{1}$ |  | 36.5 | 36.6 | 34.9 | $\mathrm{M}_{3}$ | 94.9(3) | 96.194 .7 | 93.2 |
| $\mathrm{M}_{3}$ |  | 40.6 | 40.6 \|| | 38.7 | $\mathrm{M}_{4}$ |  | $97.396 .0 \perp$ | 94.0 |
| $\mathrm{M}_{4}$ |  | 41.0 | $41.1 \perp$ | 39.4 | $\mathrm{M}_{3}$ |  | 97.8 96.5 \|| | 94.4 |
| $\mathrm{M}_{3}$ | 41.5(2) | 41.7 | 41.9 | 39.4 | $\mathrm{M}_{2}$ |  | $99.1-\perp$ | 94.4 |
| $\mathrm{M}_{2}$ | 45.5(2) | 46.1 | $46.3 \perp$ | 45.1 | $\mathrm{M}_{1}$ |  | 101.6 - \|| | 94.9 |
| $\mathrm{M}_{4}$ |  | 50.8 | $50.9 \perp$ | 50.2 | $\mathrm{M}_{1}$ |  | 105.6103 .5 \|| | 96.9 |
| $\mathrm{M}_{1}$ |  | 52.7 | 52.7 \|| | 51.3 | $\mathrm{M}_{4}$ |  | $100.398 .9 \perp$ | 97.5 |
| $\mathrm{M}_{4}$ |  | 67.8 | $68.1 \perp$ | 67.8 | $\mathrm{M}_{4}$ |  | 106.3103 .9 ค | 102.5 |
| $\mathrm{M}_{1}$ | 69.5(2) | 68.7 | 68.8 | 68.5 | $\mathrm{M}_{1}$ | 105.5(5) | 108.2106 .3 \|| | 103.5 |
| $\mathrm{M}_{4}$ |  | 76.6 | $76.3 \perp$ | 74.7 | $\mathrm{M}_{1}$ |  | 108.7109 .5 \|| | 104.1 |
| $\mathrm{M}_{1}$ |  | 78.5 | 78.0 | 75.8 | $\mathrm{M}_{4}$ |  | $106.5 \perp$ | 106.2 |

Note. ${ }^{2} \overline{\overline{\text { Data at }} 300 \mathrm{~K}}$.
Note. ${ }^{\mathrm{b}}$ LTPL data at 2 K .
Table 1. Frequencies of the phonons of $4 \mathrm{H}-\mathrm{SiC}$ in $\mathrm{meV}\left(1 \mathrm{meV} \approx 8.07 \mathrm{~cm}^{-1}\right)$ at the M -point of the Brillouin zone determined by inelastic X-ray scattering (IXS), compared with experimental data from low temperature photoluminescence (LTPL) and our lattice dynamical calculations. After the LTPL frequencies we have listed the corresponding polarizations (|| or $\perp$ with respect to the $c$-axis) of the emitted light as given in [14].

The frequencies calculated for the folded LO phonons at $\Gamma$ are 118.4, 113.6, 112.2 and 104.1 meV . The highest and lowest of these frequencies have been determined by Raman spectroscopy to be 119.5 and 103.9 meV . Note that the calculated discontinuity in the highest LO frequency near $\Gamma$, depending on whether the phonons are polarized along $c$ or perpendicular to $c$, is 0.03 meV , too small to be seen in Fig. 1. The same can be said for the TO phonon discontinuities. The folded TO phonon frequencies at $\Gamma$ are $97.1\left(\Gamma_{5}\right.$ symmetry), $96.0\left(\Gamma_{6}\right), 95.0\left(\Gamma_{6}\right)$, and 93.6 meV $\left(\Gamma_{5}\right)$ as compared with the Raman results: $98.8,97.2$, and 96.2 meV [3]. Raman data in [16] show a splitting of 2.5 meV between the uppermost, $\mathrm{TO}\left(\Gamma_{5}\right)$ mode and a $\Gamma_{6}$ mode which is compatible with our calculated $\mathrm{TO}\left(\Gamma_{5}\right)$ and $\mathrm{TO}\left(\Gamma_{6}\right)$ frequencies. Splittings of $\approx 1 \mathrm{meV}$, also compatible with our calculations, have been recently observed for the TO modes on nanosized powders of SiC with a nonstoichiometric excess of carbon [17].

Inelastic neutron scattering measurements on a large single crystal of 4H-SiC are now being performed at the Laue Langevin Institute (Grenoble) to determine with high resolution the acoustic and longitudinal optic phonon branches along the $\Gamma-\mathrm{M}$ and $\Gamma-\mathrm{K}-\mathrm{M}$ directions in the BZ [18].

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## References

[1] Silicon Carbide. A Review of Fundamental Questions and Applications to Current Device Technology, Eds. W. J. Choyke, H. Matsunami and G. Pensl (Akademie Verlag, Berlin, 1998), Vol. I and II.
[2] D. W. Feldman et al., Phys. Rev. 173, 787 (1968).
[3] S. Nakashima and H. Harima, Phys. Stat. Sol. A 162, 39 (1997); S. Nakashima, A. Wada, and Z. Inoue, J. Phys. Soc. Jpn. 56, 3375 (1987).
[4] F. Widulle et al., Phys. Rev. Lett. 82, 3089 (1999).
[5] B. Dorner et al., Eur. Phys. J. B 5, 839 (1998).
[6] J. Serrano et al., Appl. Phys. Lett. 80, 4360 (2002).
[7] C. Masciovecchio et al., Nucl. Instrum. Methods Phys. Res., Sect. B 111, 181 (1996); 117, 339 (1996).
[8] M. Schwoerer-Böhning, A. T. Macrander, and D. A. Arms, Phys. Rev. Lett. 80, 5572 (1998).
[9] M. Schwoerer-Böhning et al., Phys. Stat. Sol. B 215, 177 (1999).
[10] T. Ruf et al., Phys. Rev. Lett. 86, 906 (2001).
[11] B. Stojetz, unpublished.
[12] K. Karch et al., Phys. Rev. B 50, 17054 (1994).
[13] M. Hofmann et al., Phys. Rev. B 50, 13401 (1994).
[14] I. G. Ivanov et al., Phys. Rev. B 58, 13634 (1998).
[15] W. J. Choyke, Inst. Phys. Conf. Ser. 142, 257 (1996).
[16] C. C. Tin et al., J. Cryst. Growth 188, 509 (1996).
[17] O. O. Mikhaylyk and D. N. Batchelder, private communication.
[18] D. Strauch et al., to be published.

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