



Article

On a Composite Obtained by Thermolysis of Cu-Doped Glycine

Pedro Chamorro-Posada ¹, Roberto C. Dante ², Jesús Martín-Gil ³, Denisse G. Dante ², Alma Cioci ^{2,4}, José Vázquez-Cabo ⁵, Óscar Rubiños-López ⁵, Irene Mediavilla-Martínez ⁶ and Pablo Martín-Ramos ^{3,*}

¹ Departamento de Teoría de la Señal y Comunicaciones e IT, Universidad de Valladolid, ETSI Telecomunicación, Paseo Belén 15, 47011 Valladolid, Spain; pedcha@tel.uva.es

² R&D Department, 2Dto3D S.r.l.s., Via Revalanca 5, San Firmino, 12036 Revello, Italy; rcdante@2dto3dmaterials.com (R.C.D.); denisse.dante@2dto3dmaterials.com (D.G.D.); alma.cioci@unito.it (A.C.)

³ Advanced Materials Laboratory, ETSIIAA, Universidad de Valladolid, Avda. Madrid 44, 34004 Palencia, Spain; jesus.martin.gil@uva.es

⁴ Dipartimento di Chimica, Università di Torino, Via Pietro Giuria, 7, 10125 Torino, Italy

⁵ Departamento de Teoría de la Señal y Comunicaciones, Universidad de Vigo, ETSE Telecomunicación, Lagoas Marcosende, 36310 Vigo, Spain; jvcabo@com.uvigo.es (J.V.-C.); oscar@com.uvigo.es (Ó.R.-L.)

⁶ Escuela de Ingenierías Industriales, Universidad de Valladolid, Paseo Prado de la Magdalena 3-5, 47011 Valladolid, Spain; irene.mediavilla@uva.es

* Correspondence: pmr@uva.es

Abstract: Metal-doped carbonaceous materials have garnered significant attention in recent years due to their versatile applications in various fields, including catalysis, energy storage, environmental remediation, electronics, and sensors, as well as reinforcement. This study investigates the synthesis and characterization of a composite material featuring a carbonaceous matrix doped with copper, focusing on the thermolysis of glycine as a precursor. The synthesis methodology involved utilizing glycine and copper acetate monohydrate in varying ratios, with the mixture subjected to heating in ceramic crucibles at temperatures ranging from 450 to 550 °C, with pyrolysis yields over the 5 to 39% interval. The pristine and Cu-doped samples obtained at 500 °C underwent characterization using a diverse array of techniques, including scanning and transmission electron microscopies, multi-elemental analysis by energy dispersive X-ray spectroscopy, CHNS elemental analysis, X-ray photoelectron spectroscopy, X-ray powder diffraction, infrared and Raman spectroscopies, ultraviolet-visible spectroscopy, and terahertz time-domain spectroscopy, along with conductivity measurements. Under optimized conditions, copper (at 6.5%) was present primarily in the free metallic form, accompanied by traces of tenorite (CuO) and cuprite (Cu₂O). The carbonaceous matrix exhibited a 6:1 ratio of graphitic carbon to a carbon-nitrogen compound with the formula C₂H₂N₂O₂, such as isomers of diazetidinedione, according to multi-elemental analysis results. Conductivity measurements disclosed a significant increase in conductivity compared to the product of glycine thermolysis, showcasing the enhanced electrical properties of the new composite. Additionally, terahertz measurements showed the potential of the material as a broadband absorber for the fabrication of terahertz devices and provided compelling evidence of a significant improvement in radiation absorption upon copper doping. In conclusion, this research sheds light on the promising properties of copper-doped carbonaceous composites obtained by glycine pyrolysis, offering insights into their potential applications in emerging technological domains.



Citation: Chamorro-Posada, P.; Dante, R.C.; Martín-Gil, J.; Dante, D.G.; Cioci, A.; Vázquez-Cabo, J.; Rubiños-López, Ó.; Mediavilla-Martínez, I.; Martín-Ramos, P. On a Composite Obtained by Thermolysis of Cu-Doped Glycine. *C* **2024**, *10*, 49. <https://doi.org/10.3390/c10020049>

Academic Editors: Shayan Seyedin, Manuel Fernando Ribeiro Pereira and Jandro L. Abot

Received: 29 February 2024

Revised: 20 May 2024

Accepted: 24 May 2024

Published: 26 May 2024



Copyright: © 2024 by the authors. Licensee MDPI, Basel, Switzerland. This article is an open access article distributed under the terms and conditions of the Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>).

Keywords: copper acetate; conductivity; diazetidinedione; thermolysate; THz-TDS; TEM; FESEM; XPS

1. Introduction

Research on composites featuring a carbonaceous matrix and copper as a reinforcing element has garnered significant attention. Several reviews on copper/graphite composites [1], copper/graphene composites [2], and their corresponding preparation

methods [3] have been published. Scientific journal articles within this field extensively cover composite materials designed for applications in the electrical, thermal, structural, mechanical, and renewable energy domains. These applications encompass sliding electrical contacts, conductive wires and cables for power transmission and electronics, heat sinks and spreaders, thermal management systems in aerospace and aviation, components of brake disks, pantograph strips for high-speed railways, lightweight and high-strength aerospace and automotive industry components, sports equipment, conductive components of solar cells, and electrodes and conductive components of energy storage systems. Additionally, these materials find application in wall surface shields for future fusion devices [4–12].

Similarly, the integration of copper components into carbon-nitride (CN_x) architectures offers exciting prospects for advancing sustainable technologies centered upon renewable feedstock utilization and green chemistry principles [13].

Using amino acids as starting materials for the preparation of CN_x or carbonaceous materials presents several compelling rationales grounded in their characteristic properties and practical merits [14–16]. These include their functional diversity (ranging from carboxyl ($-COOH$), primary amine ($-NH_2$), secondary amine ($-NHR$), aromatic rings, sulfur moieties, etc.), availability from many natural sources without posing severe ecological threats (thus aligning with 'green' manufacturing practices), and facile accessibility (with consistent availability and affordability and simple purification methods facilitating rapid scale-up efforts needed for commercial ventures).

The thermal decomposition of amino acids, including glycine [17–19], has been studied to understand their thermal stability and the products formed during decomposition. These studies have shown that amino acids can decompose into various products, including water, cyclic compounds, and residues containing peptide bonds, which are stable at temperatures above 180 °C [20].

In particular, glycine, one of the simplest amino acids found in nature, has garnered significant attention in materials science due to its unique properties and versatility as a precursor for various materials, including nitrogen-doped ordered mesoporous carbons [16,21]. This interest is partly due to glycine's simple structure, high solubility in water, and its role as a building block for proteins, making it an ideal candidate for synthesizing complex materials through thermal treatment or polymerization processes. Its simplicity (composed merely of a backbone connected to a single hydrogen atom) facilitates easier manipulation and predictable outcomes during synthesis procedures, allowing for the formation of linear peptides through thermal polymerization, even under conditions that are not typically favorable for such reactions [22]. Moreover, during heat treatment, glycine displays distinctive characteristics, such as being able to participate in decarbonylation and cyclodehydration reactions, ultimately contributing to the generation of heterocycles relevant to CN_x materials. Additionally, glycine behaves as a bidentate ligand for many metal ions, allowing easy integration of transition elements such as, copper into emerging networks. Last but not least, glycine brings forth flexibility stemming from its capability to exist either neutrally charged or ionized (zwitterionic state); hence, it could potentially interact effectively with various target surfaces.

Considering these attributes collectively, employing glycine as a foundation for constructing novel CN_x materials holds promise for achieving innovative designs and tunable physicochemical properties beneficial for future advancements in optoelectronics, sensors, and environmentally friendly devices. Consequently, in this paper, we investigate a novel composite produced utilizing glycine and copper (II) acetate. The aim of the study is to provide a characterization of the pyrolysis products through a diverse range of techniques, along with electrical conductivity and radiation absorption in the terahertz band measurements. This broader scope aims to explore additional potential applications for this new material.

The importance of these explorations becomes evident when one considers the opening of the terahertz spectral region [23] to diverse applications such as material characterization,

identification, telecommunications, medical diagnosis imaging, security screening, or industrial testing. This requires the development of materials with specific properties tailored for terahertz radiation systems. Specifically, terahertz absorbers play a pivotal role in creating electromagnetic radiation shields, thermal sources, detectors, filters, attenuators, and radar absorption materials [24].

2. Materials and Methods

2.1. Synthesis

In the synthesis method employed here, mixtures of glycine (as the primary component) and copper (II) acetate (as a minor component) were blended in varying proportions, ranging from 0.5% to 10% in copper (II) acetate (equivalent to Cu percentages between 0.16% and 3.17% in the initial mixture). Specifically, the process was initiated by doping glycine [molecular weight (MW): 75.07 g/mol, linear formula: $\text{NH}_2\text{NCH}_2\text{COOH}$] with different quantities of copper diacetate monohydrate [MW: 199.65 g/mol, formula: $\text{Cu}(\text{CH}_3\text{COO})_2 \cdot \text{H}_2\text{O}$]. The mixture was homogenized in an agate mortar until the glycine exhibited a uniform blue color indicative of even dispersion of the copper complex. Subsequently, the mixture was transferred to an alumina crucible with a lid and heated in a static air atmosphere within a Eurotherm (Worthing, UK) high-temperature furnace capable of reaching up to 1100 °C.

The furnace temperature was programmed to rise from 50 °C to 200 °C, and upon reaching this temperature, it continued to increase at a rate of 10 °C per minute until one of the three selected maximum temperatures (450, 500, or 550 °C) was achieved. The temperature was then maintained for 50 minutes in an isothermal manner. Following this, the furnace was cooled down to 50 °C at a rate of 10 °C per minute. The thermolysis process led to the loss of water and ammonia, accompanied by an increase in the copper percentage, which was determined experimentally.

2.2. Characterization

Transmission electron microscopy (TEM) images were obtained using a JEM-1010 (JEOL, Tokyo, Japan) operating at 100 KeV. Scanning transmission electron microscopy (STEM) images were also obtained using a JEM-2010F (JEOL) operating at 200 KeV in both BF (bright-field) and HAADF (high-angle annular dark-field) modes, for small angles (<10 mrad) and great angles (>50 mrad). HAADF gives a contrast in Z so that the brightest parts allow the presence of metallic elements (such as Cu, a component with high Z) to be discriminated against.

For both field emission scanning electron microscopy (FESEM) images and energy-dispersive X-ray spectroscopy (EDS) analyses, a JSM-6700F microscope (JEOL), incorporating a cold cathode field emission (FE) gun, and ultra-high vacuum have been used.

Elemental analysis (CHNS) characterization was conducted using a Carlo-Erba EA-1108 microanalyzer (Thermo Fisher Scientific, Waltham, MA, USA), with sulfamethazine and sulfanilamide standards intercalated between samples.

The chemical composition and chemical environment of the samples were examined by X-ray photoelectron spectroscopy (XPS) surface measurements acquiring C1s, O1s, N1, Cu2p, CuLMN, and survey spectra. The analysis was conducted employing a Thermo Fisher Scientific NEXSA instrument featuring aluminum $K\alpha$ monochromatized radiation at a 1486.6 eV X-ray source. Due to the non-conductive nature of the samples, an electron flood gun was employed to minimize surface charging. Neutralization of the surface charge was achieved using both a low-energy flood gun (with electrons in the range of 0 to 14 eV) and a low-energy Ar ion gun. XPS measurements utilized monochromatic Al- $K\alpha$ radiation ($h\nu = 1486.6$ eV), with photoelectrons collected from a take-off angle of 90° relative to the sample surface. The measurements were conducted in constant analyzer energy mode (CAE), employing a 100 eV pass energy for survey spectra and a 20 eV pass energy for high-resolution spectra. The survey scan number was 5, and for high-resolution spectra, 20 scans were conducted. Charge referencing was accomplished by setting the lower binding

energy C1s photo peak at 284.80 eV, corresponding to the C1s hydrocarbon peak [25]. Surface elemental composition was determined using standard Scofield photoemission cross-sections. Regarding additional experimental details, the samples were prepared by pressing the sample powder onto double-sided conductive tape, which was then attached to the instrument's metal holder. The operator selected the largest X-ray spot at 400 μm . The residual vacuum in the analysis chamber was maintained at around 3×10^{-9} mbar. Data analysis and quantification were executed using Avantage software version 6.5.1 from the manufacturer (Thermo Fisher Scientific). Intensity calculations were based on the area under peaks, obtained after background subtraction. The employed background was the 'smart background', derived from the 'Shirley background', with an additional constraint ensuring that the background did not surpass the actual data intensity at any point in the region. The experimental curves were fitted using a combination of Lorentzian-Gaussian lines in variable proportions. The library used was SCOFIELD, employing the TPP-2M method for attenuation length. Sensitivity factors utilized for calculating elemental atomic percentages are detailed in Table 1.

Table 1. Sensitivity factors used to calculate elemental atomic percentages in XPS measurements.

Element/Transition Core Lines	SCOFIELD SF
C1s	1
O1s	2.930
N1s	1.8
Cu2p3	16.730

X-ray powder diffraction (XRPD) patterns were obtained using a Bruker (Billerica, MA, USA) D8 DISCOVER A25 diffractometer in reflection mode (Bragg-Brentano geometry) with a 3 kW generator, 2.2 kW type FFF Cu-ceramic tube, and LynxEye Detector operating at 40 kV and 30 mA. $\text{CuK}\alpha$ ($\lambda = 1.54 \text{ \AA}$) radiation and crystalline silicon powder (NIST SRM 640F) served as a standard under the operational conditions of 30 mA, 40 kV, and $2\theta = 5\text{--}70^\circ$. Rietveld analysis was conducted using FullProf v.5.10 [26].

Infrared spectra were collected employing a Nicolet iS50 Fourier-transform infrared (FTIR) spectrometer (Thermo Fisher Scientific), equipped with a diamond-attenuated total reflection (ATR) module. The spectral measurement conditions included room temperature, a range of $400\text{--}4000 \text{ cm}^{-1}$, 0.5 cm^{-1} spectral resolution, and the addition of 64 scans. Ten spectra were collected from each subsample in powder form, with no significant differences observed among them.

Raman spectra were acquired using a LabRAM SOLEIL spectrometer (Horiba, Kyoto, Japan) equipped with a Symphony CCD detector. The exciting beam was generated by a solid-state laser ($\lambda = 532 \text{ nm}$), maintaining a nominal laser power of 1 mW on the sample and an approximate irradiance of $100 \text{ kW}\cdot\text{cm}^{-2}$. Acquisition parameters included a 10 s acquisition time, 6 accumulations, and a spectral split aperture at $200 \mu\text{m}$, ensuring a spectral resolution better than 1 cm^{-1} . Calibration was performed using the $\nu(\text{Si-Si})$ Raman mode at 520.7 cm^{-1} of crystalline silicon from a bare (001) wafer.

Ultraviolet-visible spectroscopy (UV-Vis) diffuse reflectance spectra in powder form were recorded using an Agilent (Santa Clara, CA, USA) Cary 100 UV-Vis spectrophotometer equipped with an integrating sphere. The intensity unit of spectra reported is $F(R)$, i.e., the Schuster-Kubelka-Munk formula, which provides values related to absorbance as obtained in transmission. The bandgaps were determined by the Tauc plot using $F(R)$ [27].

Conductivity measurements. Samples obtained from the thermal treatment of pristine glycine and Cu-doped glycine (Gly-Cu 3.17% 500 $^\circ\text{C}$ sample) were thoroughly milled in an agate mortar. For the AC electrical characterization of each sample, 50 mg of the fine powder was placed in a cylindrical PLA receptacle of 0.5 mm height and 10 mm diameter with a 24 mm diameter round electrode in an FR4 PCB at its base and gently pressed with a spatula. The sample holder was then terminated with another identical electrode. Both electrodes had external electrical connections. The AC impedance of the prepared samples

was measured in the range between 100 Hz and 6 kHz using an HM8118 HAMEG (Hameg Instruments, Barcelona, Spain) programmable LCR bridge equipped with Kelvin probes.

Terahertz time-domain spectroscopy (THz-TDS) measurements. A Menlo Tera K15 (Menlo Systems GmbH, Martinsried, Germany) spectrometer with a 1560 nm fiber laser generating ultrashort pulses (90 fs, 100 MHz repetition rate) was used for the analysis. The nitrogen-purged environment minimized water vapor interference. Ten measurements per sample/reference improved the signal-to-noise ratio. Material parameters were retrieved from time-domain photocurrent traces after applying established processing techniques [28] to account for sample thickness and reflections. Further experimental details are available in [28,29].

2.3. Computations

Ground state geometries were calculated using the PM6 Hamiltonian [30] implemented in MOPAC 2016. The electronic spectra calculations were based on these geometries using the intermediate neglect of differential overlap/spectroscopic (INDO/S) configuration interaction with the singles (CIS) method [31,32] implemented in the ORCA software package [33].

3. Results

3.1. Pyrolysis Yields

The formation of the composite varied in yields depending on the Cu-doping level of glycine (Table 2). Since a noticeable yield decrease was observed upon increasing the temperature to 550 °C, the characterization primarily focused on the thermolysis products formed at 500 °C with the highest Cu-doping (Figure 1). However, the thermolysis products obtained at the three temperatures (and for the lowest Cu-doping) were studied using some techniques, namely FTIR, Raman, and UV-Vis spectroscopies, in a preliminary screening to assess temperature-related changes in their properties.

Table 2. Pyrolysis yields for pristine glycine and copper (II) acetate-doped glycine at different temperatures in static air conditions.

Temperature (°C)	Glycine	Glycine + 0.5% Copper (II) Acetate	Glycine + 10% Copper (II) Acetate
450	34.2%	38.1%	38.8%
500	29.2%	30.7%	33.8%
550	4.8%	5.3%	7.1%

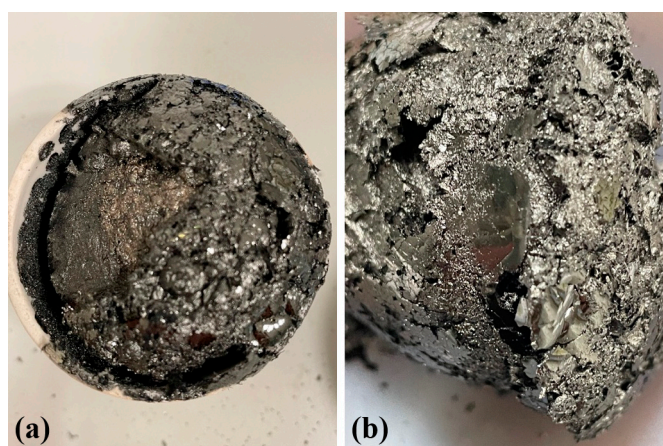


Figure 1. Thermolysis product of the glycine +10% copper acetate initial mixture, equivalent to 3.17% Cu, heated at 500 °C: (a) Gly-Cu 3.17% 500 °C sample inside the crucible, seen from above; (b) detail of Gly-Cu 3.17% 500 °C sample after unmolding.

3.2. Morphological Characterization by TEM and STEM

A TEM micrograph of the composite is displayed in Figure 2a. Figure 2b showcases a magnified STEM image (scale 10 nm), in which sheets stacking with an interplanar spacing of 2.59 Å may be observed, compatible with the presence of CuO nanoparticles dispersed in the carbonaceous matrix, as discussed below.

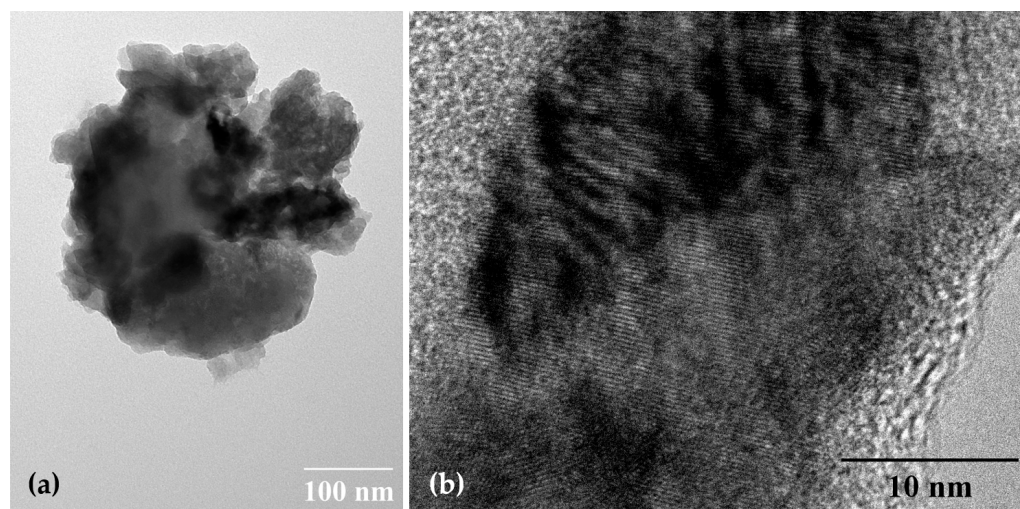


Figure 2. (a) TEM and (b) STEM images of the copper-doped glycine thermolysate (Gly-Cu 3.17% 500 °C sample).

3.3. Morphological Characterization by FESEM

The FESEM micrographs (Figure 3) facilitate the description of the new material as a composite with a homogeneous matrix, wherein the doping agent (copper) is dispersed and its particles are housed in vacuole-type formations. Particle size analysis in ImageJ [34] for non-aggregated particles resulted in an average diameter of 37 ± 9 nm.

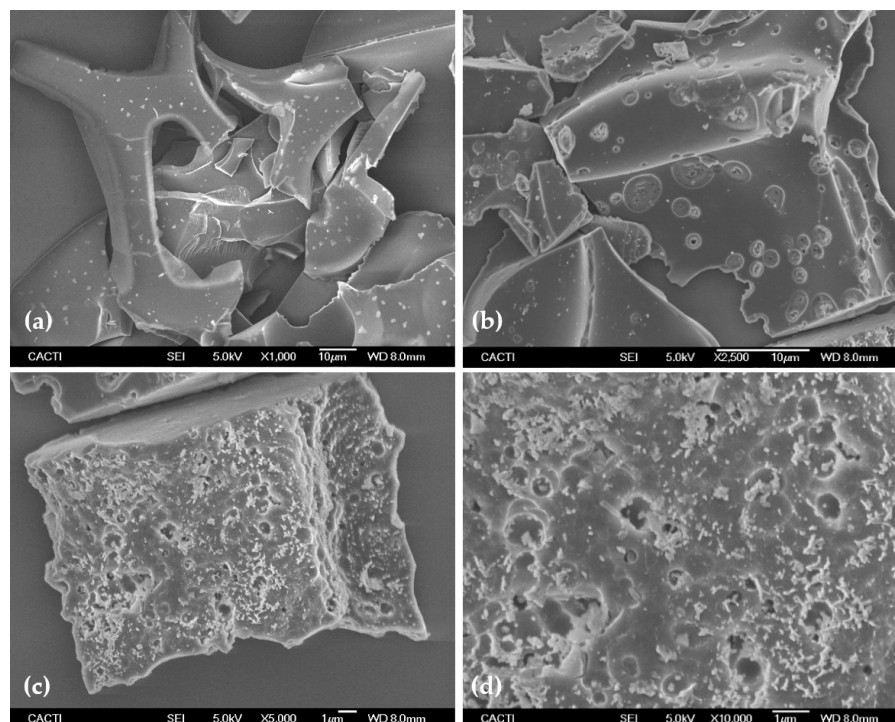


Figure 3. FESEM micrographs of (a) pristine glycine and (b–d) Cu-doped glycine thermolysates obtained at 500 °C (Gly 500 °C and Gly-Cu 3.17% 500 °C, respectively).

3.4. Multielemental Analysis by EDS

The multi-elemental composition of the pristine glycine and Cu-doped thermolysates (i.e., thermolysis products) obtained at 500 °C was determined at three randomly selected sites each (Table 3).

Table 3. Multielemental composition of the pristine glycine and Cu-doped glycine thermolysates obtained at 500 °C.

Element (wt%)	Glycine Thermolysate (Gly 500 °C)			Cu-doped Glycine Thermolysate (Gly-Cu 3.17% 500 °C)		
	Site 1	Site 2	Site 3	Site 1	Site 2	Site 3
C	61.1	62.1	69.7	63.0	58.4	58.3
N	29.6	27.7	25.5	22.0	23.2	26.5
O	7.5	8.3	4.6	7.5	9.1	8.7
Cu	-	-	-	7.3	8.8	3.5
Suggested formula	C ₁₂ H _x N ₅ O (64.3%C, 27.6%N, 6.8%O)			C ₅₀ CuH _x N ₁₇ O ₅ (59.9%C, 23.9%N, 8.4%O, 6.5%Cu)		

The empirical formula deduced from the EDS results (Table 3), identified as C₁₂H_xN₅O, suggests a hypothesis that the thermolysate of pristine glycine (C₂H₅NO₂) is a 6:1 mixture of carbon and a carbo-nitrogenated species with the formula C₆H₇N₅O:



This scenario is reminiscent of the purine bases 1-methylguanine or N-hydroxy-9-methyladenine (Figure 4).

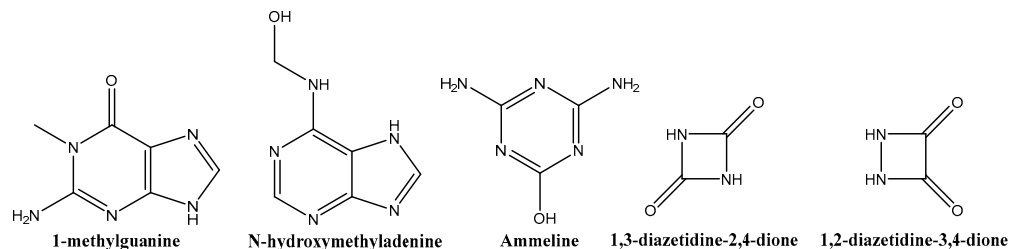
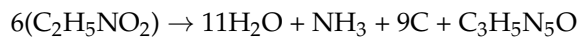


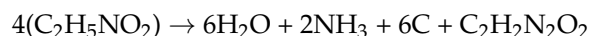
Figure 4. Chemical structures of glycine thermolysis products.

Alternatively, another probable scenario involves a 9:1 mixture of C and the species C₃H₅N₅O:



This case corresponds to ammeline or 2,4,6-triamino-1,3,5-triazin-1-ium-3-olate, also known as 2,4-diamino-6-hydroxy-1,3,5-triazine or 4,6-diamino-1,3,5-triazin-2(1H)-one. Ammeline is a dehydration product of melamine, obtainable through the heat treatment of glycine (Figure 4).

Nevertheless, according to the bibliography [20], the most likely possibility is the formation of 1,3-diazetidone-2,4-dione and 1,2-diazetidone-3,4-dione isomers (Figure 4), which involves a 6:1 mixture of C and the aforementioned chemical species with the formula C₂H₂N₂O₂:



Regarding the new copper/carbon composite prepared by thermolysis at 500 °C, the EDS results (Figure 5) guide us to consider that their empirical formula, C₅₀CuN₁₇O₅, would correspond to a mixture of copper, graphenic/graphitic carbon, and a diazetidinedione.

Copper, constituting approximately 6.5% in percentage, exists in the composite in a free metallic form, accompanied by traces of tenorite (CuO), as discussed in Section 3.6

below. The carbo-nitrogenated species $C_2H_2N_2O_2$ is postulated as a potential product of the thermal decomposition of either glycine ($C_2H_5NO_2$) (see above) or copper (II)glycinate ($C_4H_8CuN_2O_4$) formed through the reaction of glycine and copper acetate.

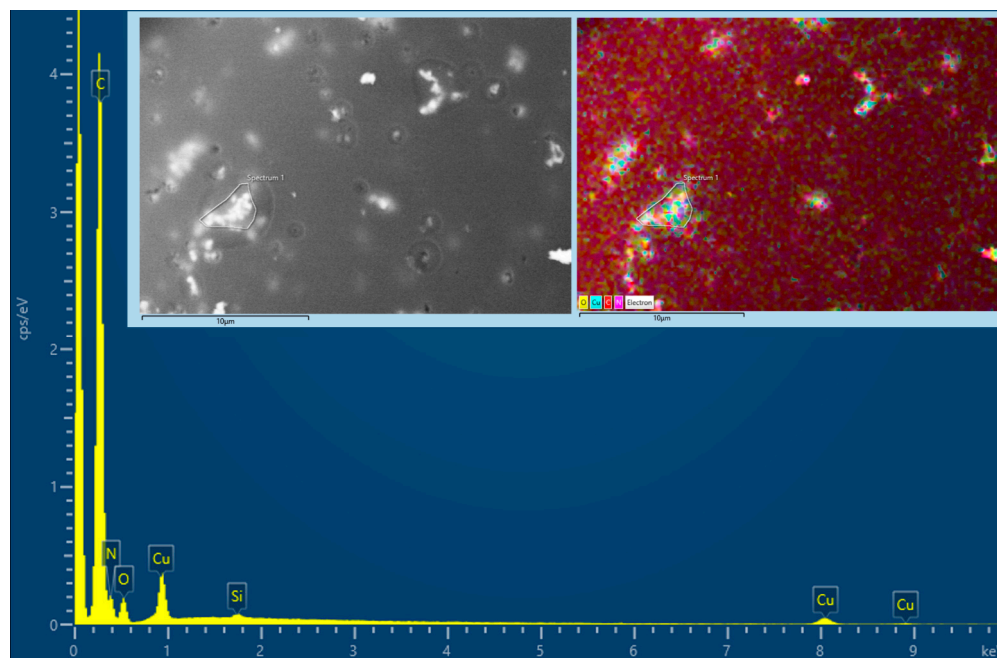
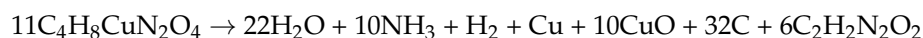


Figure 5. EDS spectrum and elemental mapping (inset) of the Cu-doped glycine thermolysate obtained at 500 °C (Gly-Cu 3.17% 500 °C sample).

3.5. CHNS Characterization

The CHNS characterization results for the thermolysis product obtained from pristine glycine at 500 °C exhibited the following composition (in wt%): $61.13 \pm 0.16\%$ C, $2.39 \pm 0.11\%$ H, $22.09 \pm 0.11\%$ N, $<0.30\%$ S, and 14.09% O (calculated by difference). In the case of the Gly-Cu 3.17% sample thermolyzed at 500 °C, the composition was $54.27 \pm 0.13\%$ C, $2.29 \pm 0.09\%$ H, $20.19 \pm 0.07\%$ N, $<30\%$ S, and approximately 16.45% O (determined by difference, assuming a 6.5% Cu content from the EDS analysis).

The respective C/H, C/N, and C/O ratios (after conversion of wt% results to at%) were 2.13, 3.23, and 5.78 for the pristine glycine thermolysis product at 500 °C, and 1.97, 3.14, and 4.40 for the Gly-Cu 3.17% sample thermolyzed at 500 °C. Considering the C/H ratio (or the H/C ratio, 0.47 and 0.51 for Gly 500 °C and Gly-Cu 3.17% 500 °C, respectively) as an indicator of the degree of aromatization, the differences between the organic matrices of the two samples would not be substantial. This conclusion is based on the premise that the hydrogen contents involved in the calculation are within the standard error of the content measurement ($\pm 0.11\%$).

3.6. XPS Characterization

Wide scan survey spectra (Figure 6a) were utilized for the identification and quantification of elements in the thermolysis products obtained at 500 °C (Gly 500 °C and Gly-Cu 3.17% 500 °C), while high-resolution narrow scans were employed to construct the chemical state assessment table (Table 4).

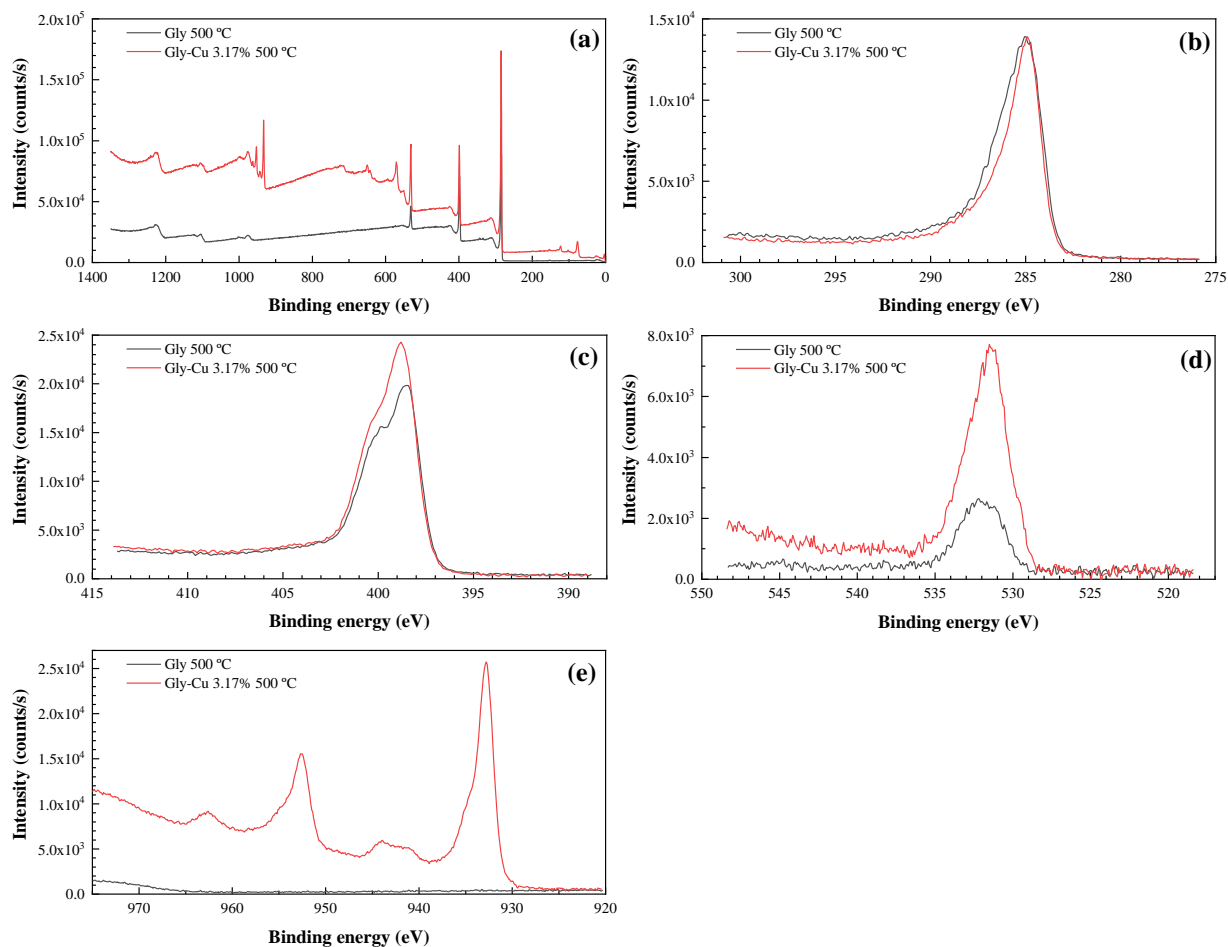


Figure 6. XPS spectra of the pristine glycine and Cu-doped glycine thermolysates obtained at 500 °C (Gly 500 °C and Gly-Cu 3.17% 500 °C samples, respectively): (a) survey spectra; (b) C1s spectra; (c) N1s spectra; (d) O1s spectra; (e) Cu2p spectra.

Table 4. Elemental composition, expressed in at%, along with C/N and C/O ratios, of the pristine glycine and Cu-doped glycine thermolysates obtained at 500 °C.

Sample	C	N	O	Cu	Si	Cl	C/N	C/O
Gly 500 °C	68.97	24.02	6.32	-	0.54	0.15	2.9	10.9
Gly-Cu 3.17% 500 °C	68.70	17.46	10.39	2.44	1.02	-	3.9	6.6

In the wide scan survey spectra, peaks corresponding to O1s, C1s, and N1s were identified at 532.5 eV, in the 285–287 eV range, and in the 399–403 eV range, respectively. The presence of copper in glycine is evident, and the atomic percentage of oxygen increases due to the combination of copper with oxygen, contributing more oxygen to the compound. The C/N ratio of pristine glycine treated at 500 °C (2.9) deviates significantly from the theoretical ratio of glycine (2.0), indicating an excess of carbon or a deficiency of nitrogen resulting from the thermal treatment.

Remarkably, the C/O ratio in the pristine glycine thermolysate at 500 °C is 11, far exceeding the theoretical value of 1 for glycine. This implies a substantial excess of carbon or a deficiency of oxygen. However, upon copper incorporation, the C/O ratio decreases to 6.6, suggesting effective oxygen incorporation in the presence of copper, as mentioned earlier.

These analytical findings align well with those obtained by EDS for the pristine glycine thermolysate, with C/O ratios of 12.55 and 10.91 for EDS (after conversion of wt% results in Table 3 to at%) and XPS, respectively. However, they differ more in the case of the Gly-Cu

3.17% sample, with a C/O ratio of 9.46 for EDS (after conversion of wt% results in Table 3 to at%) vs. 6.61 for XPS. The atomic percentages of Cu obtained for Gly-Cu 3.17% by both methodologies also differ (1.44 at% by EDS vs. 2.44 at% by XPS).

Concerning the high-resolution spectra, in the case of C1s (Figure 6b, Figure 7a, and Table 5), the two carbon atoms of glycine give rise to two photoemission signals from their different chemical environments. The C-NH_3^+ peak arises at 286.2 eV, and COO^- at 288.3 eV.

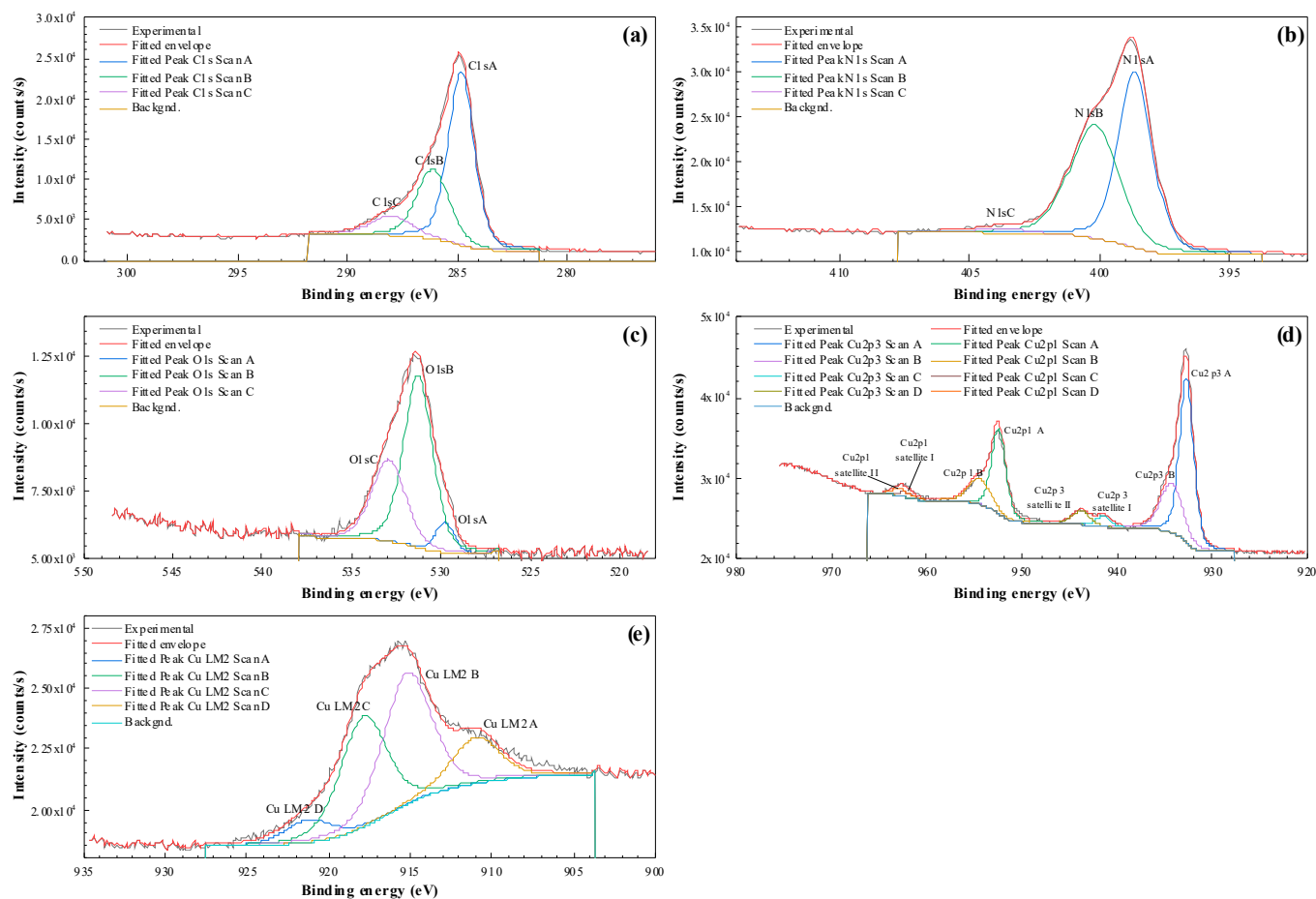


Figure 7. Fittings of the XPS spectra of the Gly-Cu 3.17% 500 °C sample: (a) C1s spectra; (b) N1s spectra; (c) O1s spectra; (d) Cu2p spectra; (e) Cu LM2 spectra.

Table 5. C1s binding energy (eV)/relative% for the pristine glycine and Cu-doped glycine thermolysates obtained at 500 °C.

Sample	C1s A C–C, CH_x (284.8 eV)	C1s B C-NH_3^+ , C–O–C, C–O–H (285.6–286.5 eV)	C1s COO^- (288–289 eV)
Gly 500 °C	284.8/45.54%	286.13/37.47%	288/9.37%
Gly-Cu 3.17% 500 °C	284.8/59.09%	286.1/30.21%	288/10.7%

Regarding N1s (Figure 6c, Figure 7b, and Table 6), glycine theoretically has a single nitrogen atom in its structure, resulting in a single photoemission signal at 401.4 eV for both polymorphic forms. However, our high-resolution signal fitting of the N1s reveals two majority components, one around 398.5 eV and the other around 400.2 eV. After copper introduction, the component at 402 eV shifts to 403.8 eV, corresponding to nitrite, although this component has a very low contribution to the overall nitrogen composition.

Table 6. N1s binding energy (eV)/relative% for the pristine glycine and Cu-doped glycine thermolysates obtained at 500 °C.

Sample	N1s A -NH ₂	N1s B amine -NH- (400.0 eV), -N-(C=O) (400.50 eV)	N1s C doped imine -NH ⁺ - (402.00 eV) nitrite (NO ₂) (403.9 eV)
Gly 500 °C	398.53/49.09%	400.17/46.26%	402/4.65%
Gly-Cu 3.17% 500 °C	398.64/51.3%	400.2/46.94%	403.8/1.76%

In relation to O1s (Figures 6d and 7c), the deconvolution peak at 529.76 eV can be ascribed to Cu–O interaction, the peak at 531.27 eV to hydroxyl groups (OH), and the one at 532.9 eV to C–O bonds. The oxygen involved in OH significantly increased due to the introduction of Cu.

As for Cu2p (Figure 6e, Figure 7d, and Table 7), comparing the spectra with those reported by Biesinger et al. [35], the shape of the Cu2p_{3/2} spectrum and the satellites suggests it is not Cu(0), and Cu(1+) hardly has any satellites. The spectrum is similar to that of Cu(OH)₂(2+), indicating a mixture of species with low-intensity satellites, possibly a combination of Cu(0) and a Cu(2+) Cu(OH)₂ hydroxide or CuO oxide (tenorite).

Table 7. Cu2p_{3/2} binding energy (eV)/relative% and CuLMM kinetic energy (in eV) for the Cu-doped glycine thermolysate obtained at 500 °C.

Cu2p _{3/2} A	Cu2p _{3/2} B	Satellite I	Satellite II	Cu LMM2 A	Cu LMM2 B	Cu LMM2 C	Cu LMM2 D
932.71/62.37%	934.28/27.5%	941.45/4.19%	943.8/5.94%	921.12	917.75	915.09	910.88

Auger parameter calculation was employed for determining the chemical state, as detailed in Table 8. The Cu LMM peak shape, along with the Auger parameter and Cu2p_{3/2} peak position and shape, also proved valuable in discerning the different chemical states of copper. The presence of Cu (I) seems confirmed and consistent with the discovery by XPD of traces of Cu₂O (cuprite).

Table 8. Cu LMM (Auger peak) analysis for the Cu-doped glycine thermolysate obtained at 500 °C.

	Cu2p _{3/2} (I)	Cu2p _{3/2} (II)	Cu L ₂₋₃ M ₄₅ M ₄₅ Average Peak Position Peaks 2–3	Auger Parameter (I)	Auger Parameter (II)
From high-resolution spectrum	932.71	934.28	916.42	1849.13 Cu ₂ O ⁻ Cu ¹⁺	1850.7 Cu(OH) ₂ ⁻ Cu ²⁺
From survey	932.9		916.42	1849.34 Cu ₂ O Cu ¹⁺	

3.7. X-ray Powder Diffraction Characterization

In the powder diffractogram of the Gly-Cu0.16% sample thermolysate (Figure 8), only the presence of Cu metal is observed, indicated by reflections (111) and (200) at 43.3° and 50.4°, respectively, associated with a face-centered cubic lattice with a unit cell length of 361 pm [36]. Crystallite sizes (coherent scattering region sizes), estimated using the Scherrer equation, would be in the 34.76–38.72 nm range. The peak at 25.7° bears resemblance to the (002) reflection of graphite, albeit typically found at 26.7°, and would be associated with an interplanar distance of 3.46 Å.

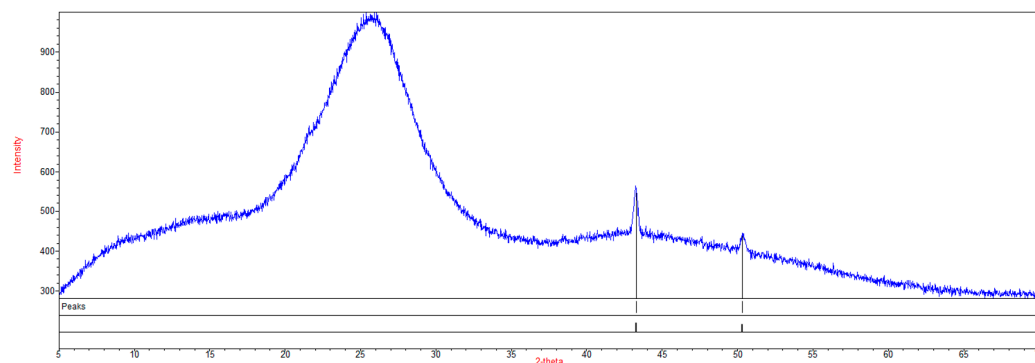


Figure 8. X-ray powder diffraction pattern of the Gly-Cu 0.16% 500 °C sample thermolysate using QualX [37].

In the Gly-Cu 3.17% sample thermolysate (Figure 9), alongside the prevailing presence of Cu metal, traces of CuO (tenorite) and Cu₂O (cuprite) are evident. Rietveld analysis (Chi2 = 5.1082) yielded weight fractions of 84.88, 7.22, and 7.90% for Cu metal, CuO, and Cu₂O, respectively. Average crystallite sizes of 35.10, 18.82, and 28.42 nm were estimated for Cu metal, CuO, and Cu₂O, respectively, using the Scherrer equation. These values are comparable to (in the case of Cu metal) or slightly lower (in the case of copper oxides) than the particle sizes obtained from SEM micrographs (in the 29 to 55 nm range, with an average value of 37 ± 9 nm). The peak associated with graphite/graphene shifts to 26.1° , corresponding to an interplanar distance of 3.41 Å. This value does not match the interplanar distance calculated from STEM images. The most likely assignment for the latter interplanar spacing would be the (11 $\bar{1}$) and (002) plane reflections of CuO at 2θ values of 35.55° and 35.57° , respectively (JCPDS card no. 00–041-0254).

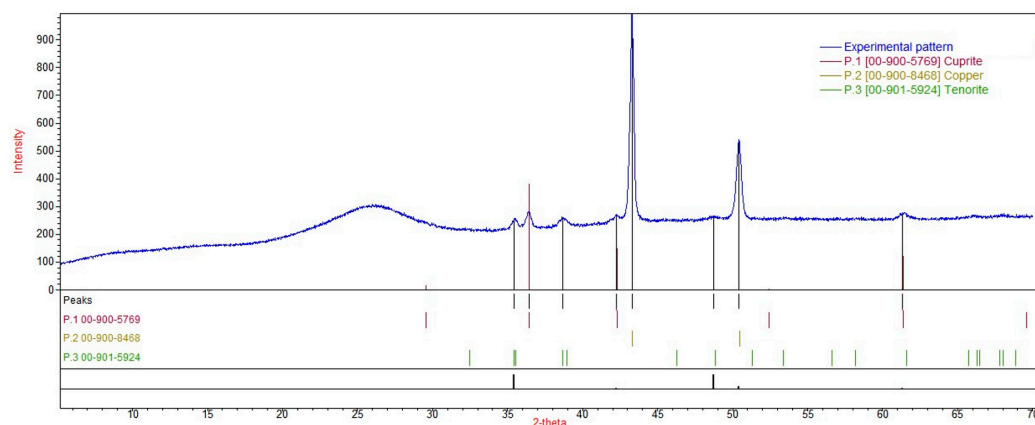


Figure 9. X-ray powder diffraction pattern of the Gly-Cu 3.17% 500 °C sample using QualX [37].

3.8. Vibrational Characterization

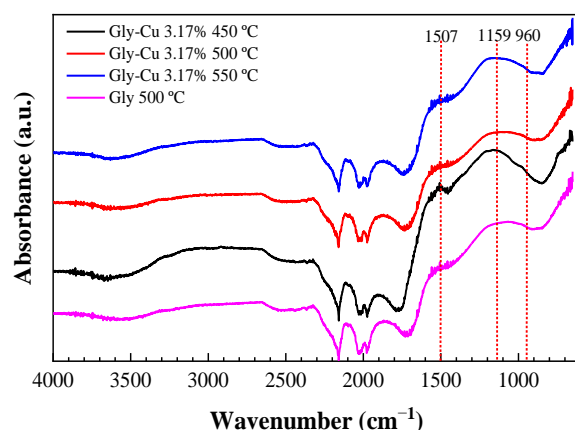
3.8.1. Infrared Spectra

The FTIR spectra of the thermolysis products from pristine glycine (Table 9) reveal three prominent bands at $3071\text{--}3074\text{ cm}^{-1}$, 1583 cm^{-1} , and $1206\text{--}1216\text{ cm}^{-1}$. The band in the range of $1206\text{--}1216\text{ cm}^{-1}$ is attributed to C–N stretching vibration, consistent with the presence of single carbo-nitrogenated species such as diazetine-diones. The band around 1580 cm^{-1} , a characteristic mode in graphene and graphite, signifies the in-plane vibration of sp^2 -bonded carbon atoms and is sensitive to the number of layers in graphite.

Table 9. Main infrared bands and assignments of pristine glycine and Cu-doped glycine thermolysates.

Gly 450 °C	Gly 500 °C	Gly 550 °C	Gly–Cu 0.16% 500 °C	Gly–Cu 3.17% 500 °C	Assignment
3074	3086	3072	3044	-	C–H stretching; amide B mode;
-	-	-	2880	2659	O–H stretching
2223	-	2161	-	-	CH ₂ stretching
1583	-	-	-	-	CH ₃ asymmetrical stretch; C=C stretch;
-	(1564)	-	1563	1564	C=O stretching
-	-	-	1507	1493	graphene/graphite; amide II stretching
1206	1212	1216	1204	1216	asymmetrical stretching of C=O in Cu carboxylates
-	-	-	1159	1155	ν (C=C); asymmetrical flexion of the C–H bond in CH ₃
-	-	-	-	1081	N–H stretching; asymmetrical vibration COO ⁻
-	-	-	-	1008	CN stretching
711	712	715	-	-	C–O bond stretching
-	-	-	-	479	Cu–N stretching mode
-	-	-	-	-	C–H bending vibration
-	-	-	-	-	Cu–N stretching vibration

Concerning the spectra of the copper/carbonaceous matrix composites, no major differences are observed as a function of the pyrolysis temperature (as shown in Figure 10 for the Gly-Cu 3.17% samples at 450, 500, and 550 °C). On the other hand, upon comparison with the spectra of the samples obtained from pristine glycine, noticeable changes are found (Table 9). These include the disappearance of the bands at 3074 cm⁻¹, 1583 cm⁻¹, and 712 cm⁻¹, a reduction in the intensity of the band at 1206 cm⁻¹, and the emergence of a primary band at 1564 cm⁻¹, followed by another at around 1500 cm⁻¹, along with three smaller bands between 1159 and 1008 cm⁻¹ (with the presence of the bands at 1081 and 1008 cm⁻¹ observed only in the sample with the highest Cu-doping). Additionally, a peak at 479 cm⁻¹ is observed solely in the Gly-Cu 3.17% sample. The frequency at 1564 cm⁻¹ is associated with –CN stretching vibrations [38], corresponding to the strong G band at 1563 cm⁻¹ attributed to graphite. The band at around 1500 cm⁻¹ corresponds to ν (C=C) or the asymmetric flexion (δ A) of the C–H bond in CH₃. Finally, the band at 479 cm⁻¹ is attributed to Cu–N stretching vibrations [39].

**Figure 10.** Infrared spectra of Cu-doped glycine thermolysis products obtained at 450, 500, and 550 °C; for comparison, pristine glycine treated at 500 °C is also shown.

3.8.2. Raman Spectra

The Raman spectra (Figure 11) exhibit peaks at 1364, 1527, and 1568 cm⁻¹. The peak at 1364 cm⁻¹ is assignable to the graphene D mode (disorder mode), akin to what is observed for pristine graphene at 1350 cm⁻¹. The peak at 1568 cm⁻¹ is a distinctive feature of

graphenic carbon (G-mode, graphitic mode), representing the in-plane vibration of sp^2 hybridized carbon atoms in the hexagonal lattice. The D band is associated with defects and disorder in the graphite structure due to intervalley scattering, while the G band corresponds to the E_{2g} vibration of sp^2 of C atoms. The peak at 1527 cm^{-1} is attributed to a C=C stretching vibration.

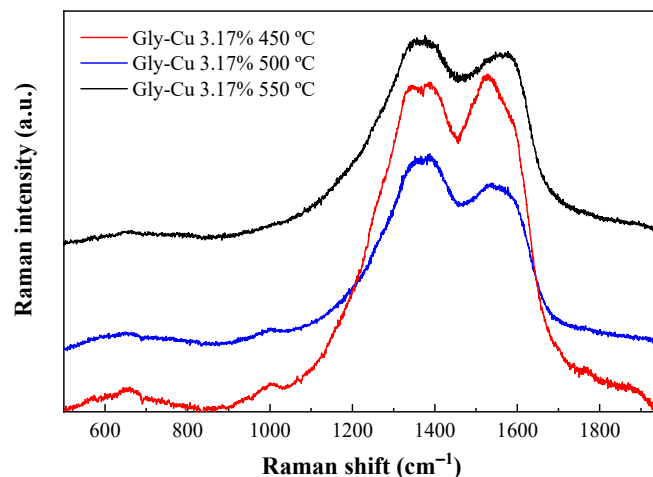


Figure 11. Raman spectra of Cu-doped glycine thermolysis products obtained at three temperatures.

3.9. UV-Vis Spectra

Figure 12 illustrates the UV-Vis spectra of Cu-doped glycine thermolysates obtained at 450, 500, and 550 °C, alongside the spectrum of pristine glycine pyrolyzed at 500 °C for comparison. The bandgaps, determined from the Tauc plot, were found to be 1.52, 1.65, and 1.22 eV for the samples with 0.16% Cu treated at 450, 500, and 550 °C, respectively. The bandgap for carbonized glycine at 500 °C was estimated at 0.90 eV. The higher bandgap observed for the Cu samples may be attributed to more localized orbitals around the Cu sites with increased energy.

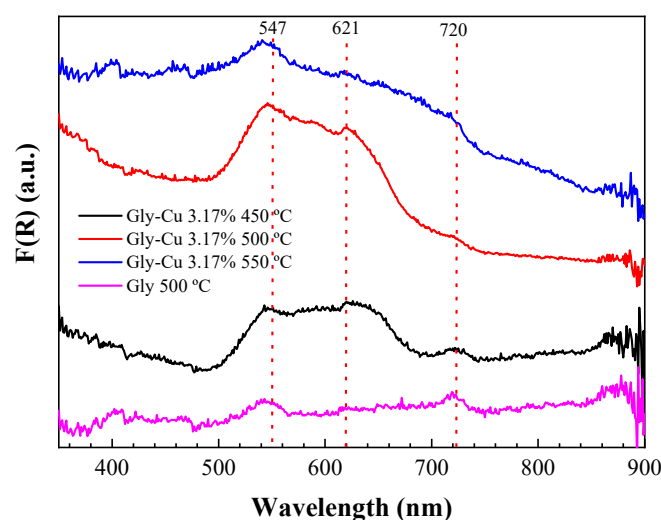


Figure 12. The UV-Vis spectra of the Cu-doped glycine thermolysates obtained at 450, 500, and 550 °C. For comparison purposes, the pristine glycine thermolysate spectrum after heating at 500 °C is also shown.

The band absorptions at 547 and 620 nm bear resemblance to those of copper (II) tetraphenylporphyrinate (CuTPP), where a Cu (II) ion is coordinated by four N atoms. These absorptions can be assigned to Q bands with some 'vibrational contour' resulting from the active band around 1507 cm^{-1} (C-C/C-N conjugated) in the infrared region,

corresponding to 0.19 eV, with an overtone around 540 nm. Additionally, they correspond to the intense band around 1159 cm^{-1} (C–N stretching) that corresponds to 0.14 eV (as observed in IR spectra) [40], with an overtone around 720 nm [40]. Both bands play a role in the coordination of Cu^{2+} for the N ligand. Furthermore, the consistently present bands at 547 and 620 nm across all samples may be associated with some sites involving pyrrole-like centers arranged in a ring, similar to what is observed in tetraphenyl porphyrin (H_2TPP) [41].

3.10. Conductivity Measurements

Figure 13 displays the conductance values of Gly 500 °C and Gly-Cu 3.17% 500 °C samples measured in the range between 100 Hz and 6 kHz, along with their corresponding least squares fits.

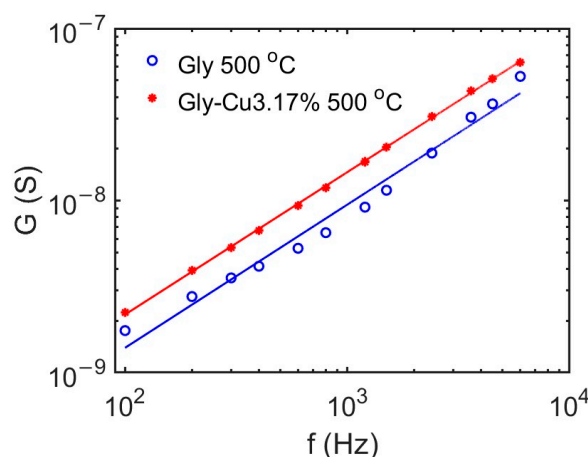


Figure 13. Log-log plot of the AC conductance values (points) measured for the material obtained from the thermal treatment of pristine glycine and Cu-doped glycine at 500 °C. Straight lines display the linear least squares fit of the measured data.

When prepared as described in Section 2.2, the samples constitute random mixtures of a conductor material (the sample) and an insulator (air). These systems are known to exhibit critical behavior due to percolation effects. In particular, it has been shown [42] that the AC conductivity of the mixture follows the power-law behavior near the percolation threshold:

$$\sigma(\omega) \sim \omega^x, \quad (1)$$

where σ is the effective bulk conductivity and ω is the angular frequency.

A very good fit between experimental measurements and the expected behavior of random conductor-insulator mixtures described by Equation (1) was found in this case. The critical exponent obtained from the measurements is $x = 0.83$ for both samples, in agreement with the previously reported value of $x = 0.86$ for a mixture of amorphous carbon and Teflon [43]. The measured results indicate a significant increase in the conductivity of the Cu-doped sample, in line with the results observed for electrodeposited particulate composite films [44].

3.11. THz Absorption Study

Carbon-based materials emerge as particularly appealing absorbers for terahertz applications [24]. Blends of these substances with transparent polymers in the terahertz spectral region provide a means to fine-tune absorption to a required value with high flexibility by adjusting the concentration of the carbon material. The results of THz measurements for the thermolysates of pristine glycine and glycine doped with 3.17% Cu obtained at 500 °C, both dispersed in PE, are presented in Figure 14. When comparing the absorption of the new material with reference materials described in the literature [29], it is evident

that its absorption is lower than that of Asbury graphite and needle coke mixed with polyethylene at the same concentration. This observation aligns with the lower carbon content in the composite.

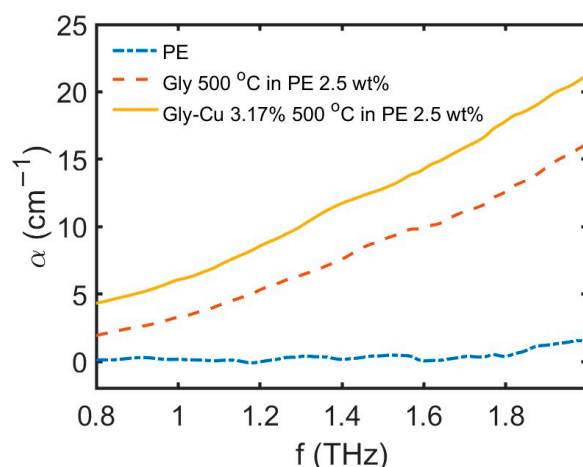


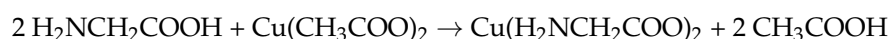
Figure 14. THz measurements on the Gly 500 °C and Gly-Cu 3.17% 500 °C samples dispersed in PE.

Moreover, terahertz measurements demonstrate that copper doping indeed enhances the absorption of radiation in the THz band. In the new copper-doped sample, the absorption appears to be comparable to that of coke, indicating an effective improvement in the material's THz absorption characteristics.

Consequently, the investigation into the terahertz spectrum properties of the new material reveals its excellent candidacy for applications as a terahertz absorber. Notably, copper doping emerges as a remarkably effective strategy for modulating the absorption properties of such materials, allowing for precise tuning when combined with adjustments to its concentration in a polymer matrix.

4. Discussion

In the conditions employed for the synthesis of the new material through a solid-state reaction, the formation of copper (II) glycinate is expected, following a non-redox dissociative substitution mechanism:



Consequently, copper (II) glycinate should be considered among the precursors of the new composite to justify its nature in terms of shape and composition. While copper acetate melts at 273 °C and glycine at 233 °C [45], copper glycinate chars at 213 °C and decomposes at 228 °C with gas evolution [45,46]. The presence of the latter component in the sample leads to its early decomposition and the generation of carbon, which subsequently dissolves into the melts of glycine and copper acetate to form a solid solution.

Regarding the composition of the new material, the presence of elemental copper, tenorite, and graphenic carbon is evident. However, the formation of copper (II) N-ring chelate complexes as a minority component requires careful analysis. The thermal behavior during the synthesis process implies the loss of the carboxyl group of the glycine/copper (II) complex at around 235 °C, the loss of aminic groups at around 286 °C, and the release of amine fragments in the range of 320–500 °C. This complexity makes it challenging to predict specific species (such as purine bases, ammeline, or others) resulting from the thermolysis. Nevertheless, it has been strongly evidenced that the pyrolysis of glycine between 300 and 400 °C leads to a mixture of C and C₂H₂N₂O₂ species (either 1,3-diazetidine-2,4-dione or 1,2-diazetidine-3,4-dione isomers, Figure 4 [20]). This supports the consideration that the ligands attached to copper (II) are provided by the diazinedione species (Figure 15). The UV-Vis and FTIR results support this possibility.

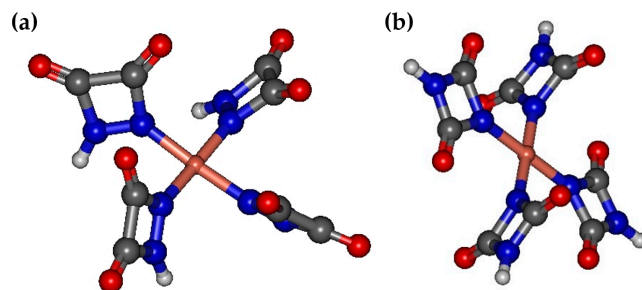


Figure 15. Ground state geometries of the tetra-coordinated Cu(II) complexes with (a) *cis* and (b) *trans* ligands obtained with the PM6 Hamiltonian.

A study of the proposed complexes with *cis* and *trans* configurations of the ligands has been conducted employing semi-empirical quantum chemistry methods. The ground-state geometries of the complexes, optimized using the PM6 Hamiltonian as implemented in MOPAC, are depicted in Figure 15. In both cases, the calculations reveal planar tetra-coordinated geometries characterized by robust interactions between the d electrons of the copper (II) ion and the in-plane ligand crystal field.

Therefore, copper occupies sites where diazetine-like nitrogen is present and, in a certain way, acts as a template, increasing these types of sites during synthesis. Similar sites without a centrally coordinated ion also seem to be present in thermolyzed glycine, as indicated by the weak bands at 547 and 720 nm, corresponding to the overtones of the two main IR bands. These diazetine-like sites are often found when pyrolyzing carbon, nitrogen, amino acids, or similar compounds [40]. Glycine likely easily condenses around a metallic ion, with nitrogen coordinated by the central ion. It is noteworthy to point out the broad hydrogen bond region between 3500 and 2800 cm^{-1} , attributable to both NH and OH stretching.

Figure 16 illustrates the electronic resonances resulting from the splitting of the d electronic levels of the Cu (II) ion due to the ligand crystal field effect, observed between 460 and 900 nm. The calculations were performed utilizing the INDO/S method. The figure also includes a visual representation of the associated absorption spectra in this band based on the computed transitions. A closer alignment between the experimental data and computations is observed for the copper (II) complex with the *trans* isomer ligand 1,3-diazetine-2,4-dione, characterized by relatively strong intensity peaks at 560, 575, and 765 nm.

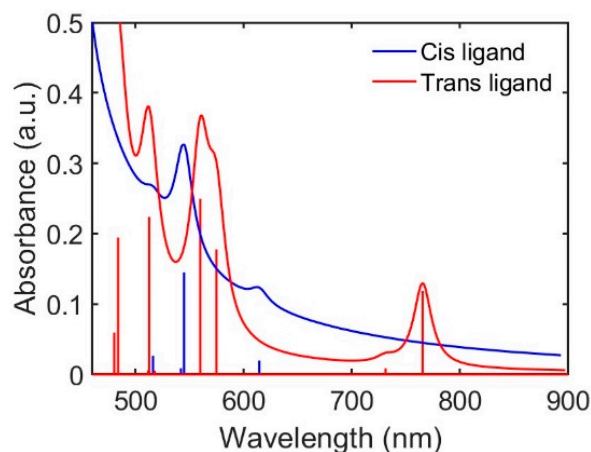


Figure 16. Calculated absorption spectra for the Cu (II) complexes with *cis* and *trans* ligand isomers.

Concerning conductivity measures, the obtained results suggest that the new copper/graphene composite material holds promise for significant applications in fields requiring high conductivity, such as electronics and energy storage, as demonstrated by Hidalgo-Manrique et al. [2] and Pan et al. [47].

Moreover, the measurement of radiation absorption in the THz band, a crucial parameter for materials used in THz shielding, indicates that copper doping enhances radiation absorption. Notably, the absorption in the copper-doped sample appears comparable to that of coke, a material recognized for its high absorption rate in the THz band. This significance is underscored by the potential applications of the new composite with copper doping as a shielding material in the THz region, akin to findings reported by Khattak et al. [48] and Song et al. [49]. Remarkably, these THz results align with the conductivity measurements, providing a coherent understanding of the material's performance across different parameters.

5. Conclusions

The reported composite material was produced through the thermal treatment of glycine and copper acetate monohydrate in varying ratios at temperatures ranging from 450 to 550 °C. According to EDS results, the composite consists of a carbonaceous matrix with a 6:1 mixture of graphenic carbon and a copper N-ring chelate resembling diazetidinedione ($C_2H_2N_2O_2$). Additionally, a significant proportion of copper (8%) is present in the free metallic form, with traces of tenorite (CuO). The inclusion of elemental copper results in a noteworthy increase in conductivity for the new copper/graphene composite. These findings suggest potential applications in fields requiring high conductivity, such as electronics and energy storage. Moreover, the results on THz radiation absorption indicate that the copper-doped material may enhance THz shielding. This could find applications in fields like medical imaging and security screening, where THz radiation is commonly used.

Author Contributions: Conceptualization, P.C.-P. and J.M.-G.; methodology, P.C.-P., R.C.D., J.M.-G. and P.M.-R.; software, P.C.-P.; validation, R.C.D. and Ó.R.-L.; formal analysis, P.C.-P., R.C.D., J.V.-C. and P.M.-R.; investigation, P.C.-P., R.C.D., J.M.-G., D.G.D., A.C., J.V.-C., Ó.R.-L., I.M.-M. and P.M.-R.; resources, P.C.-P., J.M.-G. and Ó.R.-L.; writing—original draft preparation, P.C.-P., R.C.D., J.M.-G. and P.M.-R.; writing—review and editing, P.M.-R.; visualization, P.C.-P., R.C.D., I.M.-M. and P.M.-R.; project administration, P.C.-P.; funding acquisition, P.C.-P. All authors have read and agreed to the published version of the manuscript.

Funding: This research was funded by the Spanish Ministerio de Ciencia e Innovación (MCIN) under project PID2020-119418GB-I00.

Data Availability Statement: The data supporting the findings of this study are available within the article.

Acknowledgments: XPS and electron microscopy characterizations were performed at CACTI (University of Vigo).

Conflicts of Interest: The authors declare no conflicts of interest.

References

1. Gupta, S.; Bharti, H. A review on copper-graphite composite material fabrication and its mechanical properties. *Int. J. Adv. Res. Innov. Ideas Educ.* **2016**, *2*, 594–599.
2. Hidalgo-Manrique, P.; Lei, X.; Xu, R.; Zhou, M.; Kinloch, I.A.; Young, R.J. Copper/graphene composites: A review. *J. Mater. Sci.* **2019**, *54*, 12236–12289. [[CrossRef](#)]
3. Cantürk, S.B.; Kováčik, J. Review of recent development in copper/carbon composites prepared by infiltration technique. *Energies* **2022**, *15*, 5227. [[CrossRef](#)]
4. Subramaniam, C.; Yamada, T.; Kobashi, K.; Sekiguchi, A.; Futaba, D.N.; Yumura, M.; Hata, K. One hundred fold increase in current carrying capacity in a carbon nanotube–copper composite. *Nat. Commun.* **2013**, *4*, 2202. [[CrossRef](#)] [[PubMed](#)]
5. Subramaniam, C.; Yasuda, Y.; Takeya, S.; Ata, S.; Nishizawa, A.; Futaba, D.; Yamada, T.; Hata, K. Carbon nanotube-copper exhibiting metal-like thermal conductivity and silicon-like thermal expansion for efficient cooling of electronics. *Nanoscale* **2014**, *6*, 2669–2674. [[CrossRef](#)] [[PubMed](#)]
6. Arnaud, C.; Lecouturier, F.; Mesguich, D.; Ferreira, N.; Chevallier, G.; Estournès, C.; Weibel, A.; Laurent, C. High strength—High conductivity double-walled carbon nanotube—Copper composite wires. *Carbon* **2016**, *96*, 212–215. [[CrossRef](#)]
7. Xu, G.; Zhao, J.; Li, S.; Zhang, X.; Yong, Z.; Li, Q. Continuous electrodeposition for lightweight, highly conducting and strong carbon nanotube-copper composite fibers. *Nanoscale* **2011**, *3*, 4215. [[CrossRef](#)] [[PubMed](#)]

8. Ahmad, H.; Markina, A.A.; Porotnikov, M.V.; Ahmad, F. A review of carbon fiber materials in automotive industry. *IOP Conf. Ser. Mater. Sci. Eng.* **2020**, *971*, 032011. [[CrossRef](#)]
9. Wazeer, A.; Das, A.; Abeykoon, C.; Sinha, A.; Karmakar, A. Composites for electric vehicles and automotive sector: A review. *Green Energy Intell. Transp.* **2023**, *2*, 100043. [[CrossRef](#)]
10. Li, X.; Zhou, C.; Overman, N.; Ma, X.; Canfield, N.; Kappagantula, K.; Schroth, J.; Grant, G. Copper carbon composite wire with a uniform carbon dispersion made by friction extrusion. *J. Manuf. Process.* **2021**, *65*, 397–406. [[CrossRef](#)]
11. Graça, I.; Seixas, T.; Ferro, A.C.; Guedes, M. Nanostructured copper-carbon nanotubes composites for aircraft applications. *Aircr. Eng. Aerosp. Technol.* **2018**, *90*, 1042–1049. [[CrossRef](#)]
12. Sundaram, R.M.; Sekiguchi, A.; Sekiya, M.; Yamada, T.; Hata, K. Copper/carbon nanotube composites: Research trends and outlook. *R. Soc. Open Sci.* **2018**, *5*, 180814. [[CrossRef](#)] [[PubMed](#)]
13. Kute, A.D.; Gaikwad, R.P.; Warkad, I.R.; Gawande, M.B. A review on the synthesis and applications of sustainable copper-based nanomaterials. *Green Chem.* **2022**, *24*, 3502–3573. [[CrossRef](#)]
14. Reljic, S.; Cuadrado-Collados, C.; Oliveira Jardim, E.; Farrando-Perez, J.; Martinez-Escandell, M.; Silvestre-Albero, J. Activated carbon materials with a rich surface chemistry prepared from L-cysteine amino acid. *Fluid Phase Equilib.* **2022**, *558*, 113446. [[CrossRef](#)]
15. Wei, L.; Guo, L.; Jiang, J.; Liu, M.; Yang, T. Influence of Fe₂O₃ on glycine pyrolysis characteristics and nitrogen conversion. *CIESC J.* **2019**, *70*, 1942–1950. [[CrossRef](#)]
16. Wang, W.; Tesio, A.Y.; Olivares-Marín, M.; Gómez Romero, P.; Tonti, D. Facile preparation of glycine-based mesoporous graphitic carbons with embedded cobalt nanoparticles. *J. Mater. Sci.* **2022**, *57*, 13403–13413. [[CrossRef](#)]
17. Leng, L.; Yang, L.; Zu, H.; Yang, J.; Ai, Z.; Zhang, W.; Peng, H.; Zhan, H.; Li, H.; Zhong, Q. Insights into glycine pyrolysis mechanisms: Integrated experimental and molecular dynamics/DFT simulation studies. *Fuel* **2023**, *351*, 128949. [[CrossRef](#)]
18. Li, J.; Wang, Z.; Yang, X.; Hu, L.; Liu, Y.; Wang, C. Evaluate the pyrolysis pathway of glycine and glycyglycine by TG–FTIR. *J. Anal. Appl. Pyrolysis* **2007**, *80*, 247–253. [[CrossRef](#)]
19. Hao, J.; Guo, J.; Ding, L.; Xie, F.; Xia, Q.; Xie, J. TG-FTIR, Py-two-dimensional GC–MS with heart-cutting and LC–MS/MS to reveal hydrocyanic acid formation mechanisms during glycine pyrolysis. *J. Therm. Anal. Calorim.* **2013**, *115*, 667–673. [[CrossRef](#)]
20. Weiss, I.M.; Muth, C.; Drumm, R.; Kirchner, H.O.K. Thermal decomposition of the amino acids glycine, cysteine, aspartic acid, asparagine, glutamic acid, glutamine, arginine and histidine. *BMC Biophys.* **2018**, *11*, 2. [[CrossRef](#)]
21. Shao, Y.; Hu, Z.-y.; Yao, Y.; Wei, X.-r.; Gao, X.-m.; Wu, Z.-x. Glycine-derived nitrogen-doped ordered mesoporous carbons with a bimodal mesopore size distribution for supercapacitors and oxygen reduction. *New Carbon Mater.* **2022**, *37*, 259–276. [[CrossRef](#)]
22. Cossu, F.L.; Poddighe, M.; Stagi, L.; Anedda, R.; Innocenzi, P. The birth of fluorescence from thermally polymerized glycine. *Macromol. Chem. Phys.* **2022**, *223*, 2200052. [[CrossRef](#)]
23. Siegel, P.H. Terahertz technology. *IEEE Trans. Microw. Theory Tech.* **2002**, *50*, 910–928. [[CrossRef](#)]
24. Venkatachalam, S.; Zeranska-Chudek, K.; Zdrojek, M.; Hourlier, D. Carbon-based terahertz absorbers: Materials, applications, and perspectives. *Nano Sel.* **2020**, *1*, 471–490. [[CrossRef](#)]
25. Castle, J.E. *Practical Surface Analysis by Auger and X-ray Photoelectron Spectroscopy*; John Wiley & Sons, Ltd.: Chichester, UK, 1984; Volume 6, p. 302.
26. Rodríguez-Carvajal, J. Recent developments of the program FULLPROF. *Comm. Powder Diffr. (IUCr) Newsl.* **2001**, *26*, 12–19.
27. Zanatta, A.R. Revisiting the optical bandgap of semiconductors and the proposal of a unified methodology to its determination. *Sci. Rep.* **2019**, *9*, 11225. [[CrossRef](#)]
28. Duvillaret, L.; Garet, F.; Coutaz, J.L. A reliable method for extraction of material parameters in terahertz time-domain spectroscopy. *IEEE J. Sel. Top. Quantum Electron.* **1996**, *2*, 739–746. [[CrossRef](#)]
29. Chamorro-Posada, P.; Vazquez-Cabo, J.; Rubinos-Lopez, O.; Martin-Gil, J.; Hernandez-Navarro, S.; Martin-Ramos, P.; Sanchez-Arevalo, F.M.; Tamashausky, A.V.; Merino-Sanchez, C.; Dante, R.C. THz TDS study of several sp(2) carbon materials: Graphite, needle coke and graphene oxides. *Carbon* **2016**, *98*, 484–490. [[CrossRef](#)]
30. Stewart, J.J.P. Optimization of parameters for semiempirical methods V: Modification of NDDO approximations and application to 70 elements. *J. Mol. Model.* **2007**, *13*, 1173–1213. [[CrossRef](#)] [[PubMed](#)]
31. Ridley, J.; Zerner, M. An intermediate neglect of differential overlap technique for spectroscopy: Pyrrole and the azines. *Theor. Chim. Acta* **1973**, *32*, 111–134. [[CrossRef](#)]
32. Zerner, M.C.; Loew, G.H.; Kirchner, R.F.; Mueller-Westerhoff, U.T. An intermediate neglect of differential overlap technique for spectroscopy of transition-metal complexes. Ferrocene. *J. Am. Chem. Soc.* **1980**, *102*, 589–599. [[CrossRef](#)]
33. Neese, F. The ORCA program system. *Wiley Interdiscip. Rev. Comput. Mol. Sci.* **2012**, *2*, 73–78. [[CrossRef](#)]
34. Schneider, C.A.; Rasband, W.S.; Eliceiri, K.W. NIH Image to ImageJ: 25 years of image analysis. *Nat. Methods* **2012**, *9*, 671. [[CrossRef](#)] [[PubMed](#)]
35. Biesinger, M.C.; Lau, L.W.M.; Gerson, A.R.; Smart, R.S.C. Resolving surface chemical states in XPS analysis of first row transition metals, oxides and hydroxides: Sc, Ti, V, Cu and Zn. *Appl. Surf. Sci.* **2010**, *257*, 887–898. [[CrossRef](#)]
36. Bansal, V.; Collins, D.; Luxton, T.; Kumar, N.; Shah, S.; Walker, V.K.; Shah, V. Assessing the impact of copper and zinc oxide nanoparticles on soil: A field study. *PLoS ONE* **2012**, *7*, e42663. [[CrossRef](#)] [[PubMed](#)]
37. Altomare, A.; Corriero, N.; Cuocci, C.; Falcicchio, A.; Moliterni, A.; Rizzi, R. QUALX2.0: A qualitative phase analysis software using the freely available database POW_COD. *J. Appl. Crystallogr.* **2015**, *48*, 598–603. [[CrossRef](#)]

38. Demadis, K.D.; Famelis, N.; Cabeza, A.; Aranda, M.A.G.; Colodrero, R.M.P.; Infantes-Molina, A. 2D Corrugated magnesium carboxyphosphonate materials: Topotactic transformations and interlayer “decoration” with ammonia. *Inorg. Chem.* **2012**, *51*, 7889–7896. [[CrossRef](#)]
39. Patil, B.; Prabhakar, B.; Kulkarni, V. Magnetic and spectral studies of copper (II) complexes with schiff bases derived from *o*-amino-benzoyl hydrazide. *Curr. Sci.* **1979**, *48*, 974–976.
40. Dante, R.C.; Chamorro-Posada, P.; Vázquez-Cabo, J.; Rubiños-López, Ó.; Sánchez-Árevalo, F.M.; Huerta, L.; Martín-Ramos, P.; Lartundo-Rojas, L.; Ávila-Vega, C.F.; Rivera-Tapia, E.D.; et al. Nitrogen-carbon graphite-like semiconductor synthesized from uric acid. *Carbon* **2017**, *121*, 368–379. [[CrossRef](#)]
41. Cioci, A.; Dante, D.G.; Martín-Ramos, P.; Chamorro-Posada, P.; Marabello, D.; Benzi, P.; Valsania, M.C.; Dante, R.C. CuTPP films grown on ceramic and ITO-coated glasses. *Fuller. Nanotub. Carbon Nanostruct.* **2023**, *31*, 856–867. [[CrossRef](#)]
42. Bergman, D.J.; Imry, Y. Critical behavior of the complex dielectric constant near the percolation threshold of a heterogeneous material. *Phys. Rev. Lett.* **1977**, *39*, 1222–1225. [[CrossRef](#)]
43. Song, Y.; Noh, T.W.; Lee, S.-I.; Gaines, J.R. Experimental study of the three-dimensional ac conductivity and dielectric constant of a conductor-insulator composite near the percolation threshold. *Phys. Rev. B* **1986**, *33*, 904–908. [[CrossRef](#)] [[PubMed](#)]
44. Jagannadham, K. Volume fraction of graphene platelets in copper-graphene composites. *Metall. Mater. Trans. A* **2013**, *44*, 552–559. [[CrossRef](#)]
45. Haynes, W.M. *CRC Handbook of Chemistry and Physics*, 97th ed.; CRC Press: Boca Raton, FL, USA, 2016; 2670p. [[CrossRef](#)]
46. Perry, D.L. *Handbook of Inorganic Compounds*; CRC Press: Boca Raton, FL, USA, 2016; 581p. [[CrossRef](#)]
47. Pan, C.; Gaur, A.P.S.; Lynn, M.; Olson, M.P.; Ouyang, G.; Cui, J. Enhanced electrical conductivity in graphene–copper multilayer composite. *AIP Adv.* **2022**, *12*, 015310. [[CrossRef](#)]
48. Khattak, M.I.; Anab, M.; Muqarrab, N. A duo of graphene-copper based wideband planar plasmonic antenna analysis for lower region of terahertz (THz) communications. *Prog. Electromagn. Res. C* **2021**, *111*, 83–96. [[CrossRef](#)]
49. Song, R.; Mao, B.; Wang, Z.; Hui, Y.; Zhang, N.; Fang, R.; Zhang, J.; Wu, Y.; Ge, Q.; Novoselov, K.S.; et al. Comparison of copper and graphene-assembled films in 5G wireless communication and THz electromagnetic-interference shielding. *Proc. Natl. Acad. Sci. USA* **2023**, *120*, e2209807120. [[CrossRef](#)]

Disclaimer/Publisher’s Note: The statements, opinions and data contained in all publications are solely those of the individual author(s) and contributor(s) and not of MDPI and/or the editor(s). MDPI and/or the editor(s) disclaim responsibility for any injury to people or property resulting from any ideas, methods, instructions or products referred to in the content.