

# Structural, electronic, and magnetic properties of Heusler $X_2MnZ$ with $X=Fe, Co, Ni, Cu, Ru, Rh, Pd, Ag, Pt, Au$ and $Z=Al, Si, Ga, Ge, In$ and $Sn$ : A systematic density functional theory investigation.

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## Abstract

An exhaustive study of the structural stability, electronic, and magnetic properties of the ternary  $X_2MnZ$  Heusler alloys, in cubic and tetragonal lattice cells, is presented. We performed density-functional-theory (DFT) calculations, as implemented in the SIESTA code, for  $X=Fe, Co, Ni, Cu, Ru, Rh, Pd, Ag, Pt, Au$ , and  $Z=Al, Si, Ga, Ge, In$ , and  $Sn$ . The spin polarized local electronic density of states for the different chemical components in the various alloys are calculated and the some particular examples are discussed. From those results we observe that the candidates for semi-metallic behaviour are  $Fe_2MnSi$ ,  $Co_2MnSi$ ,  $Co_2MnGe$  and  $Ru_2MnGe$ . A general feature is that Mn donates of the order of one electron to the other components. We calculated the magnetic moment per unit cell and the individual contributions coming from the other elements, and observe the change as a function of X and Z. The largest moment per unit cell corresponds to  $Fe_2MnIn$  ( $7.86 \mu_B$ ) and the smallest to  $Ru_2MnIn$  ( $2.15 \mu_B$ ). The crystalline structure, as well of the lattice parameter are discussed. We compare with the existing experimental data and find a fair agreement.

Keywords: Structure, electronic, and magnetic properties of Heusler alloys, DFT calculations,

## INTRODUCTION

Heusler alloys (HA) are some of the most interesting magnetic systems. They are ternary alloys with a chemical formula  $X_2MnZ$ , that crystallize with a  $L2_1$  structure.[? ? ? ] This structure can be described by four interpenetrating *fcc* sublattices; two of them occupied by the X atoms, and the other two by Mn and Z elements. In general, the X element is a transition or noble metal and Z is one element of the IIIA (Al, Ga, In), IVA (Si, Ge, Sn) or VA (Sb) columns of the periodic table.

As can be observed in Figure 1, in the structural ground state, the Mn atoms (purple circles) are surrounded by eight first neighbors of type X (pink circles) and six second neighbors of type Z (light brown circles). Thus, the shortest Mn-Mn pair occurs at the third neighbor shell. Due to the large distance between Mn atoms it is believed that the magnetic coupling is via the itinerant electrons of the X and Mn atoms. At finite temperatures, when the atoms exchange lattice sites, the magnetic interactions, sign and magnitude, will be determined by the amount of spatial disorder present in the system i.e., how many first, second and third Mn neighbors pairs are present in the system.

In all the HA, Mn plays an important role and depending on the chemical composition and the atomic order, they can be ferromagnets, ferrimagnets, antiferromagnets of various types, or helicoidal. In the case in which the X component is non-magnetic the magnetic moment is mainly localized in the Mn atom and can be as high as  $4.3 \mu_B$  ( $Pd_2MnIn$ ).[? ] On the other hand, if X is magnetic, both magnetic elements can give rise to  $5.11 \mu_B$  ( $Co_2MnGe$ ).[? ] Furthermore, they have very high Curie temperatures that can be as high as 985 K ( $Co_2MnSi$ ).[? ]

As a function of temperature, the three chemical species exchange lattice sites and produce a rich variety of atomic disordered systems that influence the magnetic properties. These alloys present a great opportunity to understand the interplay between magnetism and atomic order.[? ? ]. In addition, in some systems, like Ni-Mn-Ga, the coupling between structure and magnetism, induce martensitic transformations that bring interesting magnetic shape-memory and magnetic superelasticity properties.[? ].

One more property that makes these systems subject of intensive research is that some are half-metallic. There is a gap in the spin down sub-band at the Fermi level. Thus, they can be used to provide electrons with only one spin polarization; the key property for

spintronics [? ? ] and magnetically driven actuators[? ].

Our interest here is to study in a systematic way the electronic, magnetic and stability properties of 60 systems; i.e.  $X=\text{Fe, Co, Ni, Cu, Ru, Rh, Pd, Ag, Pt and Au}$  and  $Z = \text{Al, Ga, In, Si, Ga, and Sn}$ . We present in Figure 2, the periodic table showing the elements that form the  $X_2\text{MnZ}$  systems reported here: Mn, the transition metals in yellow (X element) and the IIIA and IVA column elements in green (Z element).

In Section II we present the model and computational details. In Sec. III we discuss our results; first we address the electronic structure, then we discuss the magnetic properties and give results for the magnetic properties. Finally, we report on the ground state crystalline structure. Comparison with experimental results are also given.

## DETAILS OF THE COMPUTATIONAL PROCEDURE AND MODEL

Our calculations were performed using the SIESTA DFT package,[? ] which employs numerical pseudo-atomic orbitals as basis sets to solve the single-particle Kohn-Sham equations. In this work we used the Perdew-Burke-Ernzerhof form of the generalized gradient approximation (GGA) for the exchange and correlation potential,[? ] and the atomic cores were described by nonlocal norm-conserving scalar-relativistic Troullier-Martins pseudopotentials,[? ] factorized in the Kleinman-Bylander form.[? ] The pseudo-potentials for all the different elements studied here, were generated using the valence configurations. More details on the reliability, basis sets used, and about the pertinent tests that we carried out, can be found in the literature.[? ? ] The valence states were described using a double- $\zeta$  doubly polarized basis set.

An electronic temperature of 25 meV for smearing, and a 250 Ry energy cutoff to define the real-space grid for numerical calculations involving the electron density was implemented. We have tested larger cutoffs and lower electronic temperatures for some particular cases, and verified that they do not modify substantially the results. To optimize the lattice constant and atomic positions we performed a local relaxation using the conjugate gradient algorithm,[? ] starting from cubic and tetragonal basic lattice cells. The structural optimization was finalized when each force component, on each atom in the cluster, was smaller than 6 meV/Å.

All the calculations were performed assuming cubic and tetragonal (for Pd, Ag, Pt and

Au) lattice vectors with an atomic base of 16 atoms (see Figure 1) keeping the  $X_2MnZ$  stoichiometry. The binding energy  $E_b$  is defined as follows:

$$E_b(X_8Mn_4Z_4) = \frac{E_{\text{Total}}(X_8Mn_4Z_4) - 8E_{\text{atom}}(X) - 4E_{\text{atom}}(Mn) - 4E_{\text{atom}}(Z)}{16} , , \quad (1)$$

## RESULTS

We calculated the electronic structure of the 60 systems;  $X_2MnZ$  with  $X=Fe, Co, Ni, Cu, Ru, Rh, Pd, Ag, Pt$  and  $Au$  and  $Z = Al, Ga, In, Si, Ga,$  and  $Sn$ . We included in the calculation all the  $s, p,$  and  $d,$  valence electrons. Here, we present in detail the spin up and spin down local electronic density of states (SPLDS) for some representative systems, in the energy region around the Fermi energy. We show also how the total spin up and down electronic densities of states evolve as we substitute for a particular  $X$  series the  $Z$  element. The physical properties are discussed in three tables for each series  $X=3d, 4d,$  and  $Pt,$  and noble metals. First, we note how the electrons distribute in the ordered alloy and from the local electronic density of states we analyze the half-metallic behavior. Then from the spin polarized density of states we calculate the magnetic moment at each chemical component, and the total magnetic moment per unit cell. Finally we give the cohesive energy of the alloys and the equilibrium crystalline structures and lattice constants.

### **$X_2MnZ$ with $X = Fe, Co, Ni,$ and $Z = Al, Ga, In, Si, Ga,$ and $Sn$**

In Figure 3 we show the results for the spin polarized local density of states (SPLDS) for  $Co_2MnSi$  alloy, on each of the chemical elements. In the upper panel we present the total spin-polarized electronic density of states (black curve), and the  $3d$  (in blue),  $4s$  (in red),  $Co$  electronic contributions. In the central panel we show the SPLDS on the  $Mn$  sites. One sees that the  $3d$  electron band with spin down is mostly empty, giving rise to a large magnetic moment. In the lower one the  $3s$  (in yellow) and  $3p$  (in green) contribution on the  $Si$  atoms are shown. As reported previously, we observe that the  $3d$  electrons of  $Co$  and  $Mn$  are the ones that determine the magnetic properties of the system. The  $4s$  electrons form a wide conducting band. On the other hand, the  $Si$  spectrum shows a wide  $3p$  band partially polarized and a  $3s$  band deep in energy. One more characteristic of this system is

TABLE I. Transferred electronic charge between the atoms of the  $X_2MnZ$  Heusler like systems with  $X = Fe, Co, Ni$  and  $Z = Al, Ga, In, Si, Ge$  and  $Sn$ . In the last column we give the ratio between the electronic densities of states of opposite spins (the smaller over the larger) at the Fermi Energy.

$X_2YZ$	X(Charge) (e)	Mn(Charge) (e)	Z(Charge) (e)	TLDS ratio
$Fe_2MnAl$	0.473	-0.899	-0.048	0.16
$Fe_2MnGa$	0.510	-0.674	-0.346	0.21
$Fe_2MnIn$	1.030	-0.784	-1.276	0.58
$Fe_2MnSi$	1.035	-0.706	-1.365	0.00
$Fe_2MnGe$	0.815	-0.846	-0.784	0.65
$Fe_2MnSn$	1.024	-0.693	-1.354	0.22
$Co_2MnAl$	0.348	-0.944	0.250	0.41
$Co_2MnGa$	0.317	-0.723	0.090	0.43
$Co_2MnIn$	0.941	-0.851	-1.030	0.23
$Co_2MnSi$	0.705	-0.731	-0.678	0.00
$Co_2MnGe$	0.407	-0.949	0.136	0.06
$Co_2MnSn$	0.957	-0.765	-1.148	0.25
$Ni_2MnAl$	0.315	-1.036	0.407	0.37
$Ni_2MnGa$	0.177	-0.895	0.540	0.51
$Ni_2MnIn$	0.814	-1.024	-0.603	0.27
$Ni_2MnSi$	0.600	-0.859	-0.341	0.85
$Ni_2MnGe$	0.199	-1.063	0.664	0.34
$Ni_2MnSn$	0.884	-0.881	-0.886	0.74

the negligible spin-down LDS at the Fermi energy, which gives the half-metallic character of this alloy.[? ]

In Figure 4, we show the electronic structure of the six systems  $Co_2MnZ$ ;  $Z=Al, Ga,$  and  $In$ , in the upper figures and  $Si, Ge,$  and  $Sn$ , in the lower part. We see that the general characteristics around the Fermi energy prevail, but the value in the spin-down density of states is smaller for the  $Si, Ge,$  and  $Sn$ , systems. Also, a general behavior is the donation of electrons from the  $Mn$  atoms, discussed below.

In Table ??, we summarize the electron distribution in the three atomic components of the  $3d$  magnetic series. In the second to fourth columns, we give the difference between the total number of electron of each element in the atomic case minus that number when

TABLE II. The magnetic moment per unit cell ( $\mu_B$ ), and the magnetic moments of each of the elements in the  $X_2MnZ$  Heusler alloys. The X element is a magnetic  $3d$  element, and  $Z = Al, Ga, In, Si, Ge,$  and  $Sn$ . The values in brackets and bold face are experimental data.

$X_2MnZ$	(Mag.Mom.)/ (per unit cell)	$X(\mu_B)$	$Mn(\mu_B)$	$Z(\mu_B)$
$Fe_2MnAl$	6.40	2.193	2.413	-0.400
$Fe_2MnGa$	7.22	2.374	2.828	-0.362
$Fe_2MnIn$	7.86	2.428	3.134	-0.130
$Fe_2MnSi$	3.00 ( <b>2.2</b> )[? ]	0.081	2.828	0.010
$Fe_2MnGe$	4.91	1.325	2.502	-0.242
$Fe_2MnSn$	7.10	2.118	3.043	-0.178
$Co_2MnAl$	4.07 ( <b>4.01</b> )[? ]	0.612	3.066	-0.226
$Co_2MnGa$	4.21 ( <b>4.05</b> )[? ]	0.605	3.229	-0.230
$Co_2MnIn$	4.97	0.799	3.415	-0.046
$Co_2MnSi$	5.00 ( <b>5.07</b> )[? ]	0.893	3.253	-0.037
$Co_2MnGe$	5.00 ( <b>5.11</b> )[? ]	0.901	3.395	-0.196
$Co_2MnSn$	5.09 ( <b>5.08</b> )[? ]	0.817	3.491	-0.038
$Ni_2MnAl$	4.28	0.304	3.742	-0.019
$Ni_2MnGa$	4.33 ( <b>4.17</b> )[? ]	0.299	3.871	-0.138
$Ni_2MnIn$	4.45 ( <b>4.40</b> )[? ]	0.244	3.942	0.020
$Ni_2MnSi$	4.07	0.178	3.665	0.045
$Ni_2MnGe$	4.17	0.217	3.845	-0.106
$Ni_2MnSn$	4.29( <b>4.05</b> )[? ]	0.176	3.897	0.036

they are alloyed. For the case of Fe series, Mn and the Z atoms donate electrons to Fe. In the Co and Ni series the behavior is similar except for  $Z = Al, Ga,$  and  $Ge$ . The magnetic elements Fe, Co, and Ni always increase their number of electrons, reducing thereby their atomic magnetic moment, as will be commented below.

To give an idea of the half-metallic behavior, we show in the last column the ratio between the values of the total electronic densities of states (SPLDS) of opposite spins at the Fermi level. The systems with very low ratio are  $Fe_2MnSi, Co_2MnSi$  and  $Co_2MnGe$ . A finite gap in the spin-down band makes these systems half-metallic and potential candidates for spin valves.

In Table ??, we show the results for the magnetic moment per unit cell and the contri-

TABLE III. Structural properties of the  $X_2MnZ$  Heusler alloys. The X element is a magnetic  $3d$  element, and  $Z = Al, Ga, In, Si, Ge,$  and  $Sn$ . The crystalline structure for all the systems is cubic. The lattice constant is given in ( $\text{\AA}$ ), and the binding energy in eV/atom. The values in brackets and bold face, are experimental data.

$X_2MnZ$	Lattice Parameter ( $\text{\AA}$ )	Binding Energy (eV/atom)
$Fe_2MnAl$	5.909	-4.288
$Fe_2MnGa$	6.006	-3.956
$Fe_2MnIn$	6.197	-3.659
$Fe_2MnSi$	5.651 ( <b>5.664</b> )[? ]	-4.767
$Fe_2MnGe$	5.789 ( <b>5.76</b> )	-4.385
$Fe_2MnSn$	6.142	-3.990
$Co_2MnAl$	5.789( <b>5.756</b> )[? ]	-4.736
$Co_2MnGa$	5.873( <b>5.770</b> )[? ]	-4.347
$Co_2MnIn$	6.078	-3.988
$Co_2MnSi$	5.694( <b>5.654</b> )[? ]	-5.080
$Co_2MnGe$	5.825( <b>5.743</b> )[? ]	-4.772
$Co_2MnSn$	6.066( <b>6.000</b> )[? ]	-4.382
$Ni_2MnAl$	5.909	-4.477
$Ni_2MnGa$	5.993( <b>5.825</b> )[? ]	-4.131
$Ni_2MnIn$	6.162( <b>6.068</b> )[? ]	-3.811
$Ni_2MnSi$	5.766	-4.664
$Ni_2MnGe$	5.909	-4.429
$Ni_2MnSn$	6.138( <b>6.052</b> )[? ]	-4.111

butions coming from the three chemical elements. In the cases where experimental values are reported we show them in brackets and bold face. It is worth noticing that most of the experimental values are reported at high temperatures, at which there may be a considerable amount of chemical disorder. A direct comparison is not possible, since chemical disorder may modify the magnetic moments.[? ]

The highest values are obtained for the Fe series. In this series the highest magnetic moment per unit cell is  $7.86 \mu_B$  for  $Fe_2MnIn$ . The partial contributions for Fe, Mn and In are 2.428, 3.13 and  $-0.130 \mu_B$ , respectively. The lowest occurs in  $Fe_2MnSi$ ;  $3 \mu_B$  per unit cell, and the partial values are 0.081, 2.828 and  $0.010 \mu_B$  for Fe, Mn, and Si, respectively. In

this case there are some experiments that report the properties in a chemically disordered state[? ]. The authors report values of 1.35 and 2.2  $\mu_B$ , per unit cell, at 77 and 4.2 K, respectively. For comparison with the theoretical calculation, we take the value measured at the lowest temperature. It is also important to note that the coupling of Mn with the Z element, except for the case of Si, is antiferromagnetic.

The Co series is the most studied and experimental results for five out of the six cases are reported. The largest value for the magnetic moment per unit cell ( $5.09\mu_B$ ) is obtained for  $\text{Co}_2\text{MnSn}$  which is in very good agreement with the experimental value ( $5.08 \mu_B$ ). In the other cases the theoretical and experimental values are in fair agreement. The Mn magnetic moment is larger than for the cases of the Fe systems and the magnetic coupling between Mn and the Z elements is antiferromagnetic in all cases.

In the Ni series the largest value for the magnetic moment per unit cell corresponds to  $\text{Ni}_2\text{MnIn}$  ( $4.45 \mu_B$ ), a value very similar to the one reported experimentally ( $4.40 \mu_B$ ). Manganese shows also the largest magnetic moment value ( $3.94 \mu_B$ ) of the  $3d$  alloys. The other two cases reported experimentally,  $\text{Ni}_2\text{MnGa}$  and  $\text{Ni}_2\text{MnSn}$ , differ more from our ground state (zero K) calculations. For these series In, Si, and Sn couple ferromagnetically to the other two chemical elements. That is not the case of  $Z = \text{Al, Ga, and Ge}$ , whose coupling is antiferromagnetic.

In Table ?? we give the results for the lattice parameter in  $\text{\AA}$  and the cohesive energy in eV. The crystalline structure for all the  $3d$  alloys is cubic. In the Fe series, the smallest lattice parameter for the elements in Column IIIA, of the periodic table, corresponds to Al ( $5.909 \text{\AA}$ ) and the highest cohesive energy ( $4.288 \text{ eV}$ ). On the other hand, the largest lattice parameter of the elements in that column is attained in the In case ( $6.197 \text{\AA}$ ) and the smallest cohesive energy ( $3.659 \text{ eV}$ ). For the elements in the IVA column a similar trend is observed: the smallest and largest lattice constants and the highest and smallest cohesive energy correspond to Si and Sn. In this series the experimental value reported by Niculescu et al. [? ] for  $\text{Fe}_2\text{MnGe}$ , differs only by  $0.029 \text{\AA}$ .

As mentioned above, the Co series is the most studied, with the exception of the In system, there are experimental results for the rest. In general the calculated values are in very good agreement with the measurement. The largest discrepancy corresponds to  $\text{Co}_2\text{MnGa}$  ( $0.103 \text{\AA}$ )

The Ni Heusler alloys that have been analysed experimentally are those with Ga, In and



TABLE IV. Transferred electronic charges between the atoms of the  $X_2MnZ$  Heusler like systems with  $X = Ru, Rh, Pd, Pt$  and  $Z = Al, Ga, In, Si, Ge$  and  $Sn$ . In the last column we give the ratio between the electronic densities of states of opposite spins (the larger over the smaller).

$X_2YZ$	X(Charge) (e)	Mn(Charge) (e)	Z(Charge) (e)	DOS ratio
$Ru_2MnAl$	0.767	-1.125	-0.411	0.24
$Ru_2MnGa$	0.893	-0.980	-0.807	0.10
$Ru_2MnIn$	1.406	-0.930	-1.881	0.24
$Ru_2MnSi$	1.189	-0.701	-1.677	0.02
$Ru_2MnGe$	1.026	-1.086	-0.965	0.00
$Ru_2MnSn$	1.321	-0.913	-1.728	0.02
$Rh_2MnAl$	0.697	-1.160	-0.233	0.06
$Rh_2MnGa$	0.800	-1.100	-0.500	0.16
$Rh_2MnIn$	1.283	-1.060	-1.507	0.19
$Rh_2MnSi$	1.031	-0.819	-1.241	0.91
$Rh_2MnGe$	0.826	-1.113	-0.538	0.87
$Rh_2MnSn$	1.208	-0.943	-1.475	0.57
$Pd_2MnAl$	0.332	-1.032	0.369	0.40
$Pd_2MnGa$	0.358	-1.100	0.383	0.34
$Pd_2MnIn$	0.840	-1.081	-0.598	0.37
$Pd_2MnSi$	0.575	-0.915	-0.233	0.62
$Pd_2MnGe$	0.226	-1.069	0.619	0.74
$Pd_2MnSn$	0.816	-0.906	-0.726	0.39
$Pt_2MnAl$	0.354	-1.156	0.448	0.54
$Pt_2MnGa$	0.433	-1.175	0.310	0.73
$Pt_2MnIn$	0.971	-1.182	-0.761	0.71
$Pt_2MnSi$	0.610	-0.940	-0.281	0.69
$Pt_2MnGe$	0.338	-1.119	0.443	0.76
$Pt_2MnSn$	0.944	-1.010	-0.880	0.95

Sn. The calculated (experimental) values for the lattice constants, all in angstroms, are 5.993 (5.825), 6.162 (6.008), 6.138 (6.052), respectively.

## $X_2MnZ$ with $X = Ru, Rh, Pd, Pt$ , and $Z = Al, Ga, In, Si, Ga$ , and $Sn$

Now we consider the series of  $4d$  elements and  $Pt$   $5d$ . In Figure 5 we show the results for the spin polarized local density of states (SPLDS) for the  $Rh_2MnGe$  alloy, on each of the chemical elements, with  $L2_1$  crystallographic structure. It is worth noticing that the vertical scale is different in the three panels. In the upper panel we present the total SPLDS (black curve), and the  $4d$  (in purple)  $Rh$  electronic contribution. In this case the band spin splitting is small and one expects a reduced magnetic moment. In the central panel we show the SPLDS on the  $Mn$  sites. One sees that the  $3d$  electron band with spin down is mostly empty, giving rise to an important contribution to the alloy magnetic moment. In the lower figure, the  $Ge$  electronic contribution is presented; the  $4s$  and  $3d$  bands are completely full and deeper in energy. The only valence electrons are the  $4p$  (in green) states.

As mentioned above the  $4d$   $Rh$  band appears with similar occupation of spin-up and spin-down and observe that the  $3d$  electrons of  $Mn$  are the ones that determine the magnetic properties of the system. The  $4s$  electrons form a wide conducting band. On the other hand, the  $Ge$  spectrum shows a wide  $4p$  band partially polarized.

In Figure 6 we show how the electronic structure changes as the  $Z$  element goes from  $Al$  to  $In$  (Group IIIA) and from  $Si$  to  $Sn$  (Group IVA). The spin-down band below the Fermi energy is mainly produced by the  $X$  element whereas the spin-down band above the Fermi energy comes from the  $3d$   $Mn$  electrons. One further observation is that the gap in the spin-down band observed for  $X=Co$  and the  $Z=Si, Ge$  and  $Sn$  is not present in these series. A similar behavior is obtained for  $X=Ru, Pd$ , and  $Pt$

In Table ??, we summarize the electron distribution in the three atomic components of the  $4d$  magnetic series and the  $5d$   $Pt$ . In the second to fourth columns, we give the difference between the total number of electrons of each element in the atomic case minus that number when they are alloyed. In all the cases, electrons are transferred to the  $X$  elements ( $Ru, Rh, Pd$ , and  $Pt$ ).  $Mn$  donates electrons as much as 1.182, in the case of  $Pt_2MnIn$ , but in most of the cases the transfer is about 1 electron. The  $Z$  element acts also as a donor for the  $Ru$  and  $Rh$ . In the other two series ( $Pd$  and  $Pt$ ) in the cases of  $X=Al, Ga$ , and  $Ge$  behave as an acceptor of electrons and for  $X=In, Si$ , and  $Sn$  they are donors.

To give an idea of the half-metallic behavior, we show in the last column the ratio between the values of the total electronic densities of states with opposite spins at the Fermi level.

TABLE V. Magnetic moment per unit cell ( $\mu_B$ ), and the magnetic moments of each of the elements in the  $X_2MnZ$  Heusler alloys. The X component is a 4d element and Pt, and Z = Al, Ga, In, Si, Ge, and Sn. The values in brackets are experimental data.

$X_2MnZ$	Mag. Mom. per unit cell( $\mu_B$ )	X( $\mu_B$ )	Mn( $\mu_B$ )	Z( $\mu_B$ )
Ru <sub>2</sub> MnAl	2.15	-0.379	2.868	0.039
Ru <sub>2</sub> MnGa	2.22	-0.433	3.092	-0.011
Ru <sub>2</sub> MnIn	2.30	-0.526	3.328	0.021
Ru <sub>2</sub> MnSi	3.05	-0.089	3.191	0.039
Ru <sub>2</sub> MnGe	3.06	-0.134	3.316	0.013
Ru <sub>2</sub> MnSn	3.11	-0.229	3.483	0.086
Rh <sub>2</sub> MnAl	4.29	0.237	3.860	-0.037
Rh <sub>2</sub> MnGa	4.32	0.222	3.960	-0.080
Rh <sub>2</sub> MnIn	4.49	0.211	4.090	-0.015
Rh <sub>2</sub> MnSi	4.59	0.335	3.907	0.017
Rh <sub>2</sub> MnGe	4.65 ( <b>4.30</b> )[? ]	0.348	3.995	-0.038
Rh <sub>2</sub> MnSn	4.74	0.310	4.075	0.039
Pd <sub>2</sub> MnAl	4.24( <b>4.40</b> )[? ]	0.059	4.150	-0.023
Pd <sub>2</sub> MnGa	4.30	0.084	4.242	-0.109
Pd <sub>2</sub> MnIn	4.33( <b>4.30</b> )[? ]	0.043	4.303	-0.053
Pd <sub>2</sub> MnSi	4.18	0.054	4.096	-0.024
Pd <sub>2</sub> MnGe	4.21( <b>3.2</b> )[? ]	0.076	4.181	-0.115
Pd <sub>2</sub> MnSn	4.32( <b>4.23</b> )[? ]	0.032	4.246	0.008
Pt <sub>2</sub> MnAl	4.21	0.090	4.051	-0.018
Pt <sub>2</sub> MnGa	4.25	0.105	4.121	-0.084
Pt <sub>2</sub> MnIn	4.35	0.087	4.220	-0.049
Pt <sub>2</sub> MnSi	3.96	0.048	3.924	-0.063
Pt <sub>2</sub> MnGe	4.15	0.086	4.065	-0.083
Pt <sub>2</sub> MnSn	4.17	0.040	4.128	-0.042

In these systems the half-metallic properties have not been reported experimentally. Our results indicate that a candidate may be Ru<sub>2</sub>MnGe with a negligible ratio. Other two systems with a small value (0.02) are Ru<sub>2</sub>MnSi and Ru<sub>2</sub>MnSn.

In Table ??, we show the results for the magnetic moment per unit cell and the contributions coming from the three chemical elements. In the cases where experimental values are reported, we show them in brackets and bold face. In the Ru series, this element couples

TABLE VI. Structural properties of the  $X_2MnZ$  Heusler alloys. X is an element of the 4d series, and Pt, and Z = Al, Ga, In, Si, Ge, and Sn. The crystalline structure are cubic or tetragonal, when both structures are close in energy their characteristics are given. The lattice constant is given in ( $\text{\AA}$ ), and the binding energy in eV/atom. The values in brackets are experimental data.

$X_2MnZ$	Lattice Parameter ( $\text{\AA}$ )	Binding Energy (eV/atom)
$Ru_2MnAl$	6.054	-5.475
$Ru_2MnGa$	6.138	-5.052
$Ru_2MnIn$	6.281	-4.751
$Ru_2MnSi$	5.958	-5.842
$Ru_2MnGe$	6.066	-5.516
$Ru_2MnSn$	6.284	-5.143
$Rh_2MnAl$	6.120	-5.298
$Rh_2MnGa$	6.210	-4.897
$Rh_2MnIn$	6.352	-4.660
$Rh_2MnSi$	6.023	-5.470
$Rh_2MnGe$	6.138 ( <b>5.998</b> )[? ]	-5.207
$Rh_2MnSn$	6.350	-4.965
$Pd_2MnAl$	$2 \times 6.343$ , 6.088 6.246( <b>6.165</b> )[? ]	-4.182 -4.180
$Pd_2MnGa$	$2 \times 6.430$ , 6.138 6.330	-3.856 -3.854
$Pd_2MnIn$	$2 \times 6.612$ , 6.268 6.486( <b>6.373</b> )[? ]	-3.635 -3.633
$Pd_2MnSi$	$2 \times 6.194$ , 6.129 6.162	-4.202 -4.201
$Pd_2MnGe$	6.282 ( <b>6.134</b> ) [? ]	-4.050
$Pd_2MnSn$	6.486( <b>6.380</b> )[? ]	-3.864
$Pt_2MnAl$	$2 \times 6.505$ , 5.817 ( <b>6.24</b> )[? ]	-5.286
$Pt_2MnGa$	$2 \times 6.594$ , 5.872 ( <b>6.16</b> )[? ]	-4.865
$Pt_2MnIn$	$2 \times 6.820$ , 5.923	-4.649
$Pt_2MnSi$	$2 \times 6.561$ , 5.552	-5.208
$Pt_2MnGe$	$2 \times 6.690$ , 5.665	-5.004
$Pt_2MnSn$	$2 \times 6.887$ , 5.806	-4.834

antiferromagnetically to Mn, but the Z elements couple ferromagnetically to give rise to magnetic moments per unit cell that goes from 2.15 to 3.11  $\mu_B$ . In the Rh, Pd, and Pt series the X elements and Mn contribute positively to produce magnetic moments per unit cell in the range 3.96 (Pt<sub>2</sub>MnSi) to 4.74 (Rh<sub>2</sub>MnSn)  $\mu_B$ .

It is important to recall that these results are obtained with an atomic base of 16 atoms, and all are ferromagnetic. In the reported literature, the Mn magnetic moment in Rh<sub>2</sub>MnGe, Pd<sub>2</sub>MnGe, Pd<sub>2</sub>MnGe, and Pd<sub>2</sub>MnSn, couple ferromagnetically but in Pd<sub>2</sub>MnAl, Pd<sub>2</sub>MnIn, Pt<sub>2</sub>MnAl, and Pt<sub>2</sub>MnGa they couple antiferromagnetically. The later systems must be studied with a large enough atomic unit to be able to describe the three different types of antiferromagnetism characteristic of Heusler Alloys. In the ferromagnetic cases the agreement between theory and experiment are very good, except for the case of Pd<sub>2</sub>MnGe, in which the calculated magnetic moment is 4.21  $\mu_B$  and the one measured[?] is 3.2  $\mu_B$

In Table ?? we give the results for the lattice parameter in Å and the cohesive energy in eV. The Ru, Rh and Pt series are cubic. In the Pd series, four cases are tetragonal (Z=Al, Ga, In, and Si). In the Table we give the dimensions of the base and height of the cell. The cases that we can compare are those that are ferromagnetic (Rh<sub>2</sub>MnGe, Pd<sub>2</sub>MnGe, and Pd<sub>2</sub>MnSn). The comparison with the experimental values are good. In the antiferromagnetic cases, we give the values of the ferromagnetic phase and include the measured values.

### **X<sub>2</sub>MnZ with X = Cu, Ag, Au, and Z = Al, Ga, In, Si, Ga, and Sn**

The cases in which the X element is a noble metal, X= Cu, Ag and Au are discussed now. In Fig. 7 we present the total SPLDS and the individual SPLDS for the case Cu<sub>2</sub>MnAl. In the upper figure, we see that the Cu *d* sub-bands are completely filled (blue curve). In contrast, as seen in the middle figure, the spin-down Mn sub-band is mostly empty, and as we will discuss bellow, gives rise to the magnetic properties of the alloy (see black curve in the upper panel). The Al densities of states are presented in the lower panel. Here, again the electron occupancy in spin-up and spin-down are also very similar and do not contribute significantly to the total magnetic moment. The 3*s* and 3*p* electrons just contribute to the electron conductivity.

In Figure 8, the total spin polarised densities of states for up- and down-spin of the six Cu-alloys are shown. The general behavior for the six alloys is very similar to the Al-case. In

TABLE VII. Transferred electronic charges between the atoms of the  $X_2MnZ$  Heusler like systems with  $X = Cu, Ag, Au$ , and  $Z = Al, Ga, In, Si, Ge$  and  $Sn$ . In the last column we give the ratio between the electronic densities of states of opposite spins (the larger over the smaller).

$X_2MnZ$	X(Charge) (e)	Mn(Charge) (e)	Z(Charge) (e)	DOS ratio
$Cu_2MnAl$	0.297	-1.031	0.437	0.42
$Cu_2MnGa$	0.217	-1.001	0.568	0.23
$Cu_2MnIn$	0.730	-1.216	-0.245	0.45
$Cu_2MnSi$	0.468	-1.029	0.095	0.62
$Cu_2MnGe$	0.132	-1.068	0.804	0.01
$Cu_2MnSn$	0.764	-1.015	-0.514	0.04
$Ag_2MnAl$	0.278	-1.042	0.486	0.93
$Ag_2MnGa$	0.295	-1.142	0.553	0.85
$Ag_2MnIn$	0.617	-1.235	0.000	0.33
$Ag_2MnSi$	0.367	-1.105	0.372	0.50
$Ag_2MnGe$	0.177	-1.063	0.709	0.22
$Ag_2MnSn$	0.585	-1.013	-0.157	0.53
$Au_2MnAl$	-0.0885	-0.949	1.126	0.67
$Au_2MnGa$	-0.1237	-1.026	1.272	0.20
$Au_2MnIn$	0.032	-1.092	1.029	0.22
$Au_2MnSi$	0.011	-1.041	1.018	0.45
$Au_2MnGe$	-0.103	-0.999	1.206	0.93
$Au_2MnSn$	0.284	-1.019	0.452	0.21

the systems with  $Z$  belonging to the IIIA column ( $Al, Ga$ , and  $In$ ) a weak hybridization with the  $Cu$  states, in the deep energy values, is observed, whereas in the IVA column elements of the periodic table, the  $s$ -band are so deep in energy that do not hybridize much with the rest of the electrons.

In Table ??, we present the results for the electron distribution in the three atomic components of the noble metal series. In the second to fourth columns, we give the difference between the total number of electron of each element in the atomic case minus that number when they are alloyed. As in previous cases,  $Mn$  transfers about one electron to the other components. In the  $Cu$  series there is a modest charge transfer to  $Cu$ , except for the  $In$  and  $Sn$  alloys. In those two alloys,  $In$  and  $Sn$  transfer also electrons to the  $Cu$  atoms. In

TABLE VIII. Magnetic moment per unit cell ( $\mu_B$ ), and the magnetic moments of each of the elements in the  $X_2MnZ$  Heusler alloys. The X element is a noble metal, and  $Z = Al, Ga, In, Si, Ge,$  and  $Sn$ . The values in brackets are experimental data.

$X_2MnZ$	Mag. Mom. per unit cell ( $\mu_B$ )	$X(\mu_B)$	$Mn(\mu_B)$	$Z(\mu_B)$
$Cu_2MnAl$	3.64( <b>3.80</b> )[? ]	-0.031	3.803	-0.103
$Cu_2MnGa$	3.70	-0.020	3.933	-0.200
$Cu_2MnIn$	3.79( <b>4.00</b> )[? ]	-0.080	4.050	-0.094
$Cu_2MnSi$	3.96	-0.020	3.899	0.095
$Cu_2MnGe$	4.14	0.046	4.058	-0.010
$Cu_2MnSn$	4.12 ( <b>4.11</b> )[? ]	-0.032	4.097	0.088
$Ag_2MnAl$	3.82	-0.058	4.084	-0.148
$Ag_2MnGa$	4.19	0.011	4.302	-0.129
$Ag_2MnIn$	4.22	-0.021	4.393	-0.132
$Ag_2MnSi$	4.26	-0.009	4.259	0.014
$Ag_2MnGe$	4.26	0.033	4.271	-0.081
$Ag_2MnSn$	4.24	-0.056	4.332	0.024
$Au_2MnAl$	3.82	-0.013	4.043	-0.200
$Au_2MnGa$	4.05	0.029	4.210	-0.222
$Au_2MnIn$	4.09	0.004	4.324	-0.237
$Au_2MnSi$	4.14	0.024	4.206	-0.113
$Au_2MnGe$	4.23	0.051	4.253	-0.126
$Au_2MnSn$	4.22	-0.016	4.349	-0.102

the rest of the Cu systems the Z component also receive charge. In the Ag alloys, we have a similar behavior, except in  $Ag_2MnSn$ , in which Sn donates 0.16 electrons. In the case of the Au systems, most of the charge transfer goes to the Z element, Al, Ga, In, Si, and Ge receive more than 1 electron and Sn 0.452 electrons. Au donates in the cases of  $Z= Al, Ga,$  and  $Ge$  and receives electrons for  $Z=In, Si,$  and  $Sn$ . In the last column we observe that the ratio between the value for the total spin polarized electronic densities of states, spin-up and spin-down is small (0.01) only in the case of  $Cu_2MnGe$ .

The magnetic moments per unit cell and in each one of the components are presented in Table ???. There are experimental data available only for the Cu systems with  $Z=Al, In,$  and  $Sn$ . The values are given in brackets and bold face. The difference between the calculated

TABLE IX. Structural properties of the  $X_2MnZ$  Heusler alloys. The X element is a noble metal, and  $Z = Al, Ga, In, Si, Ge,$  and  $Sn$ . The crystalline structure are cubic or tetragonal, when both structures are close in energy their characteristics are given. The lattice constant is given in ( $\text{\AA}$ ), and the binding energy in eV/atom. The values in brackets are experimental data.

$X_2MnZ$	Lattice Parameter ( $\text{\AA}$ )	Binding Energy (eV/Atom)
$Cu_2MnAl$	6.042( <b>5.950</b> ) [? ]	-3.527
$Cu_2MnGa$	6.126	-3.224
$Cu_2MnIn$	6.318( <b>6.200</b> ) [? ]	-2.990
$Cu_2MnSi$	5.993	-3.574
$Cu_2MnGe$	6.114	-3.437
$Cu_2MnSn$	6.342( <b>6.170</b> ) [? ]	-3.189
$Ag_2MnAl$	6.462	-2.898
$Ag_2MnGa$	6.570	-2.657
$Ag_2MnIn$	6.746	-2.501
$Ag_2MnSi$	6.478	-2.875
$Ag_2MnGe$	6.571	-2.826
$Ag_2MnSn$	6.798	-2.672
$Au_2MnAl$	6.462	-3.371
	$2 \times 6.482, 6.453$	-3.369
$Au_2MnGa$	6.560	-3.059
	$2 \times 6.743, 6.266$	-3.054
$Au_2MnIn$	6.726	-2.888
$Au_2MnSi$	$2 \times 6.691, 6.155$	-3.253
	6.498	-3.250
$Au_2MnGe$	6.594	-3.211
	$2 \times 6.781, 6.269$	-3.208
$Au_2MnSn$	6.831	-3.026
	$2 \times 7.010, 6.448$	-3.024

and measured values is 4.4 and 5.5 percent smaller for  $Z=Al$  and  $In$ , respectively and 0.2 percent larger for  $Z=Sn$ . One observes also that the major contribution to the magnetic moment comes from the Mn atoms and is around  $4\mu_B$ : Furthermore, the coupling between Mn and the other atoms is antiferromagnetic for most of the cases.

Finally, in Table ?? we present the results for the crystalline structure, lattice parameters, and the cohesive energy. The ground state crystalline structure in all the systems is cubic,



but in the Au series, the tetragonal structures are very close in energy. One can observe that the lattice parameter, given in the second column, the measured values in Au systems ( $Z=Al, In, \text{ and } Sn$ ), are in very fair agreement with our calculation.

## CONCLUSIONS

Here, we presented a systematic study of the  $X_2MnZ$  Heusler alloys with  $X$  being a  $3d$ ,  $4d$ ,  $Pt$ , and noble metals  $Cu, Ag, \text{ and } Au$ . The  $Y$  elements studied were the IIIA ( $Al, Si, \text{ and } Ga$ ) and IVA ( $Ge, In, \text{ and } Sn$ ) column elements. The spin polarized local electronic densities of states were calculated and the energy minimized as a function of electron occupation in the spin-up and spin-down sub-bands and cell geometry and size. The calculations were performed within the density functional theory using the SIESTA code. In general, one observes an important charge transfer from the  $Mn$  atoms to the other components. The half metallic behavior was observed in the  $Fe_2MnSi, Co_2MnSi, \text{ and } Ru_2MnGe$  alloys. A less pronounced gap in the spin down density of states is obtained in  $Co_2MnGe, Ru_2MnSi, Ru_2MnSn, Cu_2MnGe, \text{ and } Cu_2MnSn$ . The highest magnetic moment per unit cell ( $7.86 \mu_B$ ) is obtained in the  $Fe_2MnIn$  and the smallest ( $2.15\mu_B$ ) in  $Ru_2MnAl$ . With a unit of 16 atoms we obtain, for all the systems, as the ground state the phase in which all the  $Mn$  atoms couple ferromagnetically. However, it has been observed that in  $Pd_2MnAl, Pd_2MnIn, Pt_2MnAl, \text{ and } Pt_2MnGa$ , they couple antiferromagnetically giving rise to complex antiferromagnetic phases. A theoretical study with a larger atomic cell is necessary to explore those phases. The crystalline structure and the lattice parameter agree well with the experimentally reported systems.

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Figure.1 The  $X_2MnZ$  unit cell used in the electronic structure calculation. The Mn atoms are denoted by light brown circles, the X elements (Co, Fe, Ni, Cu, Ru, Rh, Pd, Ag, Pt, Au) by pink circles and the Z atom (Al, Si, Ga, Ge, In, and Sn) by blue circles.

Figure 2, Positions that occupy the chemical elements X, Mn, and Z, in the Periodic Table, constituting the Heusler alloys studied here. The X are highlighted in yellow and Z elements in green and brown. Figure 3. Electronic spin polarized local density of states (SPLDS) for the  $Co_2MnAl$  alloy. In the upper panel we show the total spin polarized densities of states (black line) and the  $3d$  Cobalt contribution (purple line). In the central panel we present the electronic structures at the Mn atoms; the  $3d$  (in blue), and the  $4s$  (in red). In the bottom panel, we show the Al electronic structure; the  $3p$  (in green) and the  $3s$  (in light brown). Notice that the lower panel has a different vertical scale.

Figure 4. Total electronic spin polarized local density of states for the six Co systems  $Z=Al, Si, Ga, Ge, In,$  and Sn.

Figure 5. Electronic spin polarized local density of states (SPLDS) for the  $Pd_2MnAl$  alloy. In the upper panel we show the total spin polarized densities of states (black line) and the  $3d$  Palladium contribution (purple line). In the central panel we present the electronic structures at the Mn atoms; the  $3d$  (in blue), and the  $4s$  (in red). In the bottom panel, we show the Al electronic structure; the  $3p$  (in green) and the  $3s$  (in light brown). Notice that the lower panel has a different vertical scale.

Figure 6. Total electronic spin polarized local density of states for the six Pd systems  $Z=Al, Si, Ga, Ge, In,$  and Sn.

Figure 7, Electronic spin polarized local density of states (SPLDS) for the  $Cu_2MnAl$  alloy. In the upper panel we show the total spin polarized densities of states (black line) and the  $3d$  Copper contribution (purple line). In the central panel we present the electronic structures at the Mn atoms; the  $3d$  (in blue), and the  $4s$  (in red). In the bottom panel, we show the Al electronic structure; the  $3p$  (in green) and the  $3s$  (in light brown). Notice that the lower panel has a different vertical scale.

Figure 8. Total electronic spin polarized total local density of states for the six Cu systems  $Z=Al, Si, Ga, Ge, In,$  and Sn.