

On the interaction of infrared radiation and nanocellular polymers: first experimental determination of the extinction coefficient

Victoria Bernardo^{*1}, Judith Martin-de Leon², Javier Pinto², Ulrich Schade³, Miguel Angel Rodriguez-Perez²

1. CellMat Technologies S.L., Paseo de Belen nº9 A, 47011 Valladolid, Spain

2. Cellular Materials Laboratory (CellMat), Condensed Matter Physics Department, University of Valladolid, Campus Miguel Delibes, Paseo de Belén nº7, 47011 Valladolid, Spain
3. Department of Locally Sensitive and Time-Resolved Spectroscopy, Helmholtz-Zentrum Berlin.

*Corresponding author: Victoria Bernardo (v.bernardo@cellmattechnologies.com)

ABSTRACT

Among the various mechanisms playing a role in the heat transfer through nanocellular polymers, the radiation contribution remains the most unknown, since there is a lack of experimental data about how infrared light interacts with such structures. In this work, we present the first experimental measurements of the transmittance in the infrared region of nanocellular polymers. Infrared transmittance spectra of a collection of polymethylmethacrylate (PMMA)-based micro- and nanocellular polymers with a constant density and a wide range of cell sizes (from 14 nm to 20 μm) were obtained and evaluated to calculate the extinction coefficient. Results show that, as expected from theoretical considerations, a reduction of the cell size increases the amount of infrared radiation transmitted, that is, the scattering is reduced as cell size reduces. Nanocellular polymers were proved to act as Rayleigh-like scattering points, showing the transmittance both an intense wavelength and cell size dependence. As a consequence, the extinction coefficient reduces in the nanoscale. From these data, it is possible to conclude that the scattering due to the cellular structure can be neglected for very small cell sizes (smaller than 200 nm), but it must be considered for larger cell sizes. The obtained results were used to model the thermal conductivity including the radiation contribution, showing that at low relative densities and small cell sizes this heat transfer term becomes significant in nanocellular polymers.

KEYWORDS

Extinction coefficient; thermal conductivity; nanocellular polymers.

1. INTRODUCTION

Thermal insulation plays a major role in controlling the efficient use of energy in buildings since the energy used to keep a pleasant temperature in indoor spaces (more than 50% of the total energy spent in the sector [1]) could be reduced with improved insulation systems. Among the different solutions used for thermal insulation [2,3], cellular polymers have the main advantage of presenting excellent insulation properties, while being reasonably low-cost and easy to produce and install [4].

The parameter describing the thermal insulation ability of a given material is the thermal conductivity, λ . For cellular polymers, the thermal conductivity is governed by four mechanisms as shown in equation (1): conduction of heat through the solid polymer, λ_s ,

43 conduction of heat through the gas inside the cells, λ_g , convection of heat through the cells,
44 λ_c , and radiation through the structure, λ_r [4,5]:

$$\lambda = \lambda_s + \lambda_g + \lambda_r + \lambda_c \quad (1)$$

45 The radiation term accounts for over 20-30% of the total heat transfer in low-density cellular
46 polymers, and it is probably the most challenging contribution to understand and quantify [5].
47 Over the years, several equations have been proposed to model the radiation term, such as
48 those of Williams and Aldo [6] or Glicksman [7].

49 One unsolved question regarding the radiation contribution concerns the limits of the current
50 models when the cell size is very small. According to the existing equations [6,7], the radiation
51 contribution decreases as the cell size decrease. However, when the cell size is no longer much
52 larger than the infrared radiation wavelength, as many models hypothesize, the current
53 approximations might not be still valid.

54 There is previous evidence to expect a different transfer of radiation in nanocellular polymers
55 (i.e., cellular polymers characterized by nanometric cell sizes). For example, it has been
56 recently proved that nanocellular polymers can be transparent to visible light when the cell
57 size is small enough [8,9]. In particular, it is necessary that the cell size is smaller than 1/10 of
58 the light wavelength to obtain transparency. Taking into account that the wavelength of visible
59 light is centered around 500 nm (green light), nanocellular polymers with cell sizes under 50
60 nm would become transparent, as already proved by Martin-de Leon and coworkers [9]. Also,
61 at this scale, the scattering mechanism suffered by the light is Rayleigh-like, meaning that
62 there is a strong dependency with the wavelength in the scattering process. As a consequence,
63 nanocellular polymers present a bluish color [10]. Then, it is evident that cell size has
64 something to say in the interaction with electromagnetic waves.

65 At room temperature, the maximum of the radiation emitted by a blackbody is centered in the
66 infrared region, so to predict the radiation contribution in nanocellular polymers, it is
67 necessary to understand how infrared radiation interacts with such structures.

68 When infrared radiation travels through a cellular structure, two main processes take place:
69 absorption and scattering. The first one is determined by the amount of solid polymer (that is,
70 the relative density) and the type of polymer, so the cell size does not affect this mechanism.
71 However, the scattering phenomenon is strongly affected by the size of the scattering points
72 (i.e., the cells). In aerogels, that present similar structural characteristics to nanocellular
73 polymers, it was proved that there is a high transmittance in the infrared region in frequencies
74 where the solid material does not have an absorption band [11], meaning that the scattering is
75 reduced for nanometric cell sizes. In the work of Hrubesh et al. [12], the radiation contribution
76 in aerogels was associated only with the absorption of the solid material, assuming that there
77 is not scattering in the cells due to their reduced cell size [13].

78 Up to date, and as far as the authors know, there is not any experimental evidence of this
79 effect in nanocellular polymers. In the 1D model proposed by Ferkl et al. [14], the radiation
80 contribution decreased with the reduction in cell size in nanocellular polymers. In the 3D
81 model of Wang and coworkers [15], the radiation contribution in nanocellular polymers was
82 shown to be significant when density reduces, although the calculation of the radiative
83 contribution is purely theoretical and was not experimentally validated.

84 In this work, we have studied for the first time the interaction of infrared light with
 85 nanocellular polymers to investigate the contribution of radiation to the total thermal
 86 conductivity. The extinction coefficient of a collection of nanocellular polymers with variable
 87 cell size has been measured. According to the Rosseland equation [16–18], the radiation term
 88 can be calculated as expressed in equation (2):

$$\lambda_r = \frac{16n^2\sigma T^3}{3K_{e,R}} \quad (2)$$

89 Where σ is the Stefan-Boltzman constant, T is the temperature, n is the refractive index, and
 90 $K_{e,R}$ is the Rosseland extinction coefficient. Thus, this parameter is essential to quantify the
 91 influence of the cell size in the radiation contribution. Our results show that, as expected from
 92 a theoretical perspective, the decrease of the cell size causes a drastic decrease of the
 93 extinction coefficient, and thus an increase of the radiation contribution.

94

95 2. EXPERIMENTAL

96

97 2.1. Materials

98 **Table 1** summarizes the main structural characteristics of the collection of
 99 polymethylmethacrylate (PMMA)-based micro- and nanocellular polymers used in this work.
 100 All the materials were produced using a two-step gas dissolution foaming process and have
 101 similar relative densities. Details about the production process used to produce each sample
 102 can be found in the corresponding reference. The PMMA grade used as polymer matrix was, in
 103 all cases, ALTUGLAS® V 825T kindly supplied by Arkema. In the samples with MAM
 104 (poly(methyl methacrylate-poly(butyl acrylate)-poly(methyl methacrylate)), the amount of
 105 block copolymer was always low (smaller than 2 wt%) [19] and the infrared spectrum was not
 106 modified due to the addition of MAM. The average 3D cell size of the samples was measured
 107 by SEM plus image analysis [20], while the relative density was determined by the Archimedes'
 108 principle after polishing the samples to remove the solid skin.

109 **Table 1.** Characteristics of the PMMA-based micro- and nanocellular polymers used in this
 110 work.

Sample #	Cell size	Relative Density	Material	Reference
1	14 nm	0.42	PMMA	[9]
2	24 nm	0.43	PMMA	[9]
3	120 nm	0.37	PMMA/MAM	[19]
4	225 nm	0.42	PMMA	[21]
5	700 nm	0.41	PMMA/MAM	[19]
6	800 nm	0.43	PMMA/MAM	[19]
7	2.5 μm	0.43	PMMA	[22]
8	3.5 μm	0.43	PMMA	[23]
9	16 μm	0.43	PMMA	[24]
10	20 μm	0.41	PMMA	[24]

111

112 Thin plane-parallel sheets were obtained from the foamed samples of **Table 1** with a precision
 113 cutting machine (Model 1000 IsoMet). Homogenous and uniform samples of various

114 thicknesses for each material in the range from 30 to 130 μm were obtained for the
 115 transmittance measurements. **The thickness of the samples is always higher than the**
 116 **wavelength of the infrared radiation used.** All the samples were produced using the same
 117 method, so the surface quality of all the samples was comparable. The thickness of the
 118 samples was measured using a DMA7 dynamic mechanical analyzer (PerkinElmer) with an
 119 accuracy of 0.5 microns.

120

121 **2.2. Transmittance measurements**

122 The infrared transmittance of the samples was measured by using an IR microscope Nicolet
 123 iN10 MX (Thermo Scientific) located at the synchrotron BESSY II (Berlin). Infrared spectra were
 124 collected in a wavelength range from 2.5 to 12.5 μm (800 to 4000 cm^{-1} in wavenumber). 128
 125 scans were acquired. Before each measurement, the background signal was corrected. The
 126 spot size was $50 \times 50 \mu\text{m}^2$, and every sample was measured at the least at two regions to assure
 127 the reproducibility of the measurements.

128

129 **2.3. Calculation of the extinction coefficient**

130 For thin homogeneous samples, the spectral extinction coefficient $K_{e,\lambda}$ can be obtained with
 131 Beer's law (equation (3)), where T_λ is the transmittance at a given wavelength λ , and L is the
 132 sample thickness. By means of a linear regression of the infrared transmittance spectra at
 133 various thicknesses, $K_{e,\lambda}$ can be calculated.

$$K_{e,\lambda} = -\frac{\ln(T_\lambda)}{L} \quad (3)$$

134 The Rosseland mean extinction coefficient, $K_{e,R}$, can be calculated from the spectral
 135 coefficient according to equation (4):

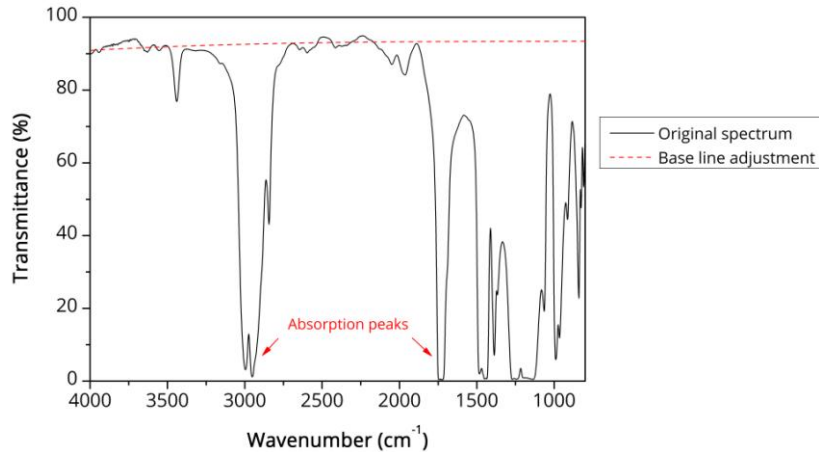
$$\frac{1}{K_{e,R}} = \frac{\int_0^\infty \frac{1}{K_{e,\lambda}} \frac{\partial e_{b,\lambda}}{\partial T} d\lambda}{\int_0^\infty \frac{\partial e_{b,\lambda}}{\partial T} d\lambda} \quad (4)$$

136 Where $e_{b,\lambda}$ is the spectral black body emissive power and T is the temperature.

137 Due to the high density of the nanocellular samples used in this work, the absorption bands
 138 corresponding to the solid PMMA in the transmittance spectra are saturated even at very low
 139 thicknesses (see an example in **Figure 1**). Therefore, it is not possible, with these samples, to
 140 perform the previous analysis and give a global Rosseland extinction coefficient. However, as it
 141 was explained in the introduction, the interaction of the infrared light with cellular polymers
 142 can be divided into two terms: first, the absorption part (that is, the amount of radiation
 143 absorbed by the polymer, that depends on the density and the polymer nature), and the
 144 scattering contribution, which is mainly due to the presence of the cellular structure. Then, it is
 145 possible to define $K_{e,R}$ as a sum of these two contributions, as already proposed by Glicksman
 146 [7]: an absorption Rosseland term, $K_{e,R,abs}$, and a scattering term $K_{e,R,scatt}$ (equation (5)):

$$K_{e,R} = K_{e,R,abs} + K_{e,R,scatt} \quad (5)$$

147 The transmittance curve can be then divided into these two contributions: the absorption
148 bands and the scattering baseline. **Figure 1** also shows the scattering baseline once the
149 absorption bands are removed. This baseline was calculated by selecting a collection of points
150 out of the absorption bands and then performing a fit of those points. In the region of
151 wavenumbers from 1900 cm^{-1} to 800 cm^{-1} , which does not present a clear baseline, we have
152 extrapolated the fit line calculated in the other region.



153

154 **Figure 1.** Example of a transmittance spectra as a function of the wavenumber of a sample of
155 thickness $50 \pm 10\ \mu\text{m}$ and cell size of $14\ \text{nm}$ (sample 1 in **Table 1**); and baseline of the spectrum
156 after removing the absorption bands of the polymer (red dashed line).

157 In this work, we have calculated the scattering extinction coefficient, $K_{e,R,scatt}$. To do so, we
158 have performed a fit of the base line of the transmittance spectra (as shown in the example of
159 **Figure 1**) and then we have performed the analysis of equations (3) and (4). As
160 aforementioned, $K_{e,R,scatt}$ accounts for the contribution of the cellular structure, so it is a
161 good parameter to evaluate the influence of the cell size in the change from the micro- to the
162 nanoscale. Regarding the absorption part, we could not get direct information from the
163 experimental data, as commented previously, so the extinction coefficient we present in this
164 paper is only the scattering part. In the last section of this work, where the total conductivity is
165 predicted, this absorption term is calculated theoretically (see **section 3.3**).

166 One issue that might affect this calculation is the surface of the samples. The samples of this
167 paper were prepared using a precision cutting machine, and as a result of this process they
168 present a texture of grooves that can act as scattering surfaces, perturbing the result (i.e.
169 increasing the scattering of the cellular materials) (see **Supplementary Information, Figure S1**
170 for images of the surface of the samples). Therefore, the scattering process in these samples
171 has two contributions: the scattering in the cellular structure itself, but also the scattering in
172 the surface. To factor out this second effect, that it is not of interest for our study, solid
173 samples of different thicknesses were cut using the same method, and the scattering
174 extinction coefficient was calculated in the same way as with the cellular materials. For the
175 solids, as there are not any other scattering points, all the scattering is associated to the
176 structure of the surface. This analysis leads to a value of the scattering coefficient of these
177 solid samples, that is associated to its surface, of 33.05 cm^{-1} . Then, this value of the scattering
178 in the surface was subtracted to calculate the $K_{e,R,scatt}$ of the cellular materials to obtain the
179 contribution of the cellular structure (see **Supplementary Information, Figure S2**, for the
180 results before and after this correction). In this way, we assure that the scattering extinction
181 coefficient presented from now on is solely due to the effect of the cellular structure.

182

183 2.4. Scattering mechanism: theoretical background

184 As aforementioned, infrared radiation would encounter scattering phenomena when traveling
185 through a cellular polymer due to the presence of the cells. The scattering process is strongly
186 affected by the size of the scattering points. When the radii of the points are clearly smaller
187 than the wavelength of the radiation, the expected scattering behavior from a theoretical
188 perspective would be Rayleigh-like. One feature that Rayleigh scatters show is a strong
189 dependence on the wavelength. The transmittance, T , depends on the wavelength, λ , as
190 shown in equation (6), according to Rayleigh law [25], where A and B are constants.

$$T = Ae^{-\frac{B}{\lambda^4}} \quad (6)$$

191 Another feature related to the Rayleigh scattering behavior is that the amount of radiation
192 scattered is reduced as the size of the scattering point decreases, that is, the efficiency of the
193 scattering reduces as size does. The scattering efficiency for a Rayleigh-like scattering behavior
194 can be computed according to equation (7) [26]:

$$\sigma_{Ray} = \frac{8\pi}{3} \left(\frac{2\pi n_m}{\lambda} \right)^4 r^6 \left(\frac{m^2 - 1}{m^2 + 2} \right)^2 \quad (7)$$

195 Where n_m is the refractive index of the surrounding medium, r is the radius of the scattering
196 particle, and $m = n_p/n_m$ is the ratio of refractive indexes (n_p is the refractive index of the
197 scattering particle). The scattering efficiency of one scattering particle, Q , is given by (9):

$$Q = \frac{\sigma_{Ray}}{\pi r^2} \quad (8)$$

198 In a general situation, the theory to describe the scattering process is the Mie theory. At low
199 particle radii, this theory matches with the Rayleigh predictions. At large sizes of the scattering
200 points, the Mie theory presents a limit value of the scattering efficiency equal to 2 [26].

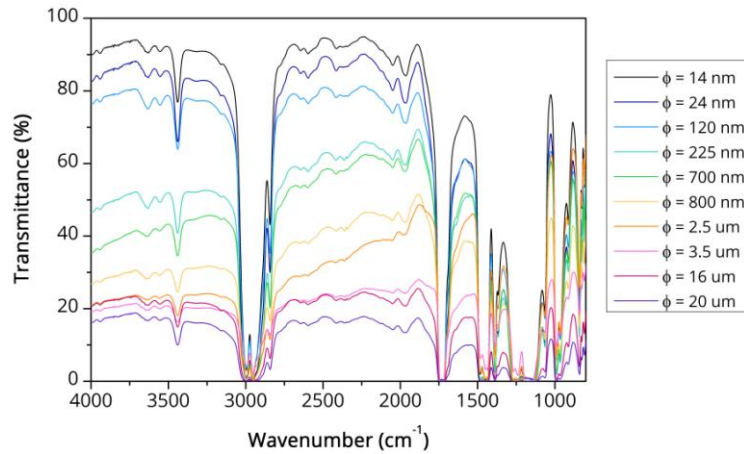
201

202 3. RESULTS AND DISCUSSION

203

204 3.1. Transmittance in the infrared region

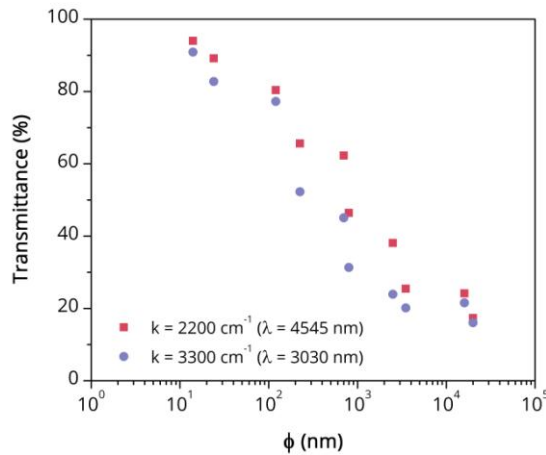
205 **Figure 2** shows the transmittance in the infrared region of the samples of **Table 1** with
206 different cell sizes. The spectra presented in **Figure 2** correspond to samples of thickness
207 around $50 \pm 10 \mu\text{m}$ for the sake of comparison. Note that all the samples present a similar
208 relative density (around 0.4, **Table 1**). In the regions of the spectra presenting absorption
209 bands, all the samples show full saturated bands, so there is no information about the cellular
210 structure. However, out of these bands (for instance in the range of wavenumbers between
211 2750 and 2000 cm^{-1}), there is a noticeable effect of the cell size. Qualitatively, we observe that
212 the reduction of the cell size from $20 \mu\text{m}$ to 14 nm drastically increases the transmittance, that
213 is, the scattering of the infrared light is reduced as the cell size decreases, reaching values of
214 transmittance close to 100% when the cell size is very small.



215

216 **Figure 2.** Transmittance as a function of the wavenumber for samples of thickness $50 \pm 10 \mu\text{m}$
 217 with variable cell size (from 14 nm to 20 microns).

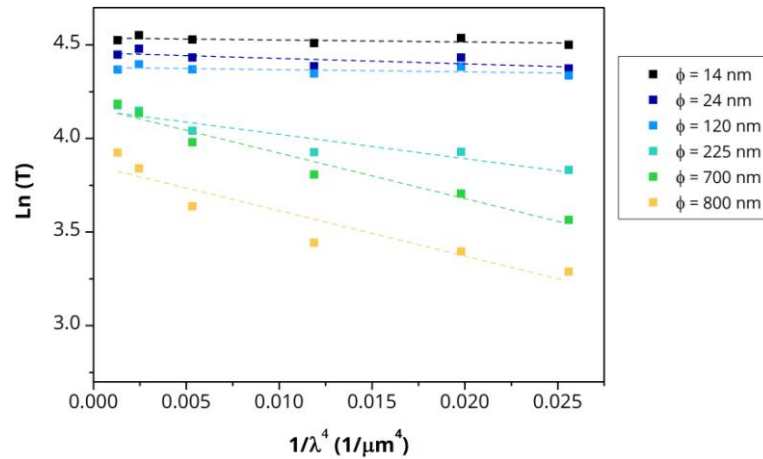
218 **Figure 3** shows the transmittance of the samples of **Figure 2** as a function of the cell size for
 219 two fixed wavenumbers: 3300 cm^{-1} (3030 nm in wavelength) and 2200 cm^{-1} (4545 nm in
 220 wavelength). These two values were selected at regions out of the absorption bands of the
 221 PMMA to see the effect of the cellular structure. As already mentioned, it is observed that a
 222 decrease in cell size increases the transmittance dramatically.



223

224 **Figure 3.** Transmittance as a function of the cell size for samples of thickness $50 \pm 10 \mu\text{m}$ at
 225 two fixed wavelengths.

226 In this experiment, the infrared light used ranged from $2.5 \mu\text{m}$ to $12.5 \mu\text{m}$. For samples with
 227 cell sizes smaller than $1/10$ of these values, the expected scattering behavior from a
 228 theoretical perspective would be Rayleigh-like. As already mentioned, Rayleigh scatters show a
 229 strong dependence on the wavelength (equation (6)). To investigate this effect, the logarithm
 230 of the transmittance at specific wavelengths is plotted as a function of the parameter $1/\lambda^4$ for
 231 the nanocellular samples (cell size smaller than 1 micron) in **Figure 4**. The wavelengths were
 232 selected in regions out of the absorption bands of the PMMA. The dependence of equation (6)
 233 is confirmed in this plot, because straight lines are obtained.



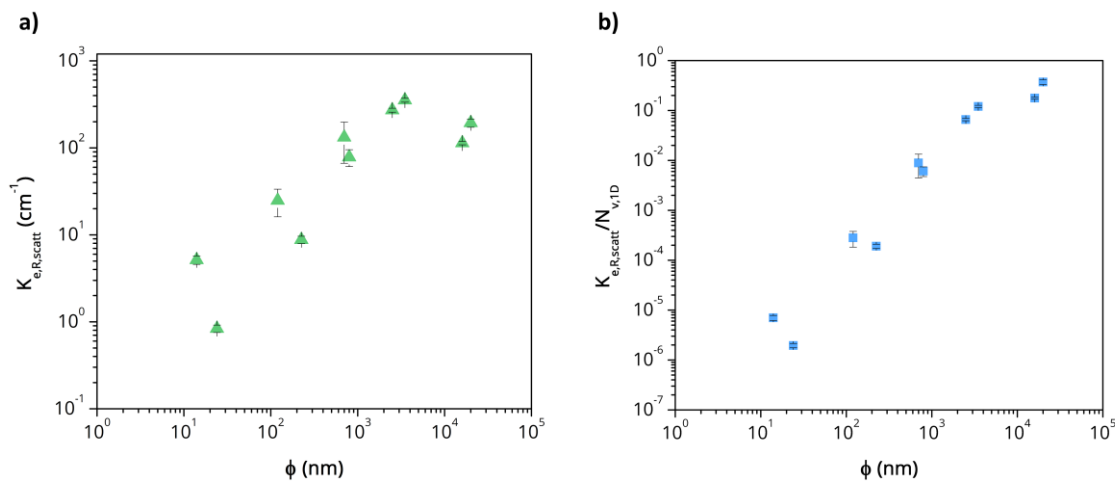
234

235 **Figure 4.** Logarithm of the transmittance at certain wavelengths (out of the absorption region)
 236 as a function of $1/\lambda^4$ to observe relation (6), for samples of thickness $50 \pm 10 \mu\text{m}$ with variable
 237 cell size (from 14 nm to 800 nm).

238

239 3.2. Scattering extinction coefficient

240 Using the transmittance spectra of samples with different thicknesses and performing a
 241 baseline fit, as explained previously, it is possible to calculate the extinction coefficient
 242 associated with the scattering baseline. **Figure 5.a** shows the Rosseland scattering extinction
 243 coefficient as a function of the cell size calculated as explained in section 2.3. This coefficient
 244 shows that there is a reduction of the extinction coefficient when the cell size is reduced to the
 245 nanoscale. At small cell sizes, the scattering extinction coefficients are very low (smaller than
 246 10 cm^{-1}), and these values start to increase as the cell size increases.



247

248 **Figure 5.** a) Scattering extinction coefficient as a function of the cell size, b) scattering
 249 extinction coefficient normalized by the 1D cell density (scattering efficiency of one cell) as a
 250 function of the cell size.

251 One interesting parameter to be calculated is the scattering efficiency of one single cell. The
 252 heat flux takes place along a given direction, and the number of cells along the sample
 253 thickness is playing a role. In a nanocellular polymer, the number of cells along the sample
 254 thickness is much higher than in a microcellular polymer for the same density. That is, the
 255 number of scattering points is higher in nanocellular polymers. Then, one way to normalize the

256 $K_{e,R,scatt}$ is considering the linear cell density, that is, the number of cells per unit of length.
 257 From the standard definition of the cell density [27], the linear cell density $N_{v,1D}$ can be
 258 calculated theoretically as shown in equation (9), where ϕ is the cell size and ρ_r is the relative
 259 density. Note that the units of $N_{v,1D}$ are cm^{-1} , so when $K_{e,R,scatt}$ is normalized by this value the
 260 result is unitless.

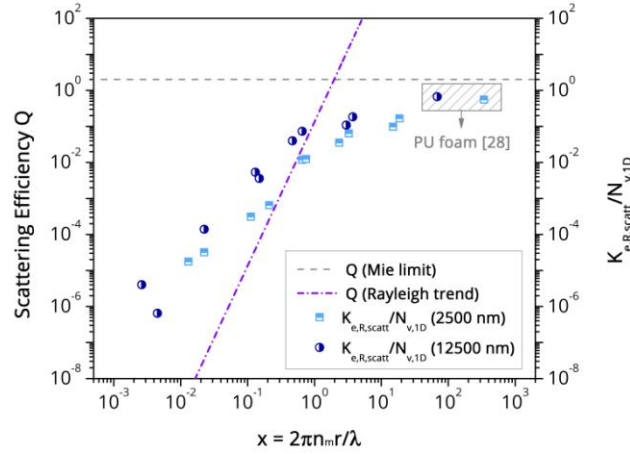
$$N_{v,1D} = \left[\frac{6}{\pi\phi^3} (1 - \rho_r) \right]^{1/3} \quad (9)$$

261

262 **Figure 5.b** shows the result of dividing $K_{e,R,scatt}$ by the linear cell density calculated as defined
 263 in (9). Note that the parameter $K_{e,R,scatt}/N_{v,1D}$ shows a much clear dependence with the cell
 264 size because of the fact that the cell density differences are corrected. Increasing the cell size
 265 causes an increment of the scattering efficiency, or in other words, the reduction of the cell
 266 size to the nanoscale reduces the efficiency of the cells as scattering centers. **Note that the cell**
 267 **density correction also corrects the small density variations among samples (see Table 1), since**
 268 **the cell density is calculated based both on the cell size and the relative density. For this**
 269 **reason, the parameter $K_{e,R,scatt}/N_{v,1D}$ presented in Figure 5.b, that accounts for the**
 270 **scattering effect of one single cell, is a general result that could be applied to any cellular**
 271 **material with this range of cell sizes independently on the density.**

272 The trend observed in **Figure 5.b** is similar to the expected theoretical behavior for Rayleigh
 273 scatters, that is, the efficiency of the scattering reduces as size does (equation (8)). To
 274 correlate these results with the theoretical predictions, **Figure 6** shows the predicted
 275 theoretical scattering efficiency of Rayleigh particles as a function of the unitless parameter
 276 $x = 2\pi n_m r / \lambda$ (size parameter). The efficiency of the scattering increases as the particle radius
 277 increases. When the particle size is very large (that is, $x \gg 1$), the scattering behavior would
 278 move to the Mie regime, reaching a limit value of Q equal 2 [26]. In **Figure 6** we have included
 279 the values of $K_{e,R,scatt}/N_{v,1D}$, what we have called “scattering efficiency of one cell”,
 280 calculated at the two limit wavelengths, 2500 and 12500 nm. Data from a PU foam with a
 281 much higher cell size (366 μm) was also included for comparison [28]. These experimental
 282 values show a similar trend in comparison with the theoretical predictions. The scattering
 283 efficiency increases linearly as cell sizes does, reaching the Mie limit for large cell sizes. **Figure**
 284 **6** proves that nanocellular polymers present also Rayleigh scattering with regard to the cell
 285 size dependence.

286 Therefore, the results of this work prove that the transmittance in the infrared region of
 287 nanocellular polymers follows the theoretical trends in terms of wavelength and cell size
 288 dependence. The values estimated in this paper give a reasonable estimation of the behavior
 289 of infrared radiation interacting with nanocellular polymers.



290

291 **Figure 6.** Right axis: Scattering efficiency Q for a Rayleigh-like behavior and Mie limit as a
 292 function of the size parameter x . Left axis: Scattering extinction coefficient normalized by the
 293 1D cell density as a function of the size parameter x .

294

295 3.3. Modeling the radiation contribution

296 Once the extinction coefficient is calculated, it is possible to use these experimental values to
 297 predict the radiation contribution and model the thermal conductivity. Recall from equation
 298 (5) that we split the extinction coefficient into two parts. The absorption contribution, $K_{e,R,abs}$,
 299 can be modelled using the extinction coefficient of the solid polymer, K_p , and the relative
 300 density, as shown in equation (10), since the amount of radiation absorbed will be
 301 proportional to the amount of solid polymer [29]:

$$K_{e,R,abs} = K_p \rho_r \quad (10)$$

302 To quantify the scattering term, the data of **Figure 5.b** corresponding to the scattering
 303 efficiency of one cell was adjusted to a potential equation of the form:

$$K_{e,R,scatt}/N_{v,1D} = A\phi^B \quad (11)$$

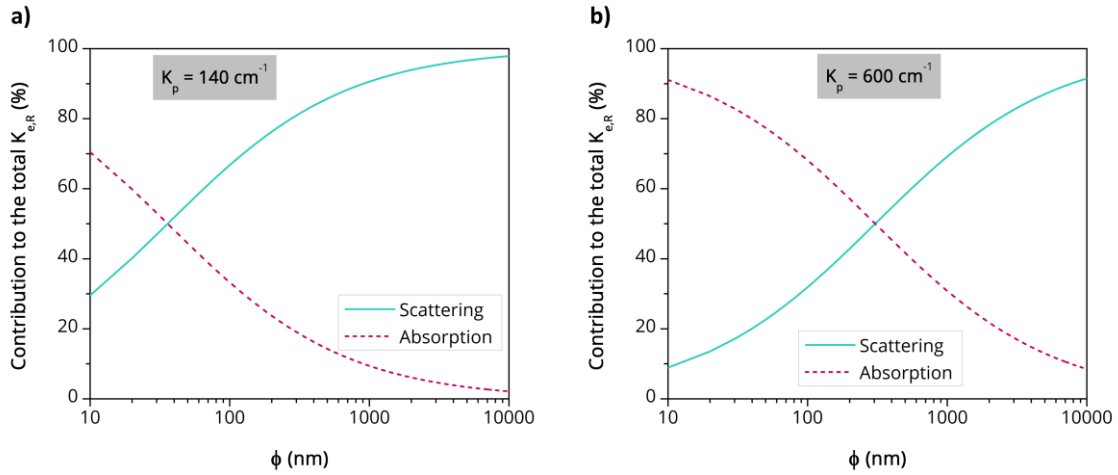
304 Where A and B are dimensionless experimental parameters. In particular, for our data, A is
 305 $5.02 \cdot 10^{-8}$ and B is 1.68. Therefore, the total extinction coefficient can be expressed as:

$$K_{e,R} = K_{e,R,abs} + K_{e,R,scatt} = K_p \rho_r + N_{v,1D} A \phi^B \quad (12)$$

306 Where $N_{v,1D}$ is calculated from the density and the cell size according to equation (9). The
 307 radiation term can be then calculated as (2) by using this semi-empirical extinction coefficient
 308 defined in (12).

309 In aerogels, it is claimed that the extinction of the infrared radiation is only due to absorption
 310 and not scattering due to the small pore size [13]. To see if this assumption can also be made
 311 in nanocellular polymers, it is interesting to calculate the relative contribution of the scattering
 312 and the absorption part to the total extinction coefficient for a low density material (relative
 313 density 0.05) (**Figure 7**). One key parameter in this study is the extinction coefficient of the
 314 solid polymer, K_p . Unfortunately, and as far as the authors know, there are not data about this
 315 parameter for PMMA in this wavelength range. To do some first estimations, two extreme
 316 values have been selected for the graphs of **Figure 7**. First, a low value of K_p 140 cm^{-1}
 317 measured for low-density polyethylene [17], and then a high value of 600 cm^{-1} calculated for

318 polyurethane foams [5]. It is observed that, in both cases, the contribution of the scattering is
 319 reduced as the cell size reduces. For very small cell sizes, it might be a good approximation to
 320 neglect the scattering part, especially if the absorption of the solid is high (**Figure 7.b**).
 321 However, as cell size increases, the contribution of the scattering becomes more relevant. For
 322 cell sizes of 200-300 nm, that are typical values reported in many works in the literature, the
 323 contribution can be higher than 75% (for $K_p = 140 \text{ cm}^{-1}$, **Figure 7.a**) or at least higher than 40%
 324 (for $K_p = 600 \text{ cm}^{-1}$, **Figure 7.b**). Thus, at these cell sizes, we cannot neglect this contribution,
 325 and to model properly the thermal conductivity both aspects, absorption and scattering, must
 326 be included in the equations.



327

328 **Figure 7.** Contribution to the total extinction coefficient of the scattering and the absorption
 329 parts for a low relative density material (relative density = 0.05), for two different extinction
 330 coefficients of the solid: a) $K_p = 140 \text{ cm}^{-1}$ and b) $K_p = 600 \text{ cm}^{-1}$.

331 Regarding the other contributions to calculate the thermal conductivity in equation (1), the
 332 convection term is known to be negligible for cell sizes smaller than 2 mm [5,24,30]. For
 333 nanocellular polymers, the conduction terms, λ_s and λ_g , are usually described by the following
 334 equations [31–33]:

$$\lambda_s = g\lambda'_s\rho_r \quad (13)$$

$$\lambda_g = \frac{\lambda'_{g0}}{1 + \frac{2\beta l_g}{\phi}} (1 - \rho_r) \quad (14)$$

335 Where λ'_s is the thermal conductivity of the solid polymer, and g is a an efficiency-structural
 336 factor proposed by Glicksman [5]. This g factor ranges from 1/3 to 1. For medium-high density
 337 materials it usually takes values close to 1 [34,35], while this factor is 2/3 for closed-cell low
 338 density materials and it reaches the minimum value of 1/3 for materials with a high fraction of
 339 the solid phase in the struts (i.e., open cell structures) [5]. Regarding the gas phase, λ'_{g0} is the
 340 thermal conductivity of the gas in the cells (26 mW/mK for air at atmospheric pressure and
 341 room temperature), β is a factor correlating the energy transfer between gas molecules and
 342 the structure [33] (1.64 for air [32]) and l_g is the mean free path of the gas molecules ($l_g \approx 70$
 343 nm for air [35,36]).

344 The thermal conductivity of nanocellular polymers was calculated using equations (1), (2), (12),
 345 (13), and (14). Thermal conductivity of PMMA is 212 mW/mK [37], and as in the previous
 346 example, two extreme extinction coefficient of the solid (K_p) have been used: 140 and 600 cm^{-1}

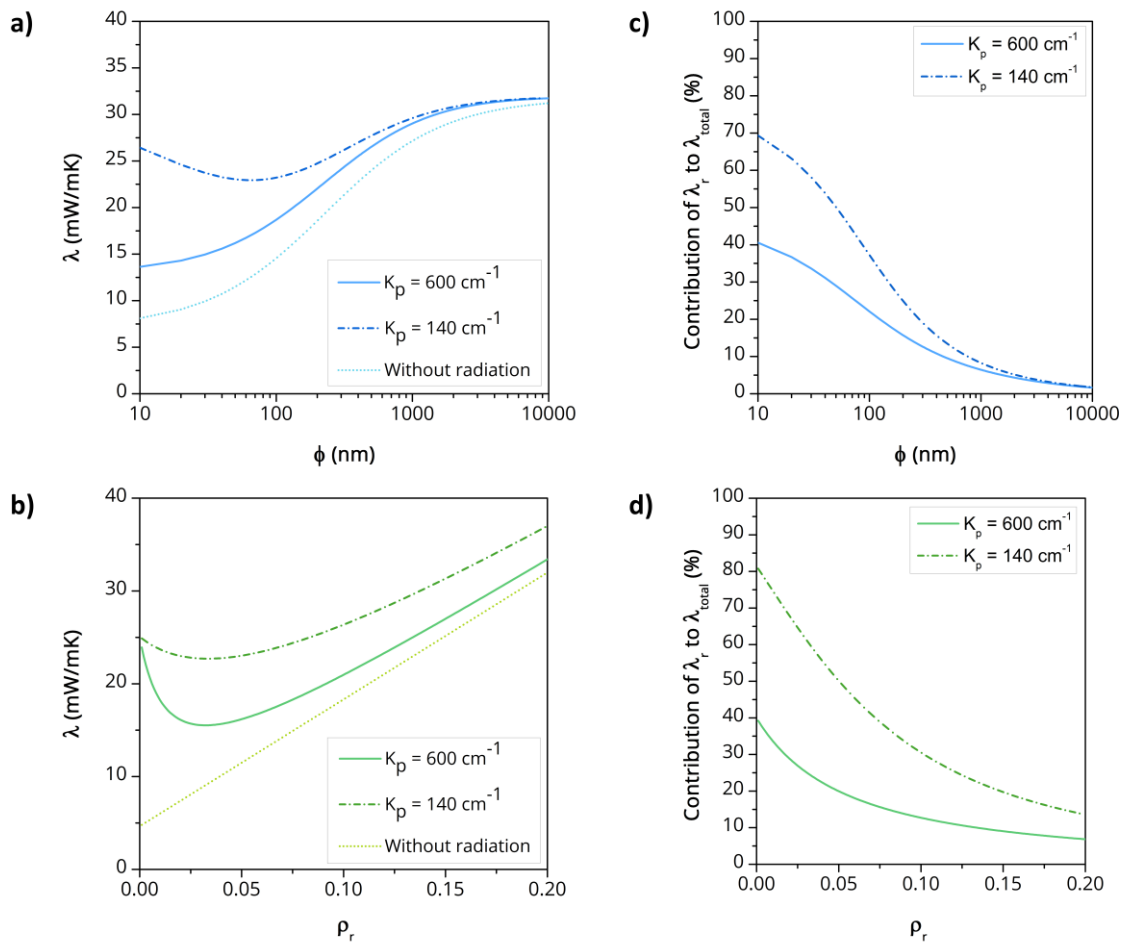
347 ¹. Regarding the structural factor g , a value of $2/3$ was selected, since the predictions were
348 mainly focused in the low-density region ($\rho_r \leq 0.2$). **Figure 8** shows the predicted thermal
349 conductivity as a function of the cell size for a fixed relative density of 0.05 (**Figure 8.a**), and as
350 a function the relative density for a fixed cell size of 50 nm (**Figure 8.b**). The predictions
351 without including the radiation term are also plotted in this graph for the sake of comparison.
352 The relative contribution of the radiation mechanism for the two values of K_p are presented in
353 **Figure 8.c** and **Figure 8.d**. Note that in these predictions, some critical assumptions have been
354 made: the cellular structure was considered monomodal (there might be an influence of the
355 cell size distribution [37]), homogeneous, 100% closed cell, etc. Thus, these predictions could
356 differ slightly from real conductivity values, but they show qualitatively the expected trends.

357 It is observed that there is a significant difference between the models presented in this work
358 and the predictions without including the radiation term. Also, the predictions are very
359 sensitive to the value of K_p . For a low K_p and relative density of 0.05, a minimum conductivity
360 appears at a cell size of around 70 nm, but this minimum is not observed with a higher K_p
361 (**Figure 8.a**). In the worse situation (lowest K_p), for a relative density of 0.05, the minimum
362 conductivity calculated is around 23 mW/mK. The contribution of the radiation term to the
363 total thermal conductivity increases as cell size decreases, ranging the maximum values from
364 40 to 70% depending on K_p (**Figure 8.c**).

365 Regarding the density dependency (**Figure 8.b**), the conductivity reduces with density up to a
366 minimum. For 50 nm, the minimum conductivity is 15.5 mW/mK for a density of 0.032 for the
367 high K_p , whereas the value is 22.7 mW/mK for a density of 0.034 for $K_p = 140 \text{ cm}^{-1}$. The
368 contribution of the radiation term to the total thermal conductivity increases as density
369 decreases, ranging the maximum values from 40 to 80% depending on K_p (**Figure 8.d**).

370

371



373

374 **Figure 8.** a) Theoretical thermal conductivity of nanocellular polymers based on PMMA
 375 according to the model including the radiation term and without it and for two different
 376 extinction coefficients of the solid a) as a function of the cell size for relative density of 0.05,
 377 and b) as a function of the relative density for cell size of 50 nm. Contribution of the radiation
 378 mechanism to the total thermal conductivity: b) for a relative density of 0.05 as a function of
 379 the cell size and d) for a cell size of 50 nm as a function of the relative density.

380

381 The great potential of nanocellular polymers as excellent thermal insulators has been pointed
 382 out in various works [35,38,39]. The results of this work do not contradict these claims. Our
 383 predictions show that even though the radiation term is included, it would be possible to
 384 obtain low thermal conductivities with nanocellular polymers, for instance, for a relative
 385 density of 0.05 and a cell size of 70 nm the predicted thermal conductivity would be as low as
 386 23 mW/mK. However, the most optimistic predictions of the early years of the field of
 387 nanocellular polymers (reductions of 2.5 times [35,40]) seem now almost impossible to reach
 388 using homogeneous polymeric systems once the radiation contribution has been understood
 389 and experimentally characterized.

390 Further work would be needed to understand how to reduce this contribution in nanocellular
 391 polymers. The use of bimodal structures, with micron-sized pores able to scatter radiation and
 392 also helping to reduce the density, might be beneficial in this sense [37]. Also, IR nanometric
 393 opacifiers could be included to reduce the K_p of the solid matrix and decrease the radiation

394 contribution[41]. The results of this work would pave the way for future developments to
395 further exploit the exciting properties of nanocellular polymers.

396

397 **4. CONCLUSIONS**

398 Transmittance spectra in the infrared region of a collection of micro- and nanocellular
399 polymers were measured to investigate the interaction of nanocellular structures with infrared
400 radiation. A collection of PMMA-based micro- and nanocellular polymers with a constant
401 density and a wide range of cell sizes (from 14 nm to 20 μm) were used for these
402 measurements. Results show that there is strong increase of the transmittance out of the
403 absorption bands as the cell size reduces to the nanoscale. The dependencies of the
404 transmittance with both the wavelength and the cell size prove that nanometric cells act as
405 Rayleigh scattering particles.

406 By performing a fit of the scattering baseline, it is possible to use the transmittance spectra to
407 calculate the scattering extinction coefficient of nanocellular polymers. The results of this
408 paper show that the extinction coefficient reduces as the cell size decreases. The values were
409 corrected by the linear cell density to calculate the scattering efficiency of one cell as a
410 function of the cell size.

411 From these values, it is possible to calculate the total extinction coefficient and to weight the
412 relevance of the scattering part. We conclude that the scattering can be neglected for very low
413 cell sizes (smaller than 200 nm), but it must be considered for larger cell sizes. The thermal
414 conductivity of nanocellular polymers was modeled including the radiation contribution,
415 showing that at low relative densities and small cell sizes this term becomes significant in
416 nanocellular polymers.

417

418 **Acknowledgments**

419 Financial assistance from MINECO, FEDER, UE (MAT2015-69234-R), the Junta de Castile and
420 Leon (VA275P18) and Spanish Ministry of Science, Innovation and Universities (RTI2018-
421 098749-B-I00) are gratefully acknowledged. Financial assistance from EREN (Ente Regional de
422 la Energía de Castilla y León EREN_2019_L4_UVA) is gratefully acknowledged. This project has
423 received funding from the European Union's Horizon 2020 research and innovation
424 programme under grant agreement No 730872.

425

426 **REFERENCES**

- 427 [1] International Energy Agency (IEA), Technology Roadmap. Energy efficient building
428 envelopes, (2013). doi:10.1007/SpringerReference_7300.
- 429 [2] B.P. Jelle, Traditional, state-of-the-art and future thermal building insulation materials
430 and solutions - Properties, requirements and possibilities, Energy Build. 43 (2011)
431 2549–2563. doi:10.1016/j.enbuild.2011.05.015.
- 432 [3] L. Aditya, T.M.I. Mahlia, B. Rismanchi, H.M. Ng, M.H. Hasan, H.S.C. Metselaar, O.
433 Muraza, H.B. Aditya, A review on insulation materials for energy conservation in
434 buildings, Renew. Sustain. Energy Rev. 73 (2017) 1352–1365.

- 435 doi:10.1016/j.rser.2017.02.034.
- 436 [4] D. Eaves, Handbook of Polymer Foams, Rapra Technology, United Kingdom, 2004.
- 437 [5] N.C. Hilyard, A. Cunningham, Low density cellular plastics--Physical basis of behaviour,
438 Chapman and Hall, London, 1994.
- 439 [6] R.J.J. Williams, C.M. Aldao, Thermal conductivity of plastic foams, Polym. Eng. Sci. 23
440 (1983) 293–298. doi:10.1002/pen.760230602.
- 441 [7] L. Glicksman, M. Schuetz, M. Sinofsky, Radiation heat transfer in foam insulation, Int. J.
442 Heat Mass Transf. 30 (1987) 187–197. doi:10.1016/0017-9310(87)90071-8.
- 443 [8] S. Perez-Tamarit, B. Notario, E. Solorzano, M.A. Rodriguez-Perez, Light transmission in
444 nanocellular polymers: are semi-transparent cellular polymers possible?, Mater. Lett.
445 210 (2017) 39–41. doi:10.1016/j.matlet.2017.08.109.
- 446 [9] J. Martín-de León, V. Bernardo, M.A. Rodríguez-Pérez, Key Production Parameters to
447 Obtain Transparent Nanocellular PMMA, Macromol. Mater. Eng. 1700343 (2017) 1–5.
448 doi:10.1002/mame.201700343.
- 449 [10] J. Martín-de León, J.L. Pura, V. Bernardo, M.Á. Rodríguez-Pérez, Transparent
450 nanocellular PMMA: Characterization and modeling of the optical properties, Polymer
451 (Guildf). 170 (2019) 16–23. doi:10.1016/j.polymer.2019.03.010.
- 452 [11] R. Baetens, B. Petter, A. Gustavsen, Aerogel insulation for building applications : A
453 state-of-the-art review, Energy Build. 43 (2011) 761–769.
454 doi:10.1016/j.enbuild.2010.12.012.
- 455 [12] L.W. Hrubesh, R.W. Pekala, Thermal properties of organic and inorganic aerogels, J.
456 Mater. Res. 9 (1994) 731–738. doi:10.1557/JMR.1994.0731.
- 457 [13] U. Heinemann, R. Caps, J. Fricke, Radiation-conduction interaction : An investigation on
458 silica aerogels, Int. J. Heat Mass Transf. 39 (1996) 2115–2130. doi:10.1016/0017-
459 9310(95)00313-4.
- 460 [14] P. Ferkl, R. Pokorný, M. Bobák, J. Kosek, Heat transfer in one-dimensional micro- and
461 nano-cellular foams, Chem. Eng. Sci. 97 (2013) 50–58. doi:10.1016/j.ces.2013.04.018.
- 462 [15] G. Wang, C. Wang, J. Zhao, G. Wang, C.B. Park, G. Zhao, Modelling of thermal transport
463 through a nanocellular polymer foam: Toward the generation of a new superinsulating
464 material, Nanoscale. 9 (2017) 5996–6009. doi:10.1039/c7nr00327g.
- 465 [16] L.R. Glicksman, Heat transfer in foams, in: N.C. Hilyard, A. Cunningham (Eds.), Low
466 Density Cell. Plast., Springer, Dordrecht, 1994: pp. 104–152. doi:10.1007/978-94-011-
467 1256-7_5.
- 468 [17] R.A. Campo-Arnáiz, M.A. Rodríguez-Pérez, B. Calvo, J.A. De Saja, Extinction coefficient
469 of polyolefin foams, J. Polym. Sci. Part B Polym. Phys. 43 (2005) 1608–1617.
470 doi:10.1002/polb.20435.
- 471 [18] O.A. Almanza, J.A. de Saja, M.A. Rodríguez-Pérez, Prediction of the Radiation Term in
472 the Thermal Conductivity of Crosslinked Closed Cell Polyolefin Foams, J. Polym. Sci. Part
473 B Polym. Phys. 38 (2000) 993–1004. doi:10.1002/(SICI)1099-
474 0488(20000401)38:7<993::AID-POLB10>3.0.CO;2-J.
- 475 [19] V. Bernardo, J. Martín-de León, J. Pinto, T. Catelani, A. Athanassiou, M.A. Rodríguez-
476 Pérez, Low-density PMMA/MAM nanocellular polymers using low MAM contents:

- 477 Production and characterization, *Polymer (Guildf)*. 163 (2019) 115–124.
478 doi:10.1016/j.polymer.2018.12.057.
- 479 [20] J. Pinto, E. Solórzano, M.A. Rodríguez-Perez, J.A. De Saja, Characterization of the
480 cellular structure based on user-interactive image analysis procedures, *J. Cell. Plast.* 49
481 (2013) 555–575. doi:10.1177/0021955X13503847.
- 482 [21] J. Martín de-León, V. Bernardo, M.A. Rodríguez-Perez, Low Density Nanocellular
483 Polymers Based on PMMA Produced by Gas Dissolution Foaming: Fabrication and
484 Cellular Structure Characterization, *Polymers (Basel)*. 8 (2016) 1–16.
485 doi:10.3390/polym8070265.
- 486 [22] V. Bernardo, J. Martín-de León, E. Laguna-Gutierrez, T. Catelani, J. Pinto, A.
487 Athanassiou, M.A. Rodríguez-Perez, Understanding the role of MAM molecular weight
488 on the production of PMMA/MAM nanocellular polymers, *Polymer (Guildf)*. 153 (2018)
489 262–270. doi:10.1016/j.polymer.2018.08.022.
- 490 [23] V. Bernardo, F. Van Loock, J. Martín-de León, N.A. Fleck, M.A. Rodríguez-Perez,
491 Mechanical Properties of PMMA-Sepiolite Nanocellular Materials with a Bimodal
492 Cellular Structure, *Macromol. Mater. Eng.* 1900041 (2019) 1–12.
493 doi:10.1002/mame.201900041.
- 494 [24] L.J. Gibson, M. Ashby, *Cellular solids: structure and properties*, 2nd Editio, Cambridge
495 University Press, 1997.
- 496 [25] Wanqing Cao, A.J. Hunt, Improving the visible transparency of silica aerogels, *J. Non.*
497 *Cryst. Solids*. 176 (1994) 18–25. doi:10.1016/0022-3093(94)90206-2.
- 498 [26] A.J. Cox, A.J. DeWeerd, J. Linden, An experiment to measure Mie and Rayleigh total
499 scattering cross sections, *Am. J. Phys.* 70 (2002) 620–625. doi:10.1119/1.1466815.
- 500 [27] V. Kumar, N.P. Suh, A process for making microcellular parts, *Polym. Eng. Sci.* 30 (1990)
501 1323–1329. doi:https://doi.org/10.1002/pen.760302010.
- 502 [28] M. Santiago-Calvo, J. Tirado-Mediavilla, J.C. Rauhe, L.R. Jensen, J.L. Ruiz-Herrero, F.
503 Villafañe, M.Á. Rodríguez-Pérez, Evaluation of the thermal conductivity and mechanical
504 properties of water blown polyurethane rigid foams reinforced with carbon nanofibers,
505 *Eur. Polym. J.* 108 (2018) 98–106. doi:10.1016/j.eurpolymj.2018.08.051.
- 506 [29] M.E. Fajardo, C.H. Neel, D.G. Lacina, Using mid-infrared external reflectance
507 spectroscopy to distinguish between different commercially produced poly[methyl
508 methacrylate] (PMMA) samples - A null result, *AIP Conf. Proc.* 1979 (2018).
509 doi:10.1063/1.5044948.
- 510 [30] M. Alvarez-Lainez, M.A. Rodríguez-Pérez, J.A. de Saja, Thermal Conductivity of Open-
511 Cell Polyolefin Foams, *J. Polym. Sci. Part B Polym. Phys.* 46 (2008) 212–221.
512 doi:10.1002/polb.
- 513 [31] P.G. Collishaw, J.R.G. Evans, An Assessment of Expressions for the Apparent Thermal
514 Conductivity of Cellular Materials, *J. Mater. Sci.* 29 (1994) 2261–2273.
515 doi:10.1007/BF00363413.
- 516 [32] S. Song, M.M. Yovanovich, F.O. Goodman, Thermal Gap Conductance of Conforming
517 Surfaces in Contact, *J. Heat Transfer*. 115 (1993) 533–540. doi:10.1115/1.2910719.
- 518 [33] Z. Li, C. Zhu, X. Zhao, A theoretical and numerical study on the gas-contributed thermal
519 conductivity in aerogel, *Int. J. Heat Mass Transf.* 108 (2017) 1982–1990.

- 520 doi:10.1016/j.ijheatmasstransfer.2017.01.051.
- 521 [34] M. Saadatfar, C.H. Arns, M.A. Knackstedt, T. Senden, Mechanical and transport
522 properties of polymeric foams derived from 3D images, *Colloids Surfaces A*
523 *Physicochem. Eng. Asp.* 263 (2005) 284–289. doi:10.1016/j.colsurfa.2004.12.040.
- 524 [35] B. Notario, J. Pinto, E. Solorzano, J.A. de Saja, M. Dumon, M.A. Rodriguez-Perez,
525 Experimental validation of the Knudsen effect in nanocellular polymeric foams, *Polymer*
526 *(Guildf)*. 56 (2015) 57–67. doi:10.1016/j.polymer.2014.10.006.
- 527 [36] C. Forest, P. Chaumont, P. Cassagnau, B. Swoboda, P. Sonntag, Polymer nano-foams for
528 insulating applications prepared from CO₂ foaming, *Prog. Polym. Sci.* 41 (2015) 122–
529 145. doi:10.1016/j.progpolymsci.2014.07.001.
- 530 [37] V. Bernardo, J. Martin-de Leon, J. Pinto, R. Verdejo, M.A. Rodriguez-Perez, Modeling the
531 heat transfer by conduction of nanocellular polymers with bimodal cellular structures,
532 *Polymer (Guildf)*. 160 (2019) 126–137. doi:10.1016/j.polymer.2018.11.047.
- 533 [38] G. Wang, J. Zhao, L.H. Mark, G. Wang, K. Yu, C. Wang, C.B. Park, G. Zhao, Ultra-tough
534 and super thermal-insulation nanocellular PMMA/TPU, *Chem. Eng. J.* 325 (2017) 632–
535 646. doi:10.1016/j.cej.2017.05.116.
- 536 [39] S. Liu, J. Duvigneau, G.J. Vancso, Nanocellular polymer foams as promising high
537 performance thermal insulation materials, *Eur. Polym. J.* 65 (2015) 33–45.
538 doi:10.1016/j.eurpolymj.2015.01.039.
- 539 [40] S. Costeux, CO₂-blown nanocellular foams, *J. Appl. Polym. Sci.* 131 (2014) 41293(1)-
540 41293(16). doi:10.1002/app.41293.
- 541 [41] M. Arduini, J. Manara, C. Vo, Modeling of radiative properties of polystyrene foams
542 containing IR-opacifiers, *Cell. Polym.* 35 (2016) 49–66.
- 543
- 544
- 545

Supplementary Material

[Click here to download Supplementary Material: Supplementary Information_review.docx](#)

Credit Author Statement

Victoria Bernardo: Investigation, Formal analysis, Writing – Original Draft; **Judith Martin-de Leon:** Investigation, Writing – Review and Editing; **Javier Pinto:** Investigation, Writing – Review and Editing; **Ulrich Schade:** Methodology: Writing – Review and Editing; **Miguel Angel Rodriguez-Perez:** Conceptualization, Supervision, Writing – Review and Editing

Declaration of interests

The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests: